

November 6, 2009: Cheiron School 2009 @ SPring-8

# **Small-Angle X-ray Scattering**

## **Basics & Applications**

Yoshiyuki Amemiya and Yuya Shinohara  
Graduate School of Frontier Sciences,  
The University of Tokyo

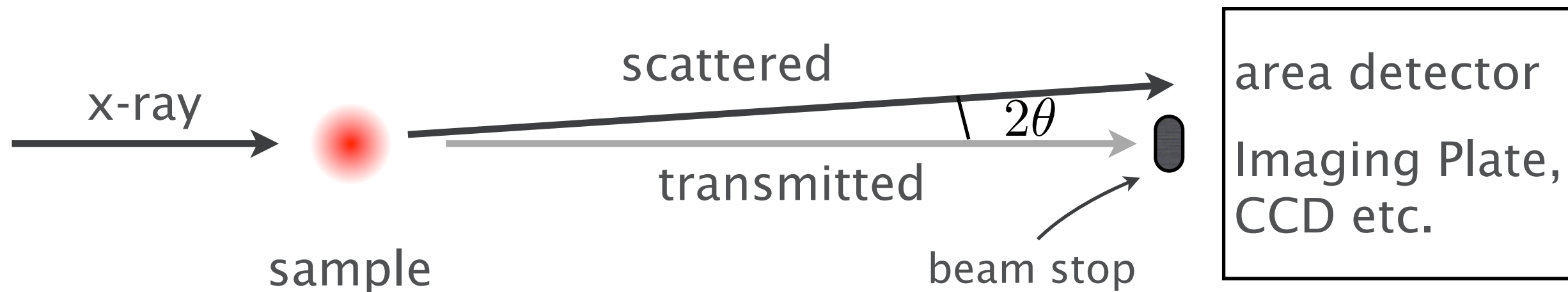
# Overview

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- ❧ Introduction
  - ❧ What's SAXS ?
  - ❧ History
  - ❧ Application field of SAXS
- ❧ Theory
  - ❧ Structural Information obtained by SAXS
- ❧ Experimental Methods
  - ❧ Optics
  - ❧ Detectors
- ❧ Advanced SAXS
  - ❧ Microbeam, GI-SAXS, USAXS, XPCS etc...



# What's **S**mall-**A**ngle **X**-ray **S**cattering ?



Bragg's law:  $\lambda = 2d \sin \theta$

small angle  $\longrightarrow$  large structure  
(1 – 100 nm)

crystalline sample --> small-angle X-ray diffraction: SAXD

solution scattering / inhomogeneous structure --> SAXS



# History of SAXS (< 1936)

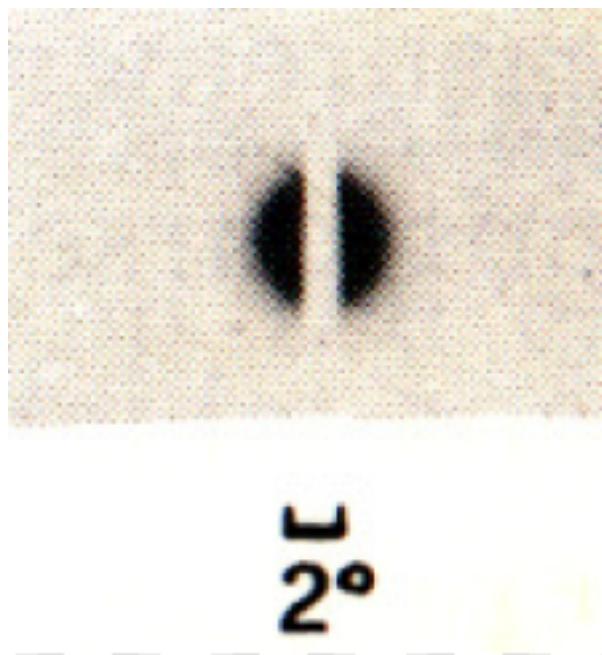
Krishnamurty (1930)

Hendricks (1932)

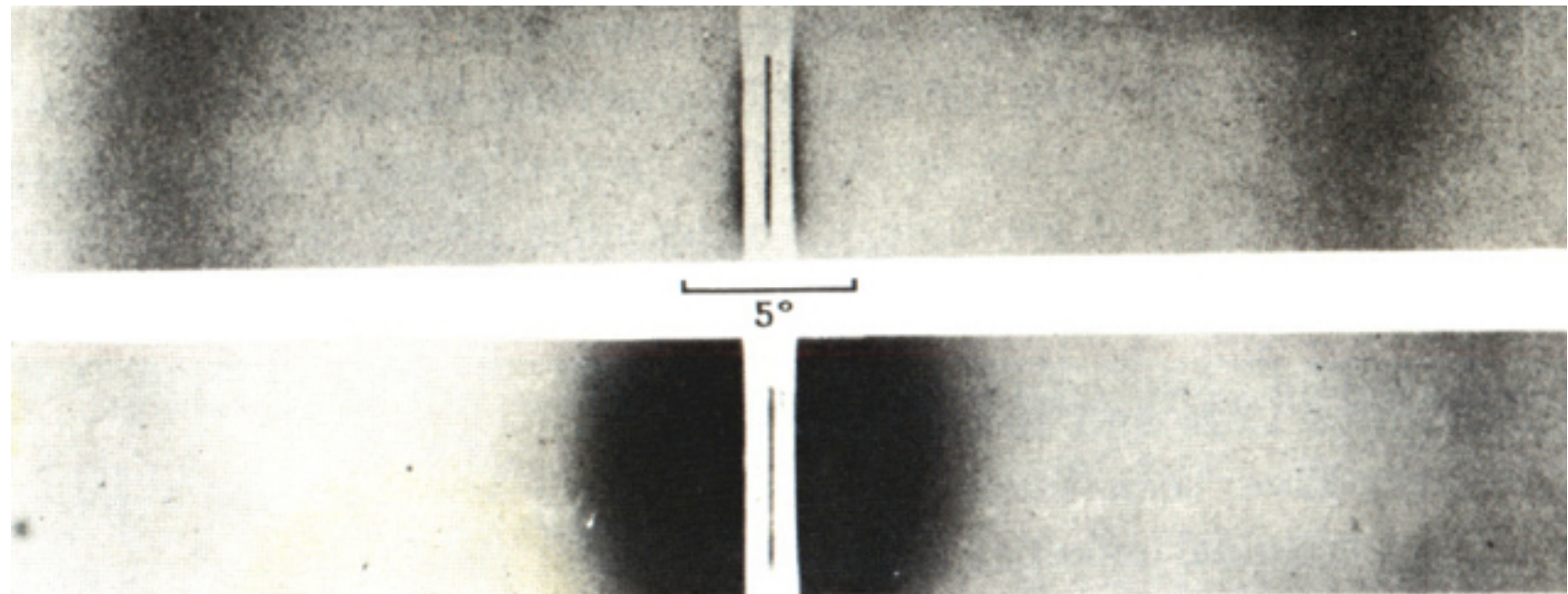
Mark (1932)

Warren (1936)

Observation of scattering  
from powders, fibers, and colloidal  
dispersions



carbon black

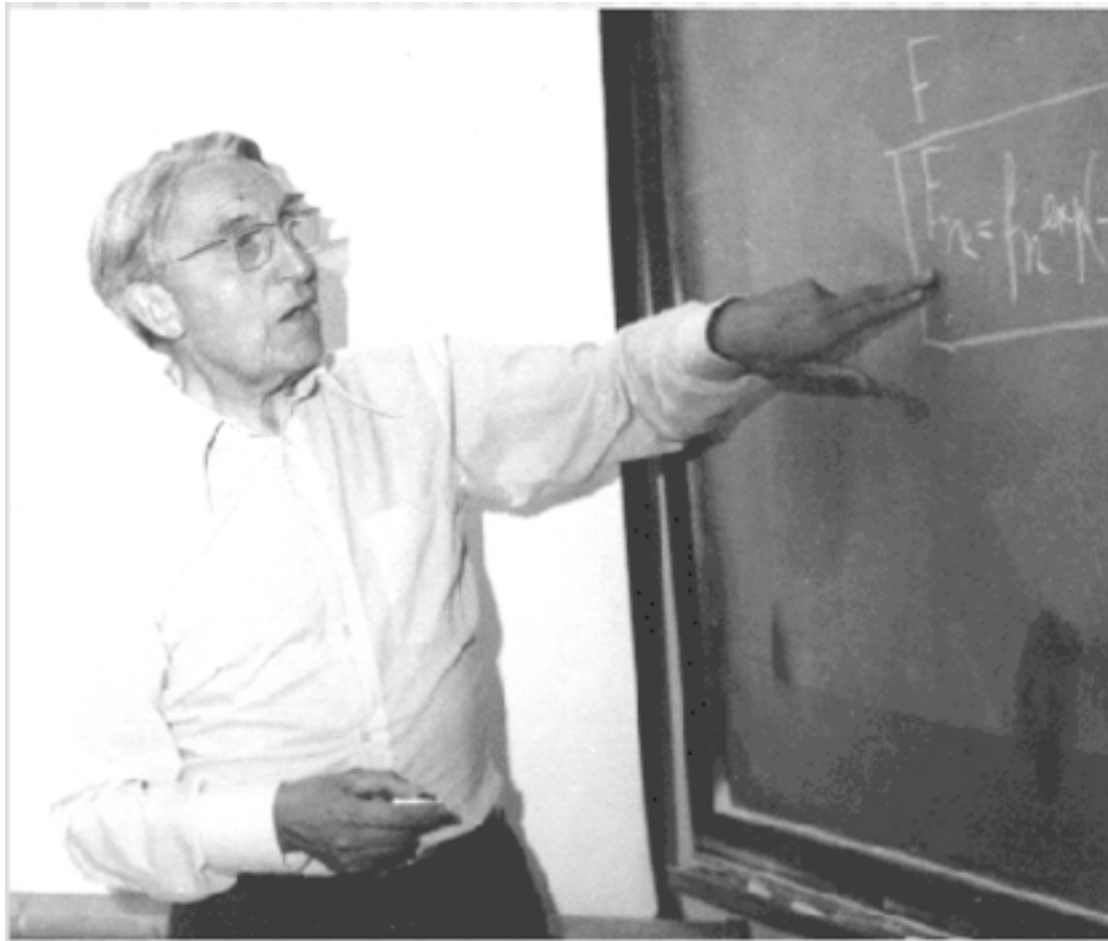


Molten silica - silica gel





# History (> 1936)



[A. Guinier](#) (1937, 1939, 1943)

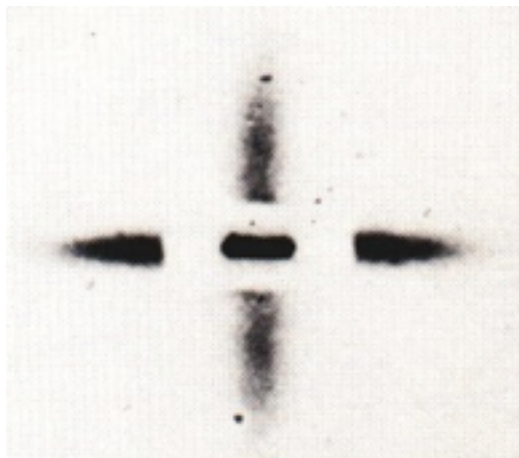
Interpretation of inhomogeneities in Al alloys “G-P zones”, introducing the concept of “particle scattering” and formalism necessary to solve the problem of a diluted system of particles.

[O. Kratky](#) (1938, 1942, 1962)

[G. Porod](#) (1942, 1960, 1961)

Description of dense systems of colloidal particles, micelles, and fibers.

Macromolecules in solution.



Single crystals of Al-Cu hardened alloy

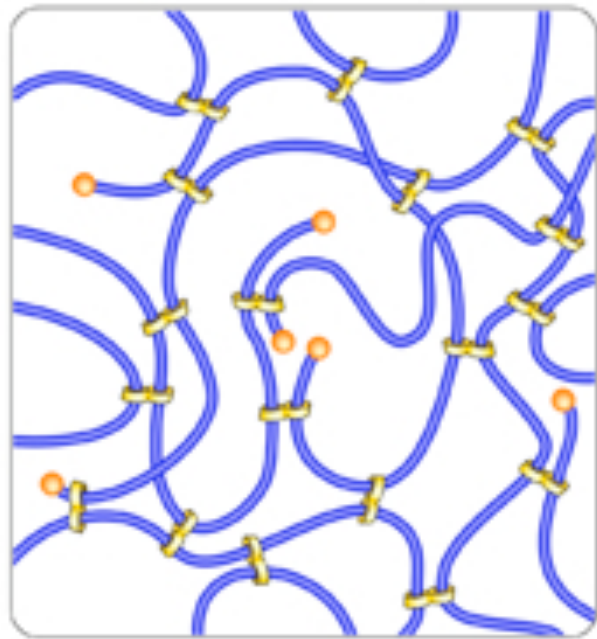


Hemoglobin

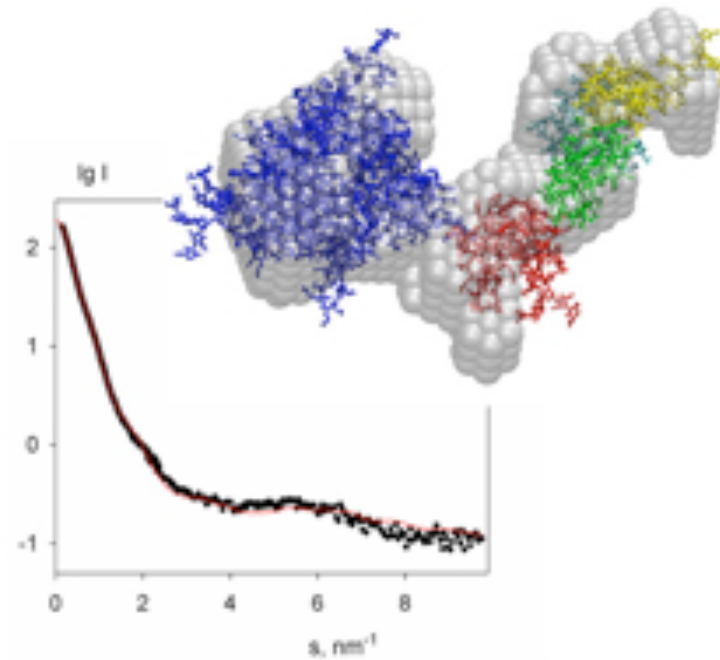


courtesy to Dr. I.L.Torriani

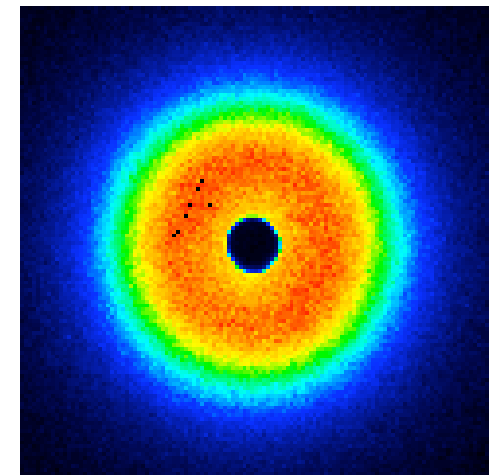
# Application of SAXS



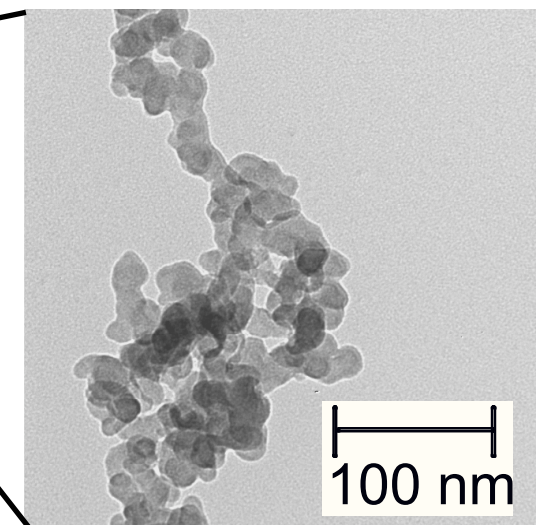
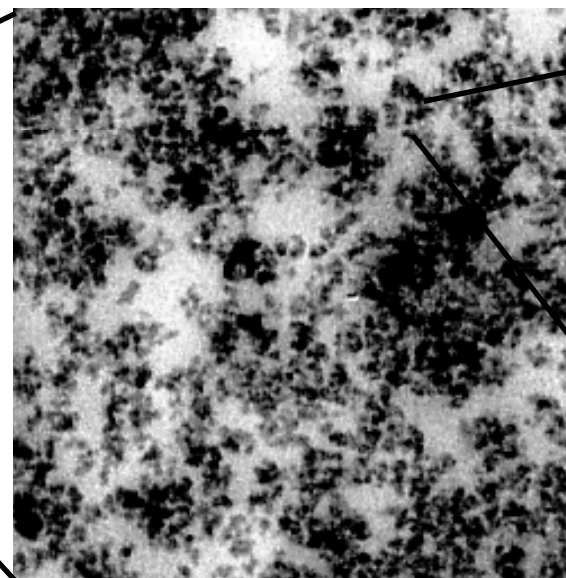
gel



Proteins in solution (Dr. Svergun, EMBL)



Typical SAXS image



Nanocomposite



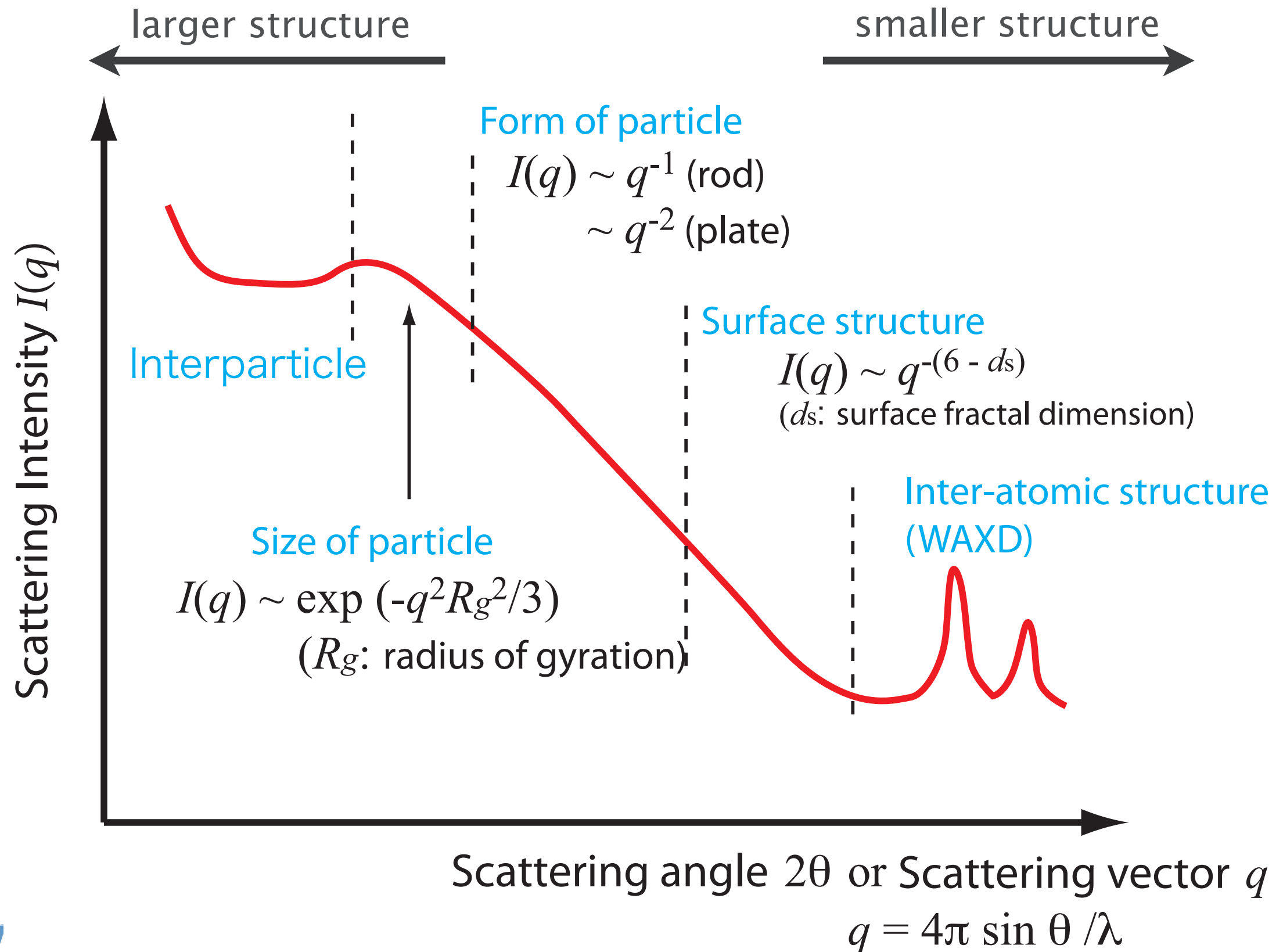
# Application of SAXS

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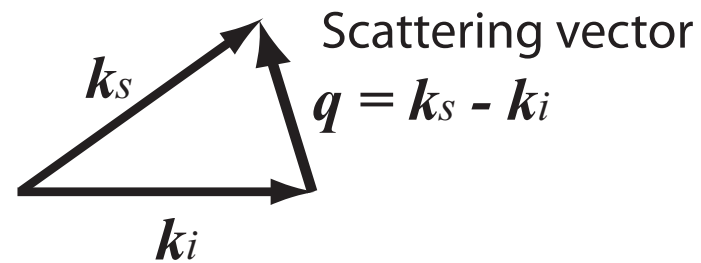
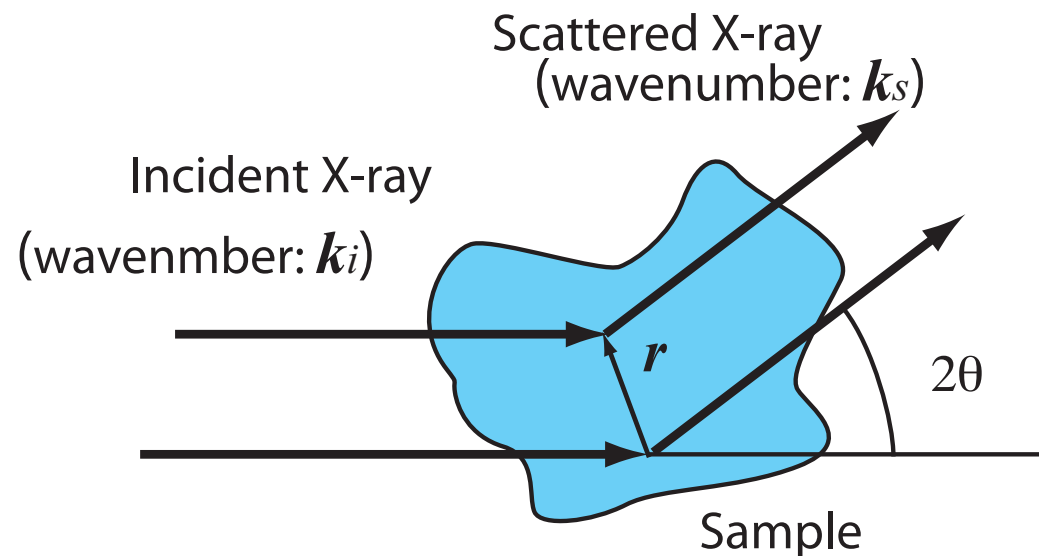
- Size and form of particulate system
  - Colloids, Globular proteins, etc...
- Inhomogeneous structure
  - Polymer chain, two-phase system etc.
- Distorted crystalline structure
  - Crystal of soft matter



# SAXS of particulate system



# Basic of X-ray scattering



$\rho(\mathbf{r})$ : electron density

$$q = |\mathbf{q}| = 4\pi \sin \theta / \lambda$$

Amplitude of scattered X-ray

$$A(\mathbf{q}) = \int_V \rho(\mathbf{r}) \exp(-i\mathbf{q} \cdot \mathbf{r}) d\mathbf{r}$$

Fourier transform of electron density

Scattering intensity per unit volume: 
$$I(\mathbf{q}) = \frac{A(\mathbf{q})A^*(\mathbf{q})}{V}$$





# Correlation Function & Scattering Intensity

Correlation function of electron density per unit volume

$$\gamma(\mathbf{r}) = \frac{1}{V} \int_V \rho(\mathbf{r}') \rho(\mathbf{r} + \mathbf{r}') d\mathbf{r}' = \frac{1}{V} \underline{P(\mathbf{r})}$$

Patterson Function

(Debye & Bueche 1949)

asymptotic behavior of the correlation function

$$\gamma(\mathbf{r} = 0) = \langle \rho^2 \rangle \qquad \gamma(\mathbf{r} \rightarrow \infty) \rightarrow \langle \rho \rangle^2$$

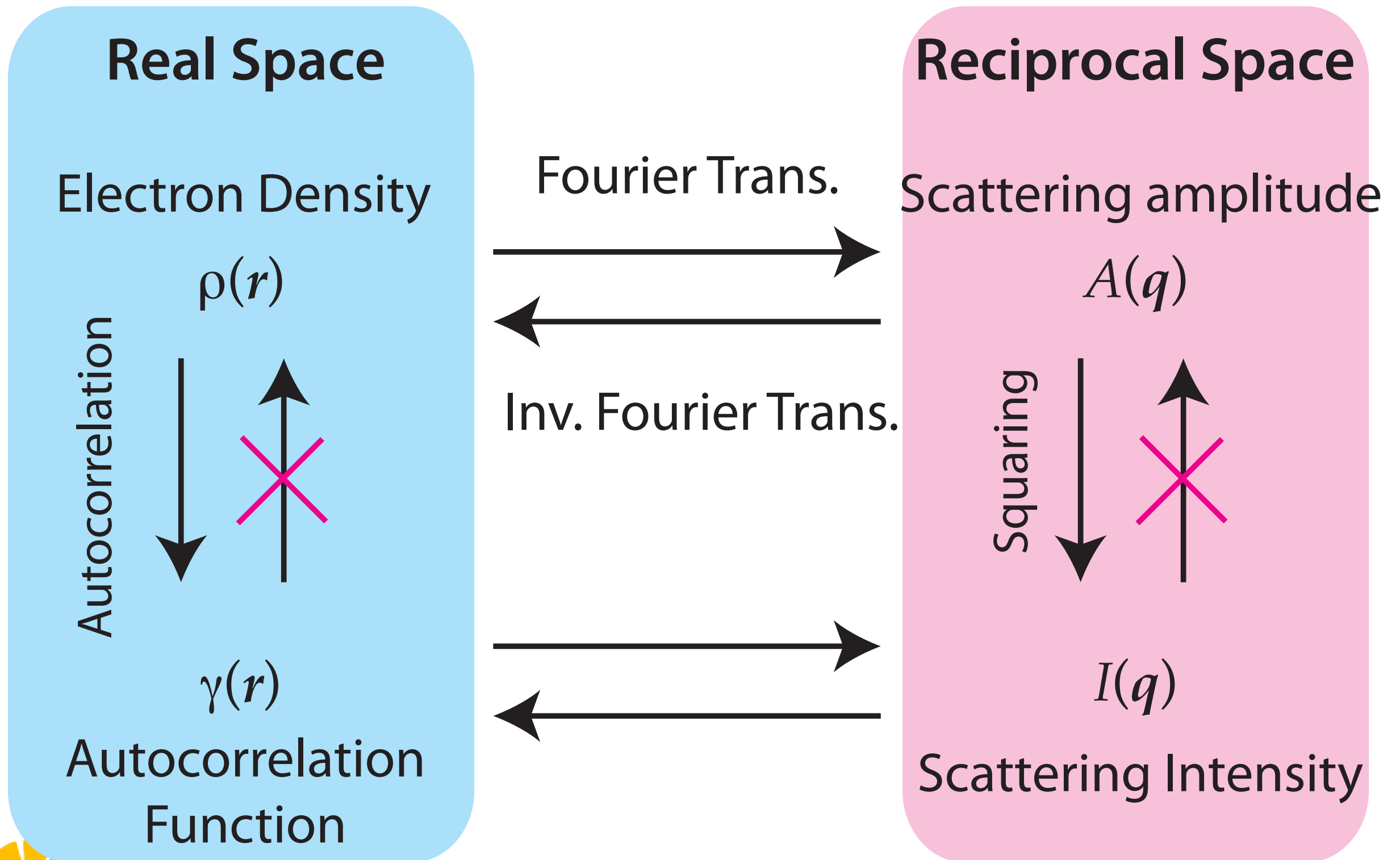
Scattering Intensity : Fourier Transform of correlation function

$$I(\mathbf{q}) = \int_V \gamma(\mathbf{r}) \exp(-i\mathbf{q} \cdot \mathbf{r}) d\mathbf{r}$$



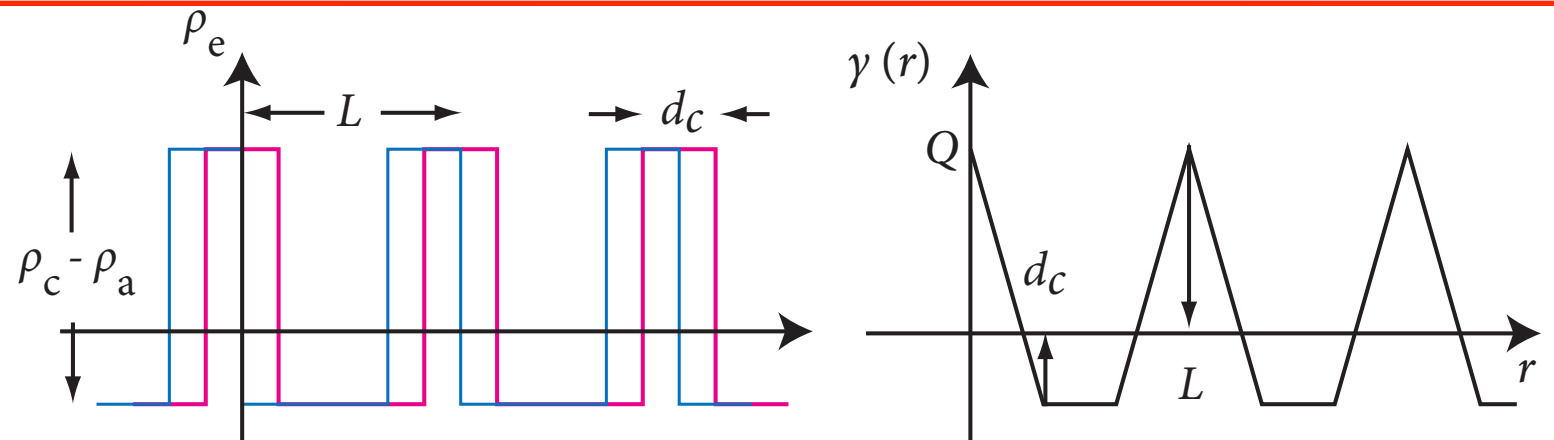


# Real space and Reciprocal Space

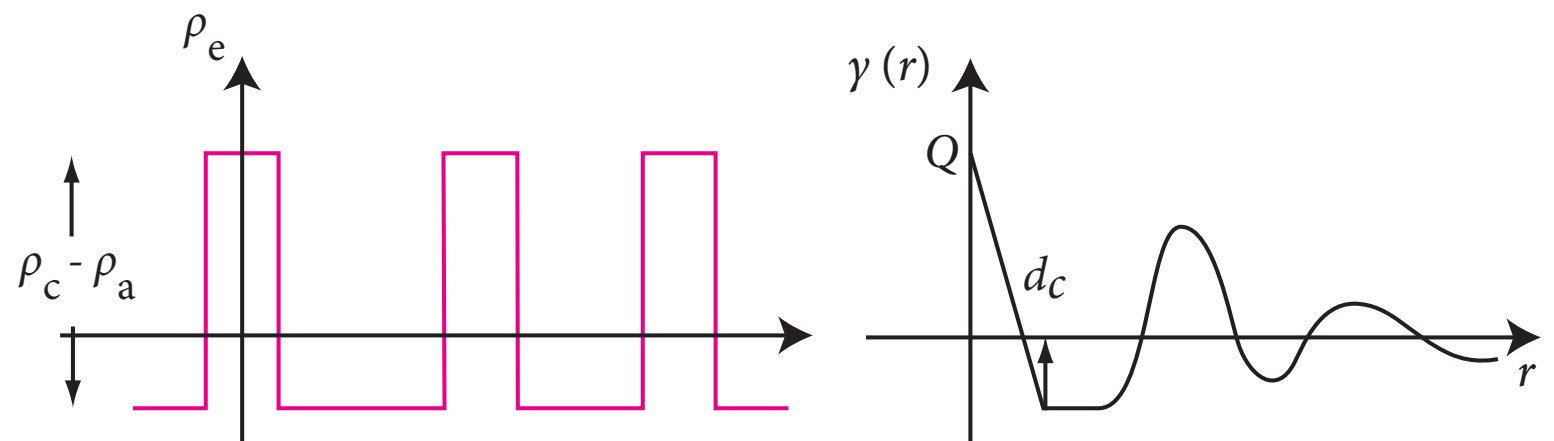


# Diffraction from Lamellar Structure

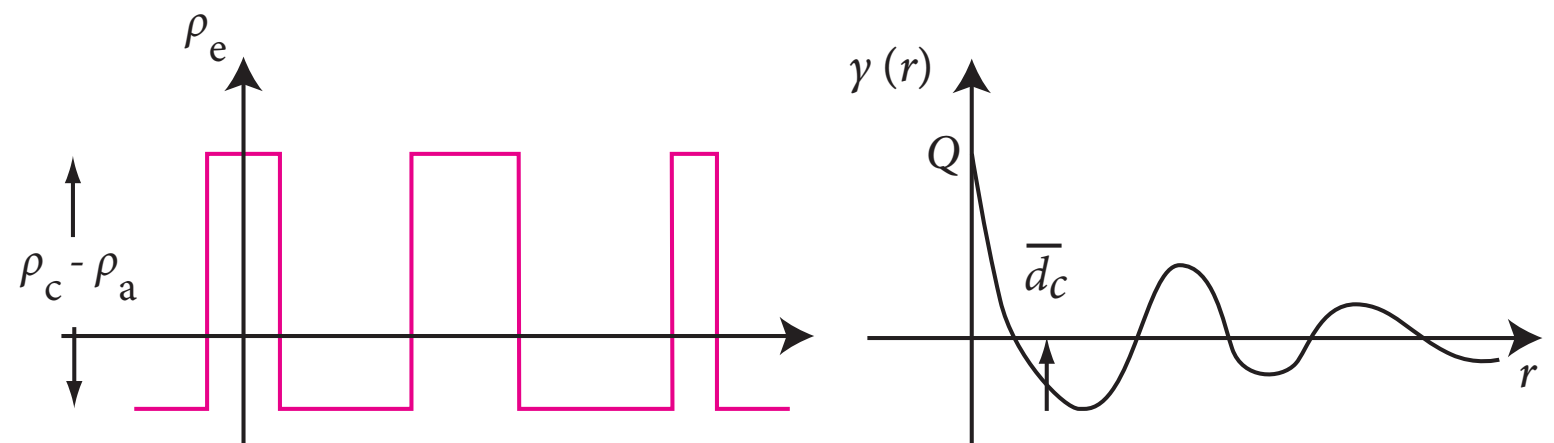
ideal ordering



Long period changes.



Thickness of crystal changes.



real space

autocorrelation



# Normalized Correlation Function

Local electron density fluctuations:  $\eta(\mathbf{r}) = \rho(\mathbf{r}) - \langle \rho \rangle$

$$\longrightarrow \langle \eta^2 \rangle = \langle (\rho(\mathbf{r}) - \langle \rho \rangle)^2 \rangle = \langle \rho^2 \rangle - \langle \rho \rangle^2$$

average density fluctuations

Normalized Correlation Function

$$\gamma_0(\mathbf{r}) = \frac{\rho(\mathbf{r}) - \langle \rho \rangle}{\langle \eta^2 \rangle} = \frac{1}{\langle \eta^2 \rangle} \frac{1}{V} \int_V \eta(\mathbf{r}') \eta(\mathbf{r} + \mathbf{r}') d\mathbf{r}'$$

substitution  $\longrightarrow I(\mathbf{q}) = \int_V \rho(\mathbf{r}) \exp(-i\mathbf{q} \cdot \mathbf{r}) d\mathbf{r}$

$$I(\mathbf{q}) = \underbrace{\langle \eta^2 \rangle}_{\text{Only the average density fluctuations contribute to the signal.}} \int_V \gamma_0(\mathbf{r}) e^{-i\mathbf{q} \cdot \mathbf{r}} d\mathbf{r} + \underbrace{\langle \rho \rangle^2 \delta(\mathbf{q})}_{\text{Not observable.}}$$



Only the average density fluctuations contribute to the signal.

Not observable.

# Invariant Q

$$I(\mathbf{q}) = \langle \eta^2 \rangle \int_V \gamma_0(\mathbf{r}) e^{-i\mathbf{q} \cdot \mathbf{r}} d\mathbf{r} + \underbrace{\langle \rho \rangle^2 \delta(\mathbf{q})}_{\text{Omitted.}}$$

Parseval's equality

$$\int I(\mathbf{q}) d\mathbf{q} = (2\pi)^3 \langle \eta^2 \rangle$$

$$4\pi \int I(q) q^2 dq$$

Parseval's equality

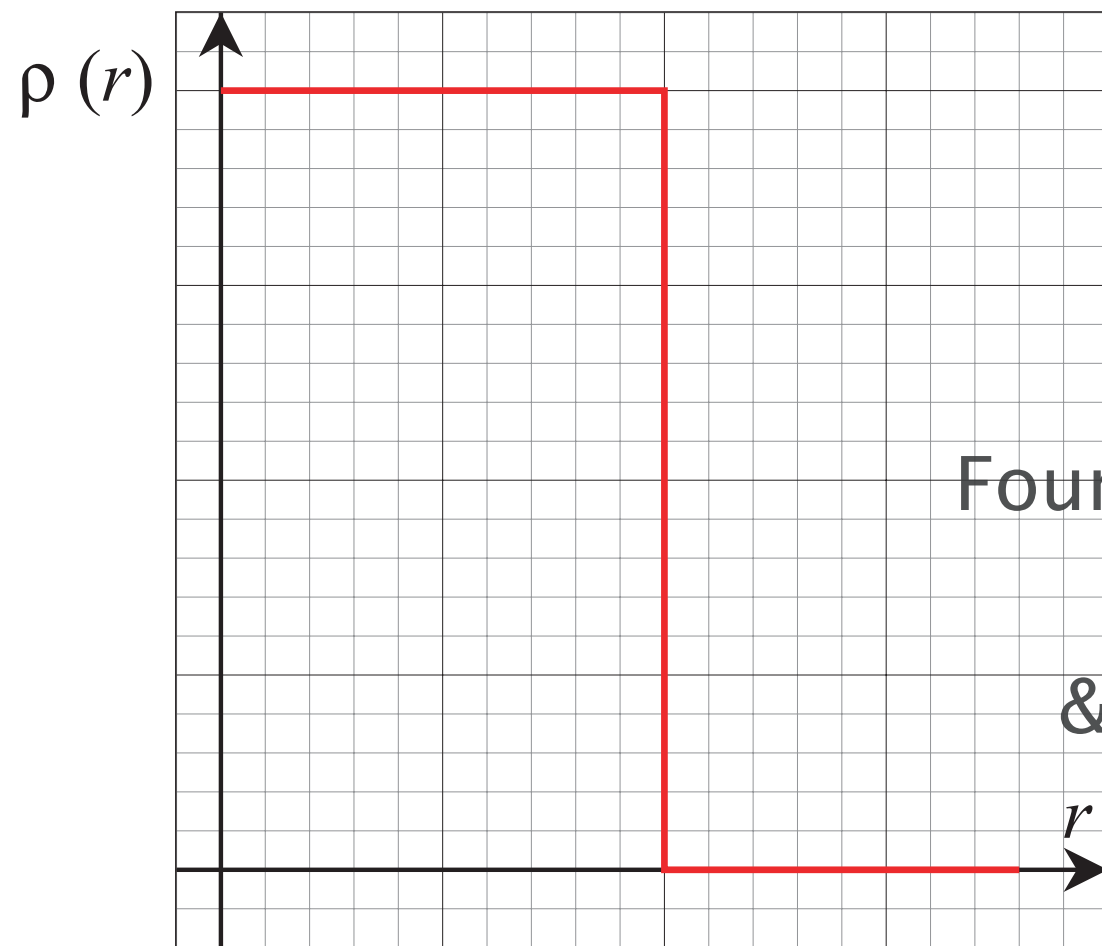
$$A(\mathbf{q}) \xleftrightarrow{\text{Fourier Trans.}} \eta(\mathbf{r})$$

$$\int |A(\mathbf{q})|^2 d\mathbf{q} = (2\pi)^3 \int |\eta(\mathbf{r})|^2 d\mathbf{r}$$

**Invariant:**  $Q = \int_0^\infty I(q) q^2 dq = 2\pi^2 \langle \eta^2 \rangle$



# Spherical sample

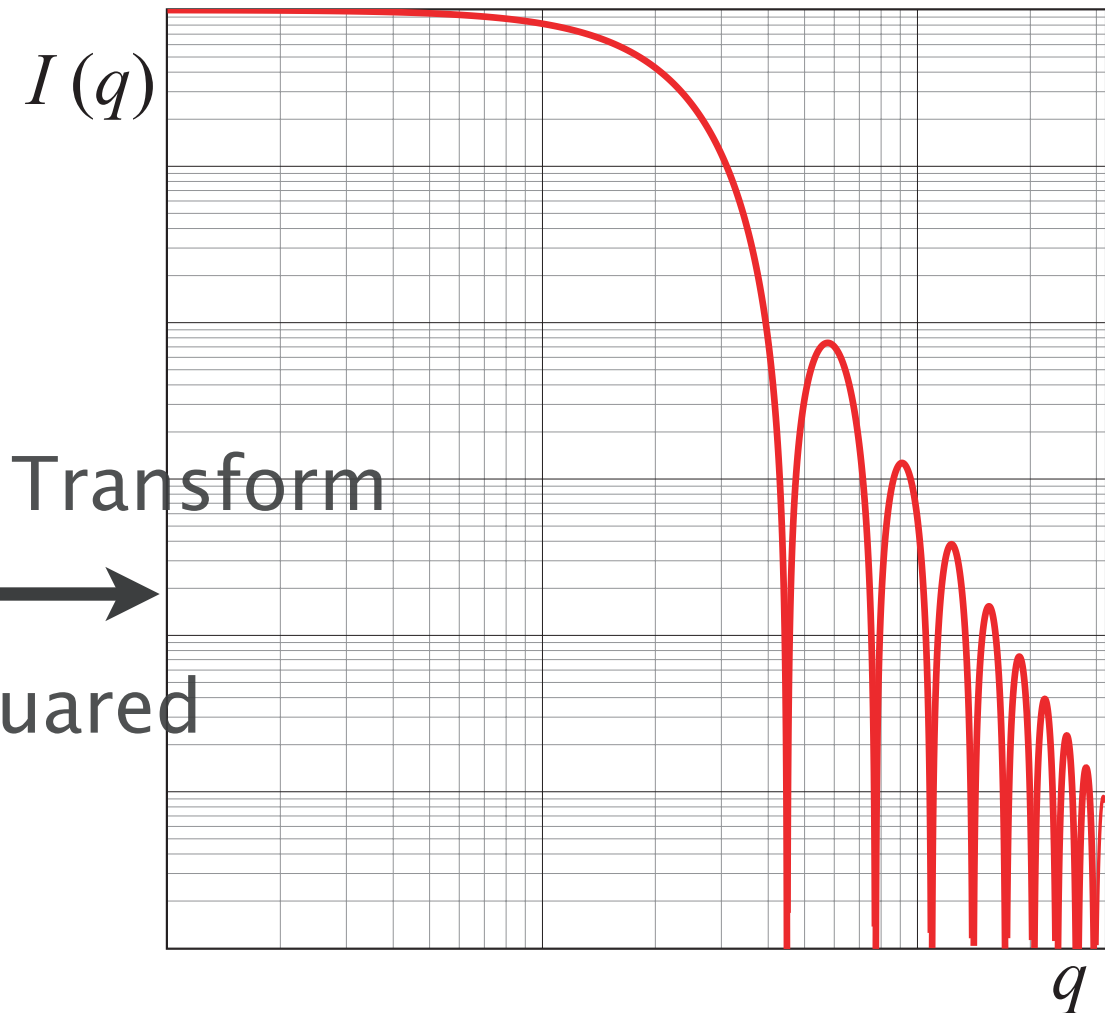


$$\rho(r) = \begin{cases} \Delta\rho & r < R \\ 0 & \text{else} \end{cases}$$

Fourier Transform



& squared

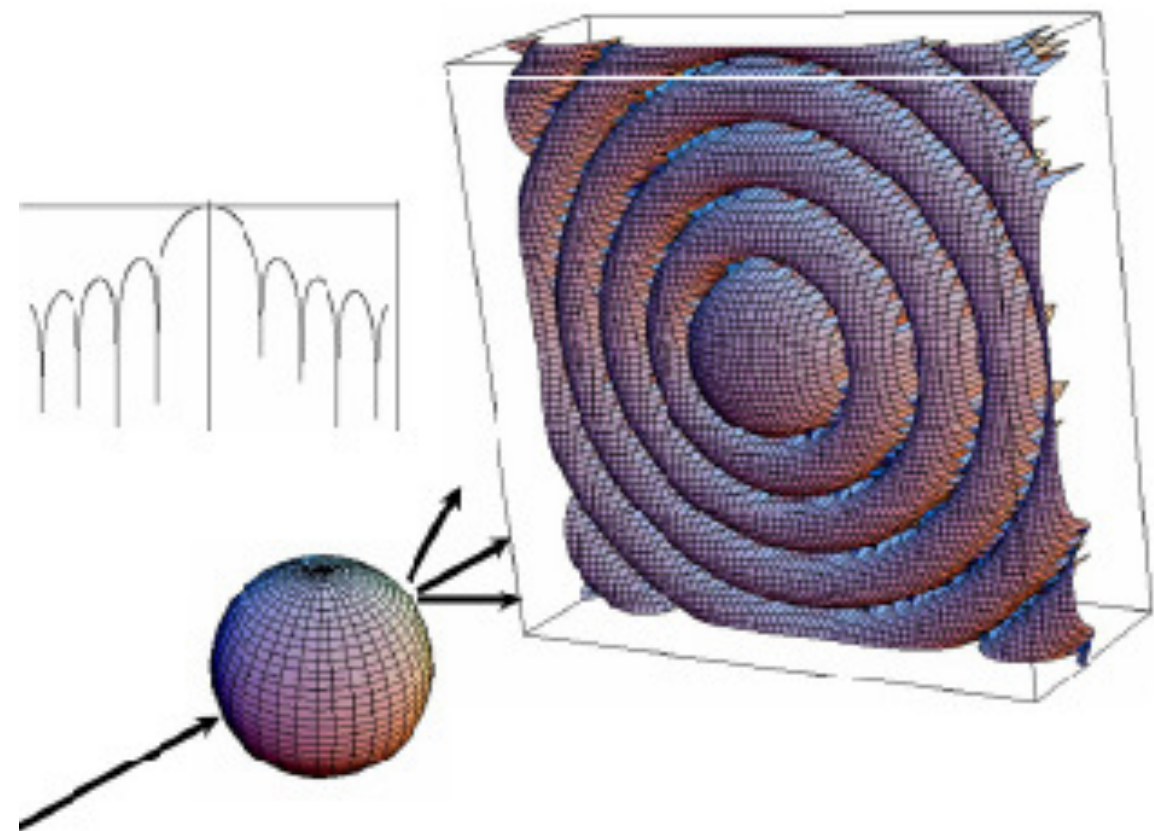
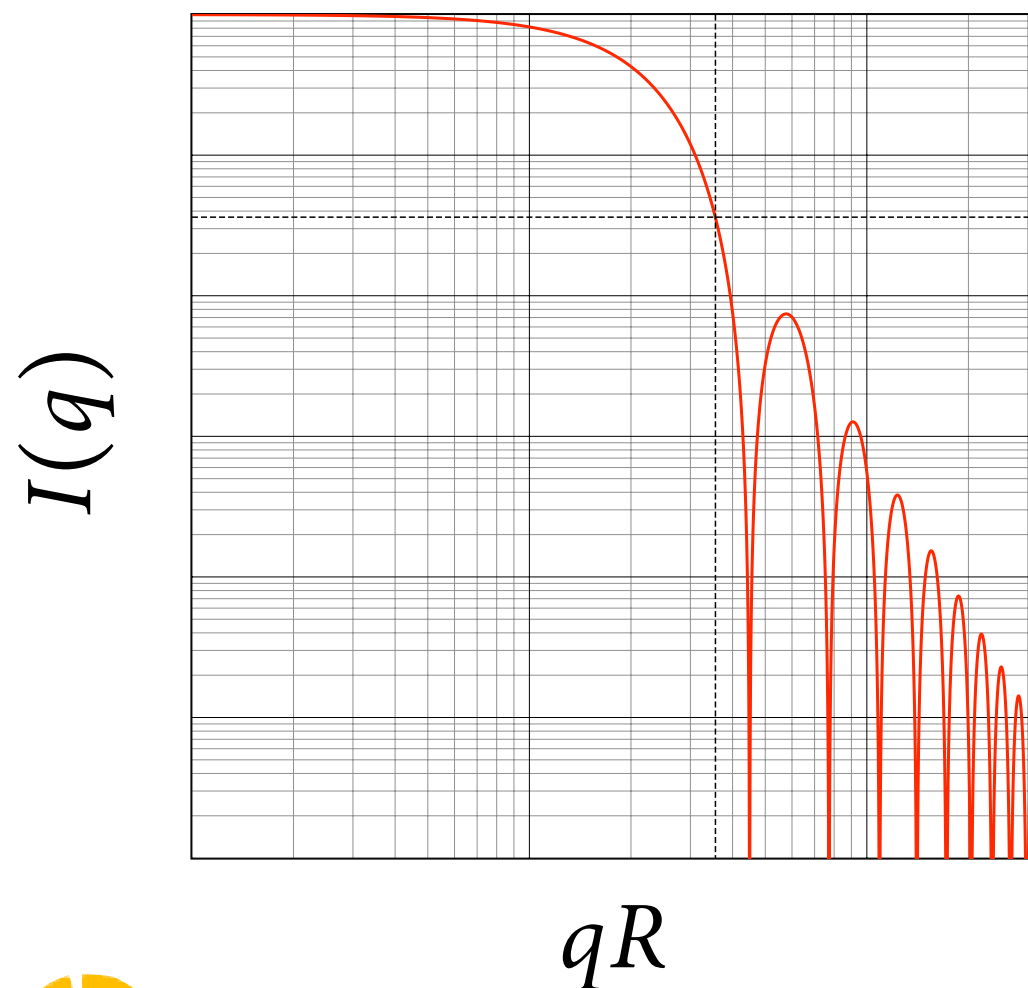


$$I(q) = \frac{(\Delta\rho)^2 V_{\text{particle}}^2}{V} \left[ 3 \frac{\sin qR - qR \cos qR}{(qR)^3} \right]$$



# Homogeneous sphere

$$I(q) = \frac{(\Delta\rho)^2 V_{\text{particle}}^2}{V} \left[ 3 \frac{\sin qR - qR \cos qR}{(qR)^3} \right]$$



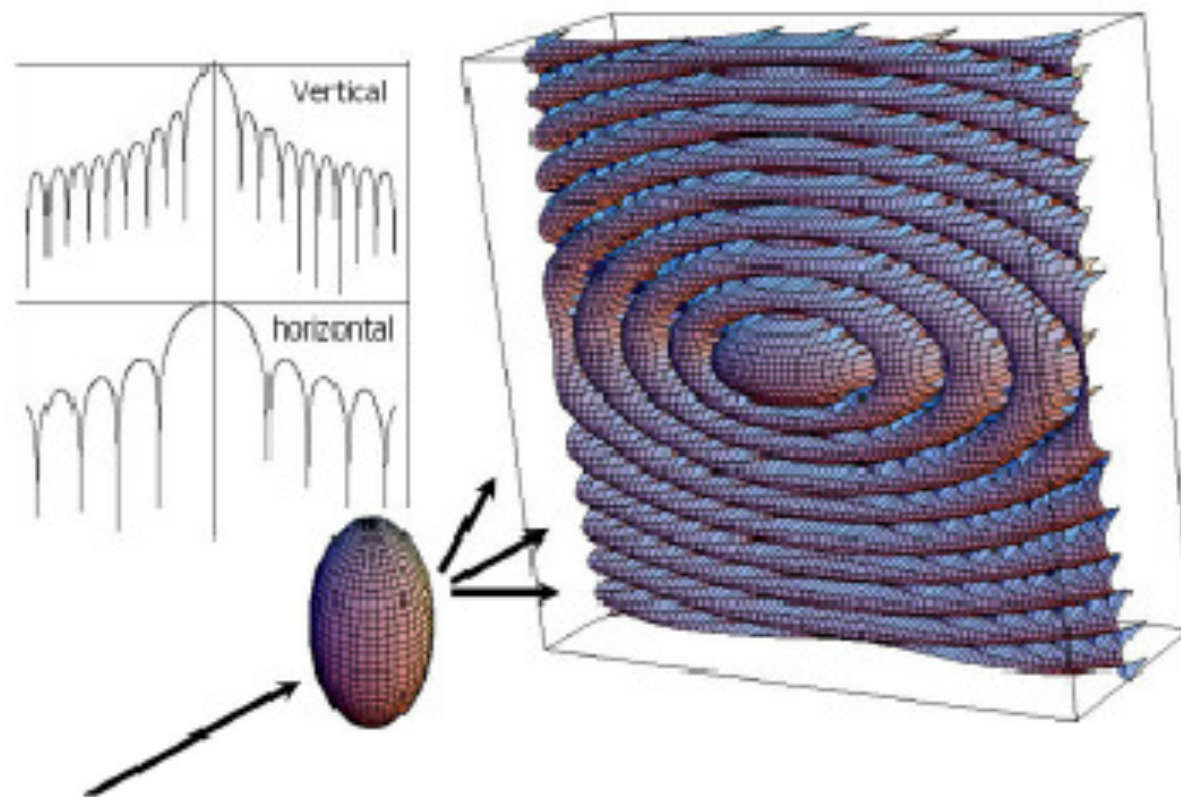
isotropic scattering





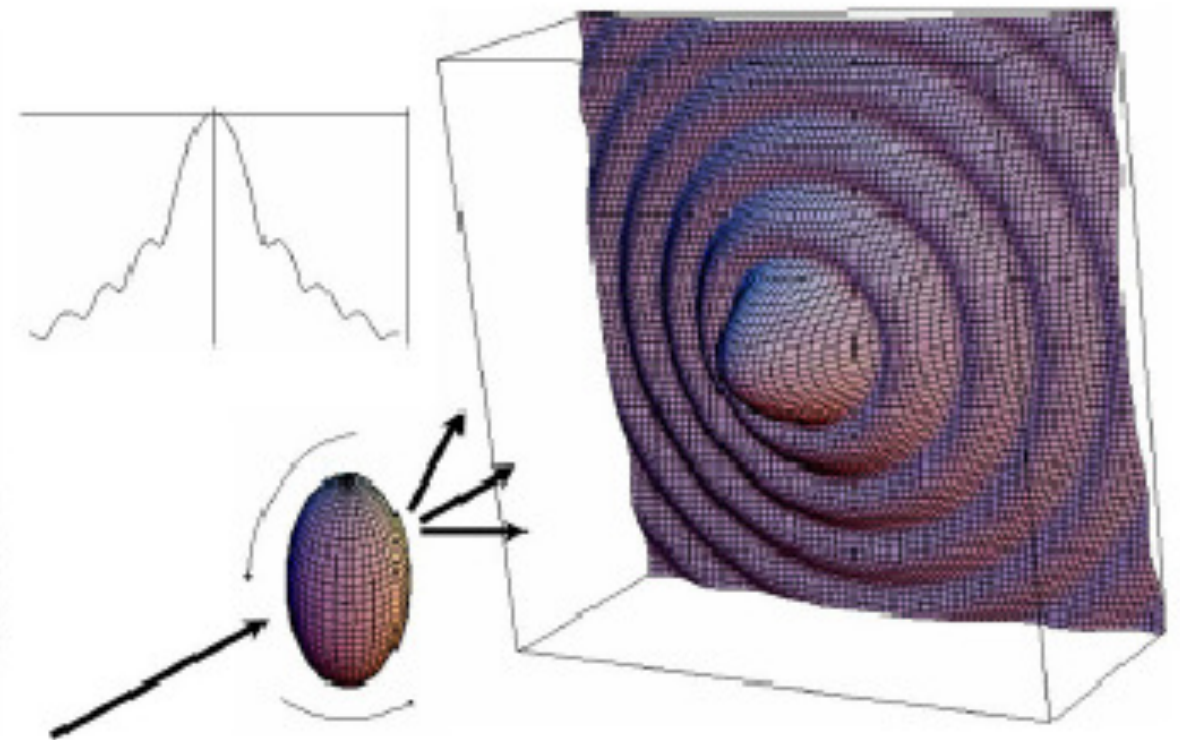
# Homogeneous elipsoid

Fixed particle



anisotropic scattering

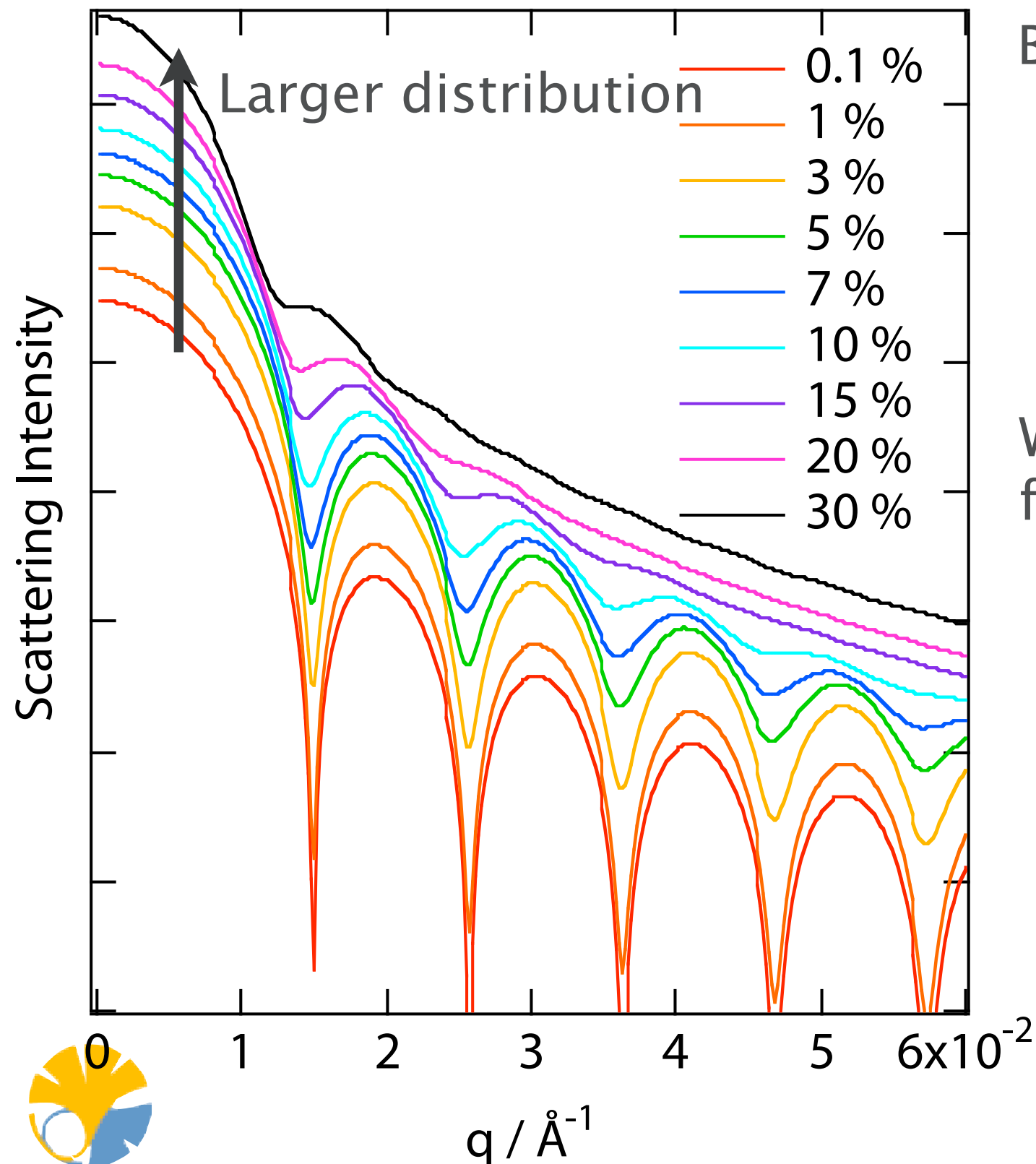
Random orientation



isotropic scattering



# Size distribution

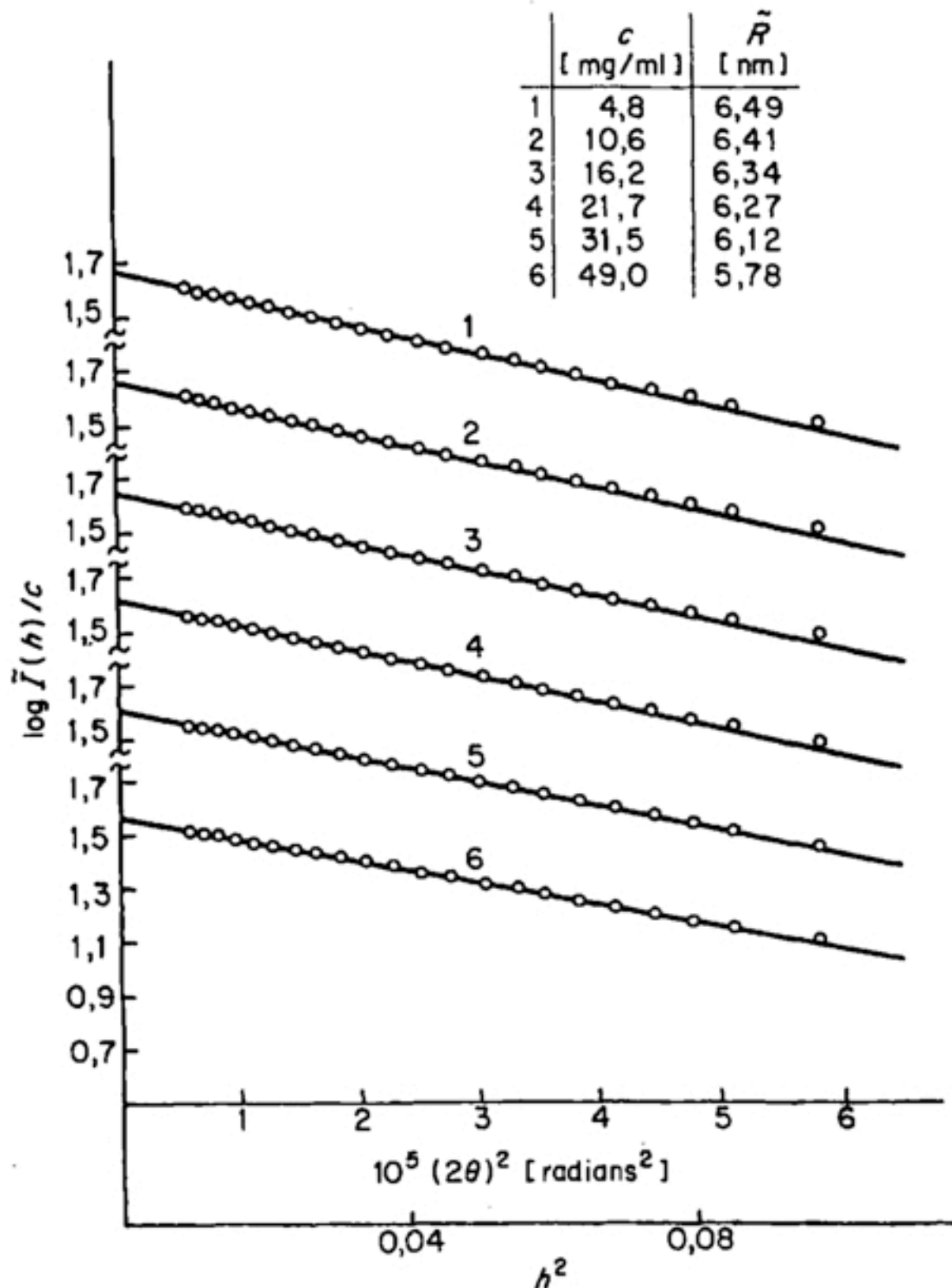


Based on Gaussian distribution

When the form has distribution, fringes are missed.



# Radius of Gyration -- Guinier Plot



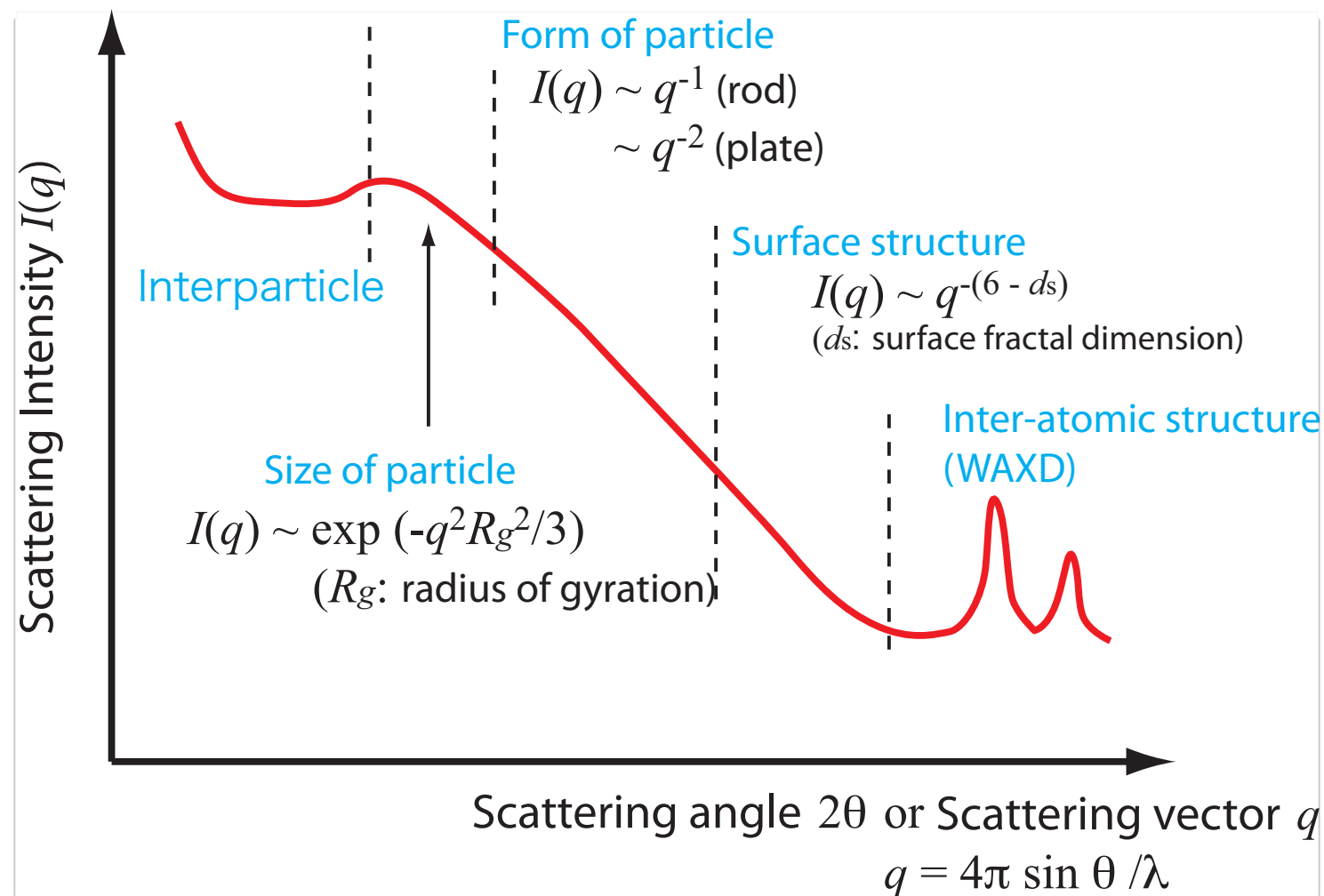
$$I(q) \sim \exp\left(-\frac{q^2 R_g^2}{3}\right)$$

$$\log(I(q)) = -\frac{q^2 R_g^2}{3}$$

Guinier plot:  $\log(I(q))$  vs  $q^2$

O. Glatter & O. Kratky ed., "Small Angle X-ray Scattering", Academic Press (1982).

# Structure Factor & Form Factor



$$I(q) = \phi V_{\text{particle}} \underline{S(q)} \underline{F(q)}$$

Structure Factor      Form Factor

↓

inter-particle structure      intra-particle structure

Separation of  $S(q)$  &  $F(q)$

→ Everlasting issue  
 (especially, for non-crystalline sample)

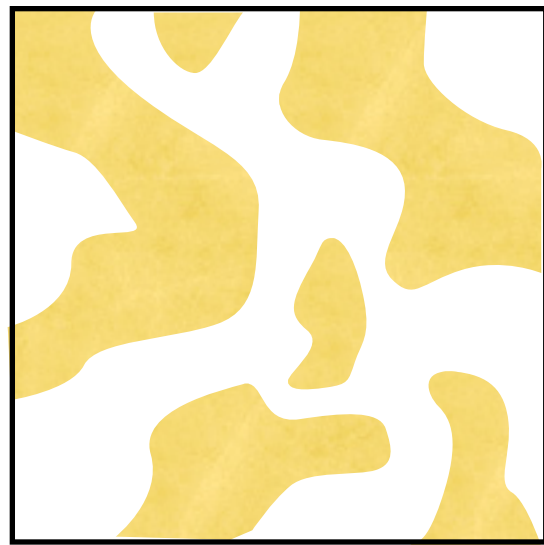
Proposed remedy:

- GIFT (Generalized Inverse Fourier Trans.) by O. Glatter



# Scattering from Inhomogeneous Structure

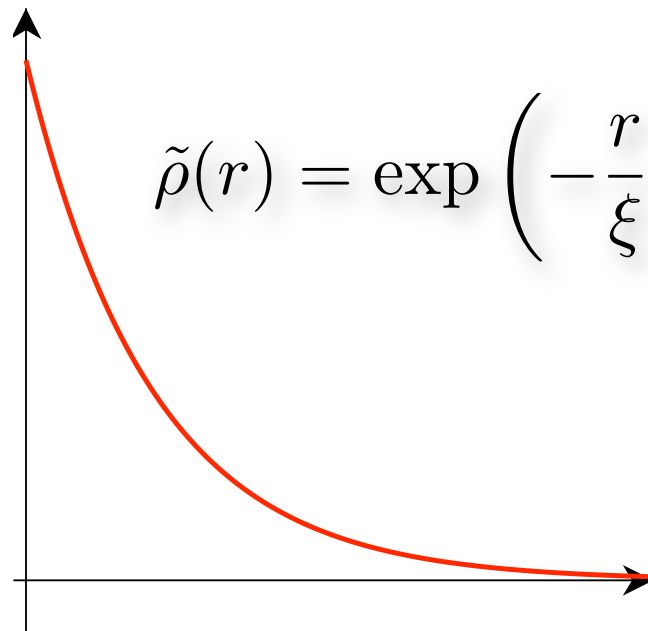
## Electron Density



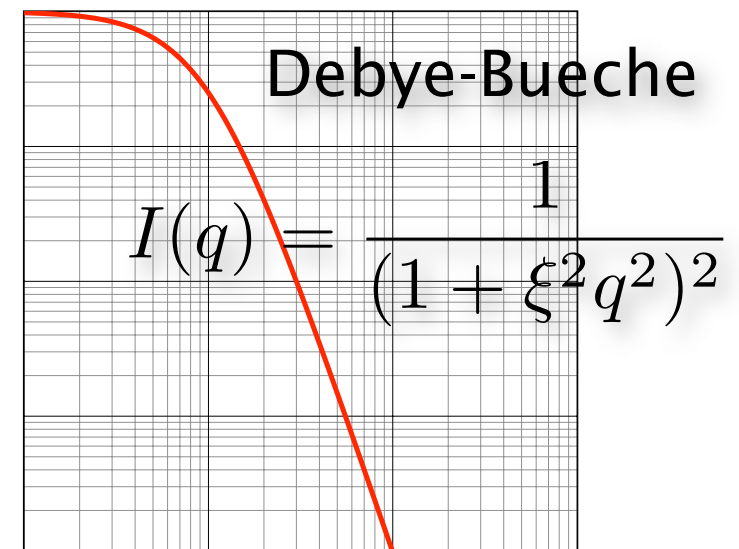
two phase system

## Autocorrelation Function

$$\tilde{\rho}(r) = \exp\left(-\frac{r}{\xi}\right)$$

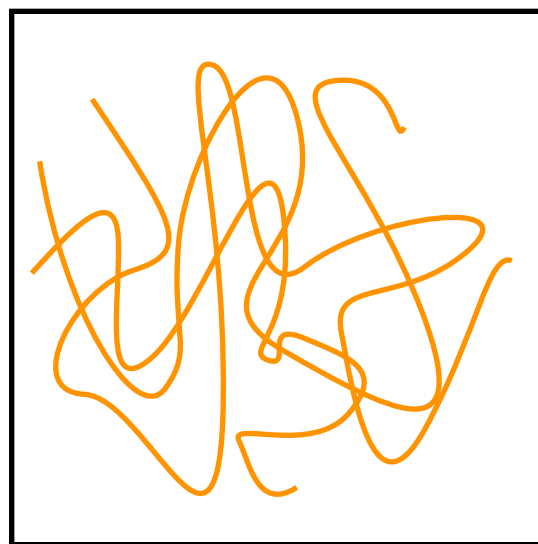


## Scattering Intensity



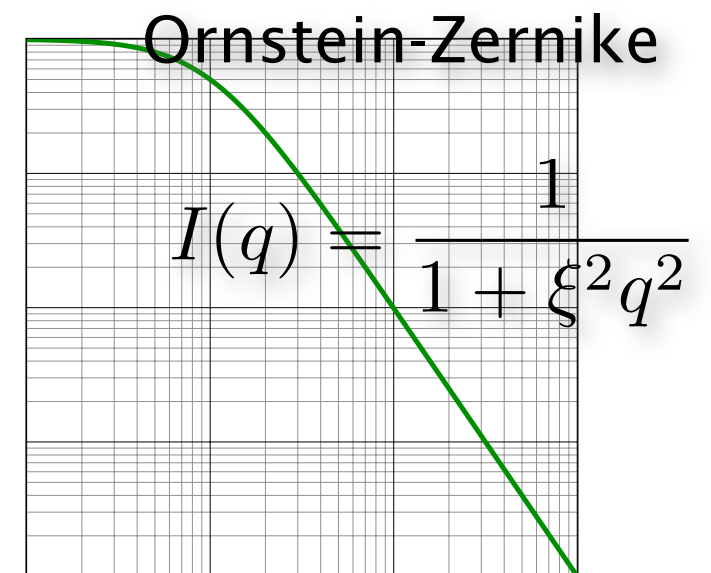
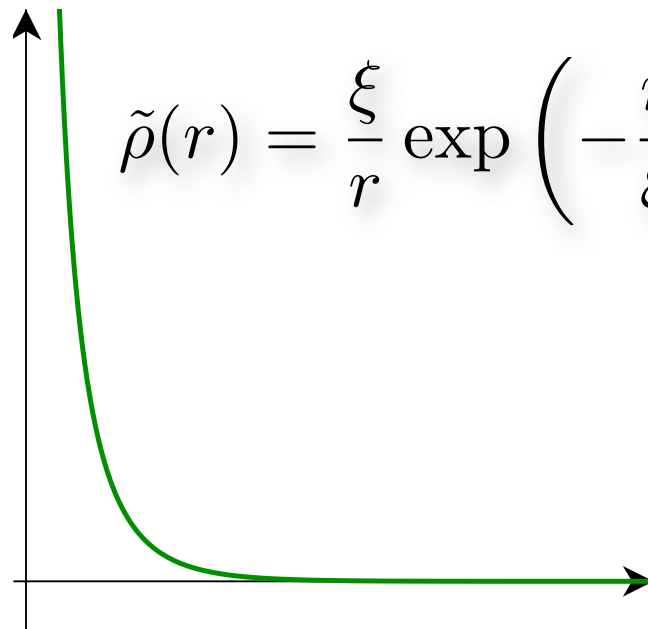
Autocorrelation

Fourier trans.



polymer chain etc.

$$\tilde{\rho}(r) = \frac{\xi}{r} \exp\left(-\frac{r}{\xi}\right)$$



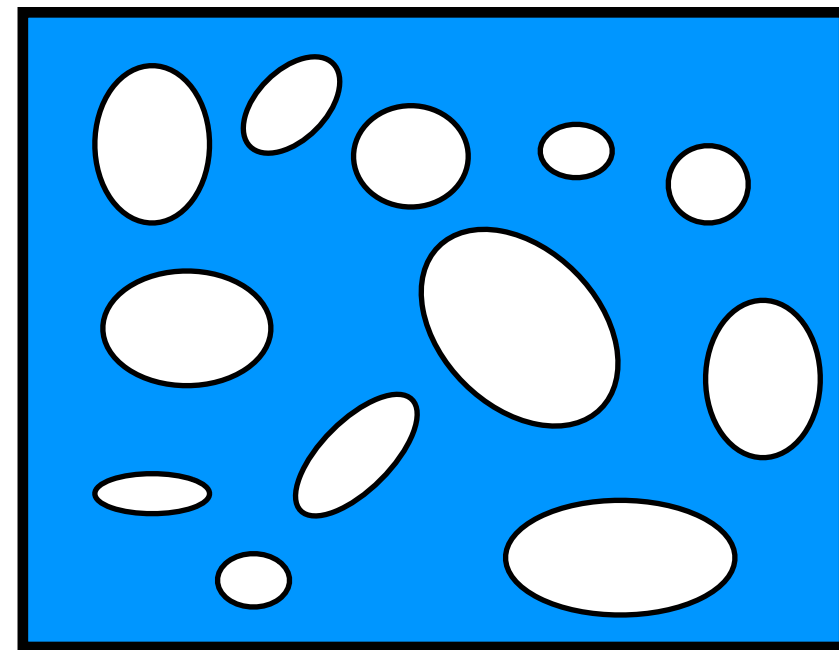
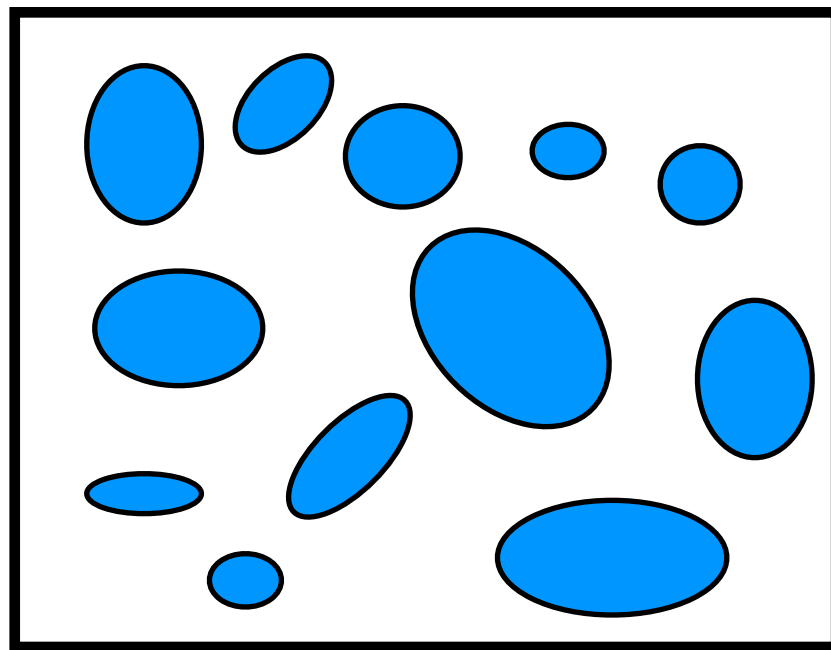


# Two-phase system

Phase 1:  $\rho_1$ , volume fraction  $\phi$     Phase 2:  $\rho_2$  volume fraction  $1 - \phi$

$$A(\mathbf{q}) = \int_{\phi V} \rho_1 e^{-i\mathbf{q}\cdot\mathbf{r}} d\mathbf{r} + \int_{(1-\phi)V} \rho_2 e^{-i\mathbf{q}\cdot\mathbf{r}} d\mathbf{r}$$
$$= \int_{\phi V} (\rho_1 - \rho_2) e^{-i\mathbf{q}\cdot\mathbf{r}} d\mathbf{r} + \rho_2 \int_V e^{-i\mathbf{q}\cdot\mathbf{r}} d\mathbf{r}$$

$$A(\mathbf{q}) = \int_V \Delta\rho e^{-i\mathbf{q}\cdot\mathbf{r}} d\mathbf{r} + \rho_2 \delta(\mathbf{q})$$



## Babinet's principle



Two complementary structures produce the same scattering.



# Two-phase system -- cont.

Averaged square fluctuation of electron density

$$\langle \eta^2 \rangle = \phi(1 - \phi)(\Delta\rho)^2 \quad \text{where} \quad \Delta\rho = \rho_1 - \rho_2$$

$$I(q) = 4\pi \langle \eta^2 \rangle \int_0^\infty \gamma_0(r) \frac{\sin(qr)}{qr} r^2 dr$$

$$I(q) = 4\pi \phi(1 - \phi)(\Delta\rho)^2 \int_0^\infty \gamma_0(r) \frac{\sin(qr)}{qr} r^2 dr$$

$$Q = \int_0^\infty I(q) q^2 dq = 2\pi^2 \phi(1 - \phi)(\Delta\rho)^2$$

**Invariant:** does not depend on the structure of the two phases but only on the **volume fractions** and **the contrast between the two phases**.



# Porod's law

For a sharp interface, the scattered intensity decreases as  $q^{-4}$ .

$$I(q) \rightarrow (\Delta\rho)^2 \frac{2\pi}{q^4} \underline{S/V}$$

internal surface area

Combination of Porod's law & Invariant

$$\pi \cdot \frac{\lim_{q \rightarrow \infty} I(q) q^4}{Q} = \boxed{\frac{S}{V}}$$

surface-volume ratio

important for the characterization of porous materials



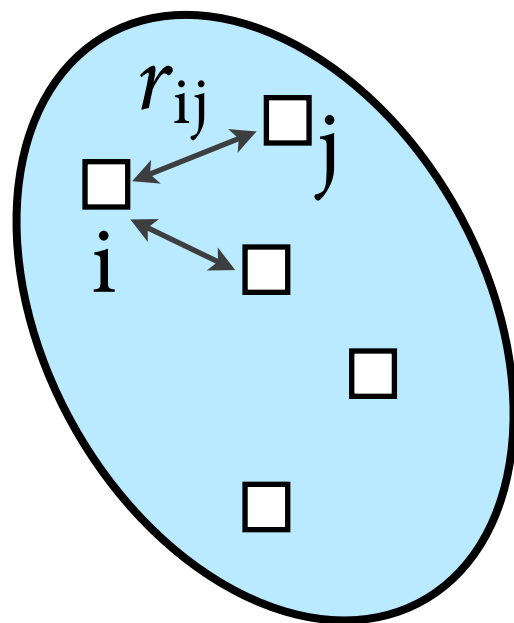
# Intensity for random particle system

Scattering intensity: 
$$I(q) = 4\pi \int_0^\infty \gamma_0(r) \frac{\sin(qr)}{qr} r^2 dr$$

Pair distance distribution function :PDDF  $p(r) = r^2 \gamma_0(r)$

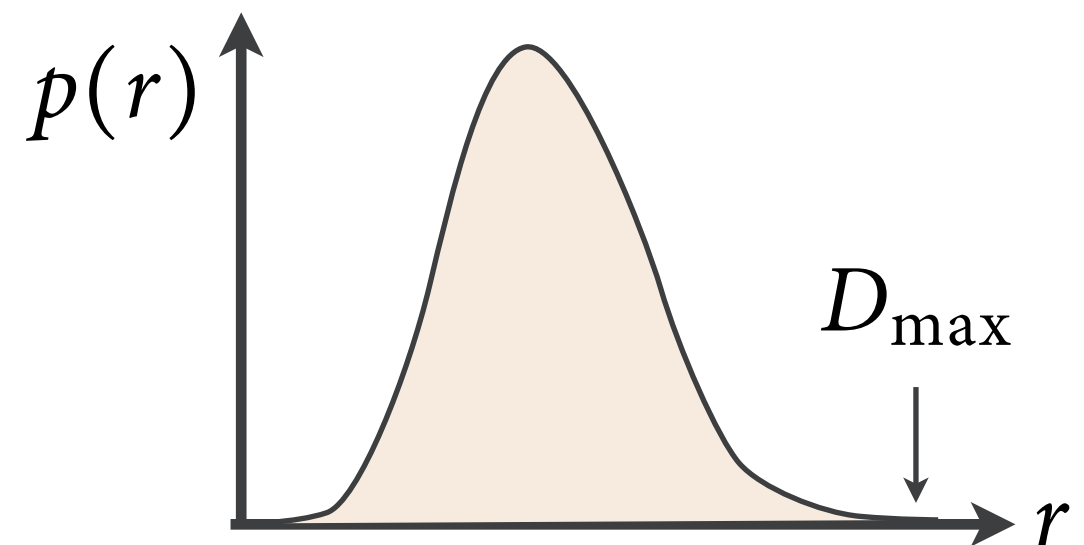
the set of distances joining the volume elements within a particle, including the case of non-uniform density distribution.

Particle's **SHAPE** and maximum **DIMENSION**.



pairs of volume elements i-j

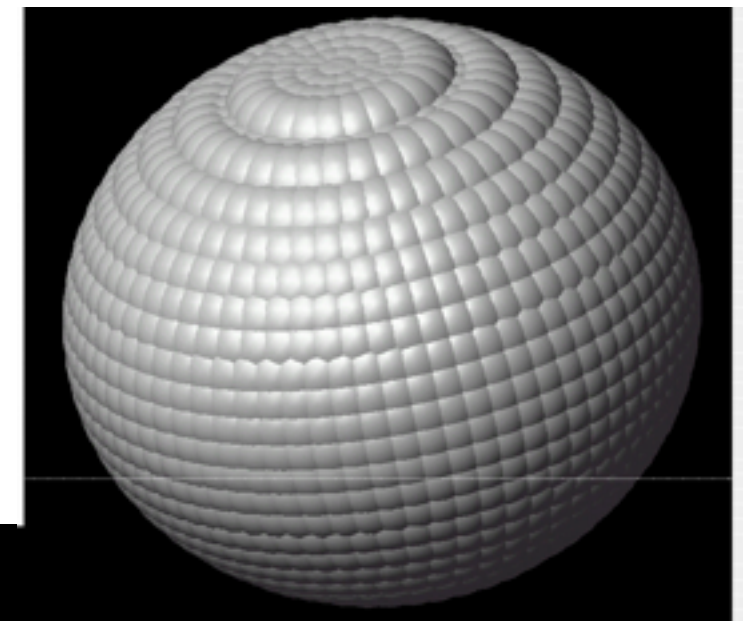
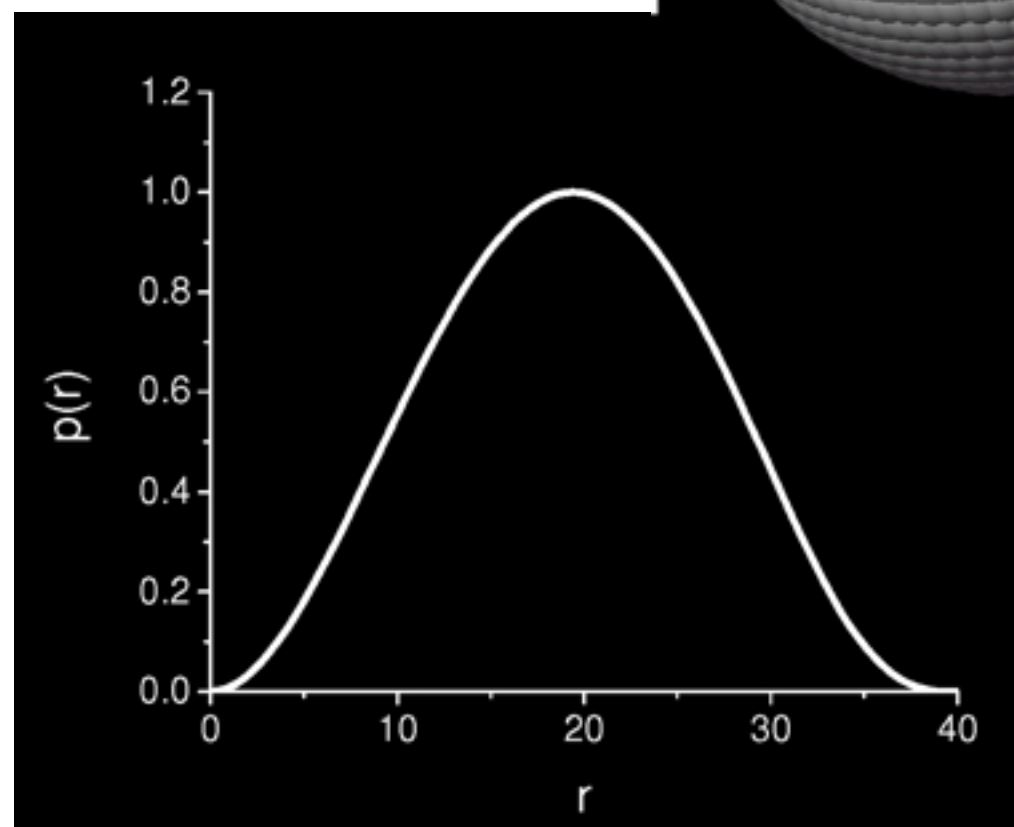
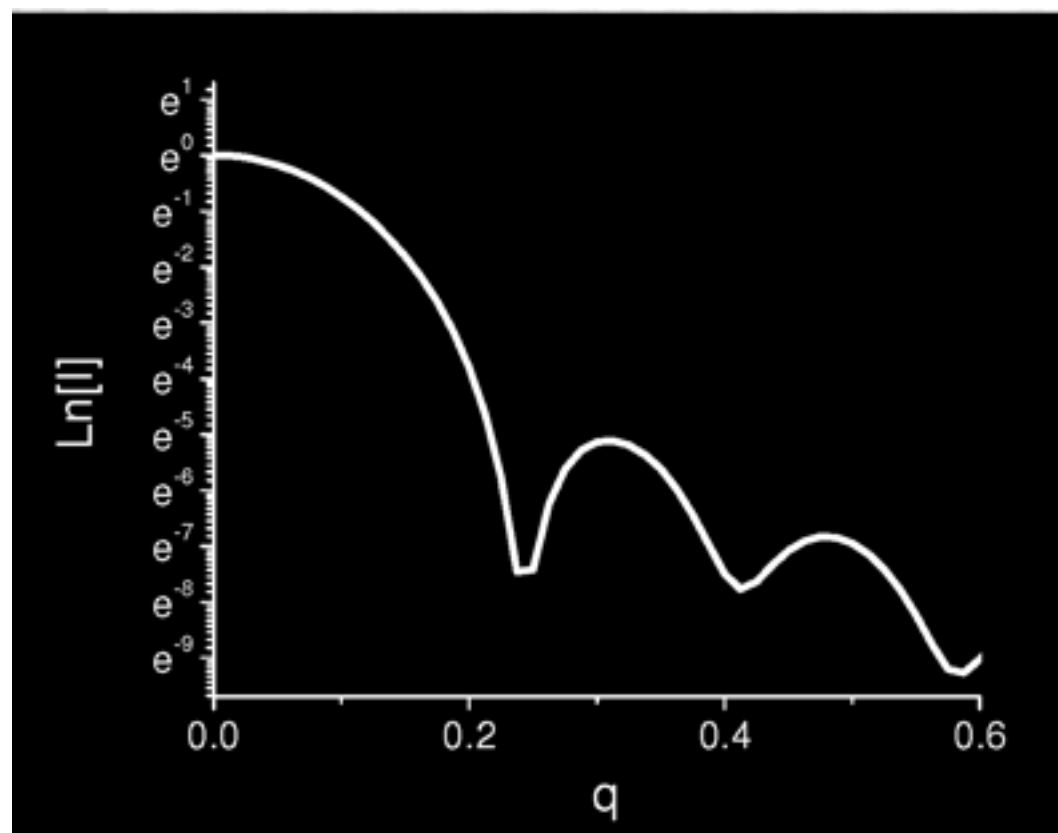
histogram o all intra-particle distances



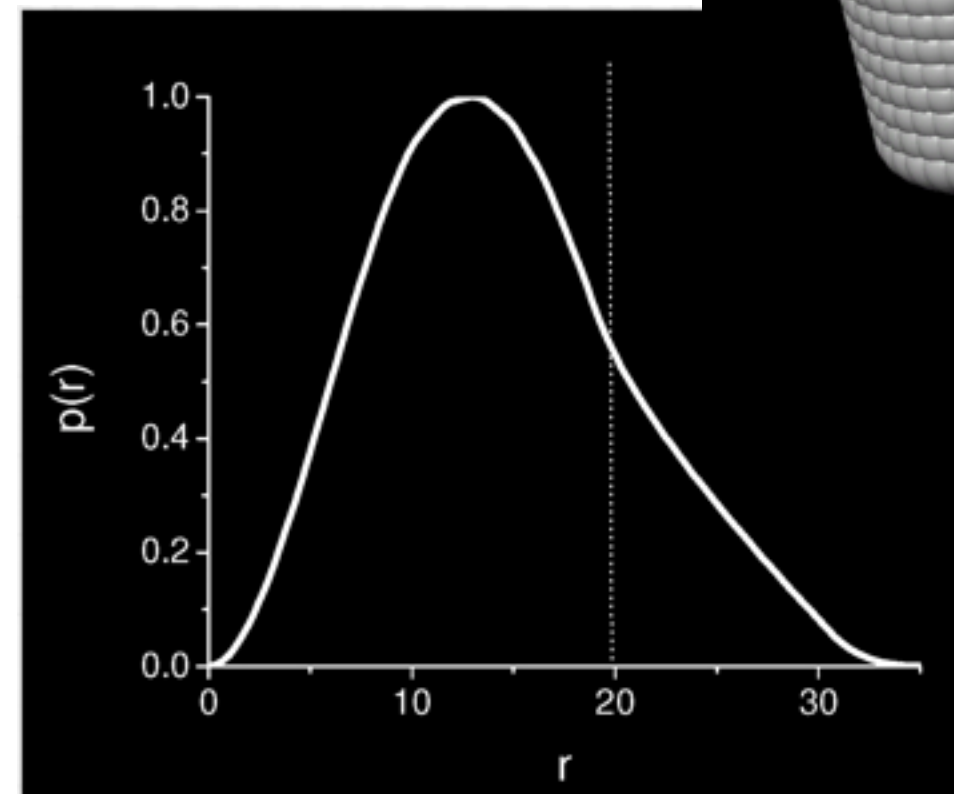
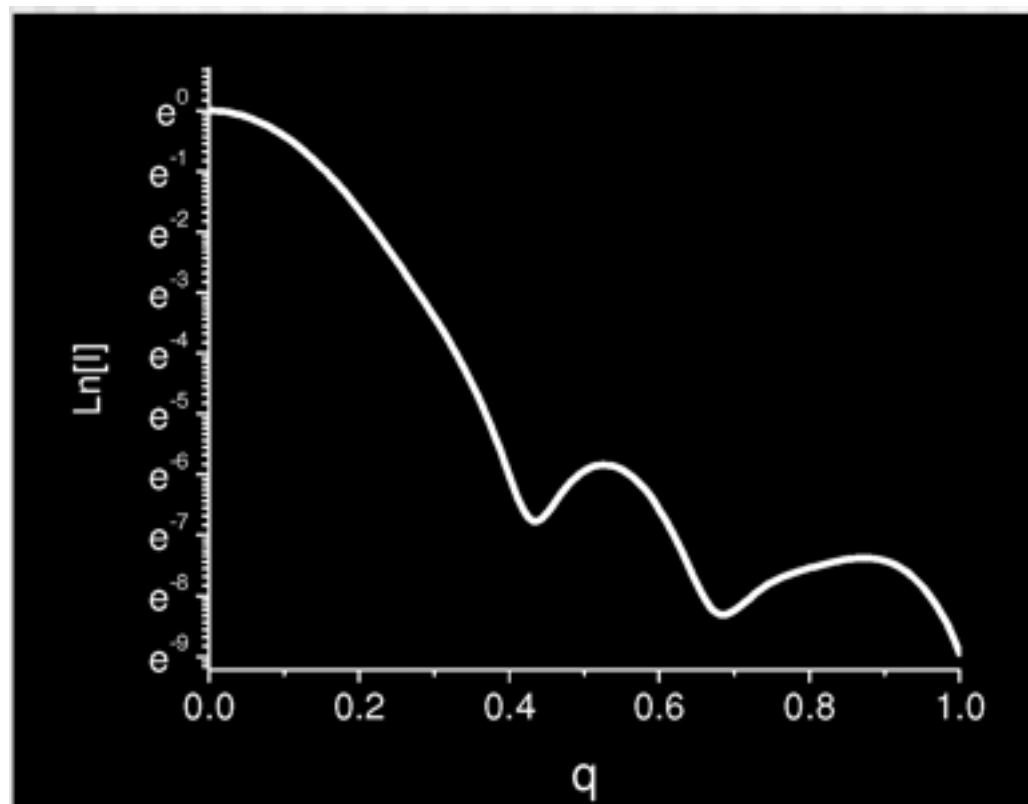
$$I(q) = 4\pi \int_0^\infty p(r) \frac{\sin(qr)}{qr} dr$$



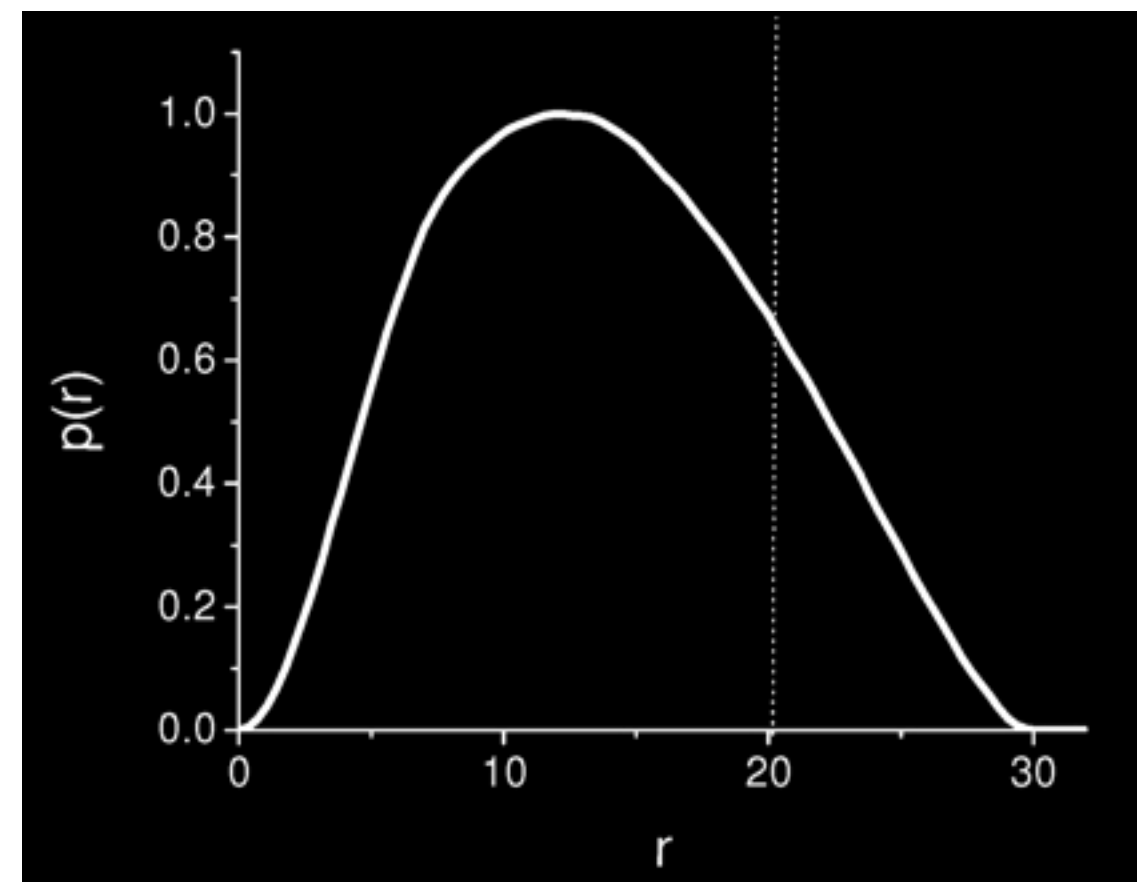
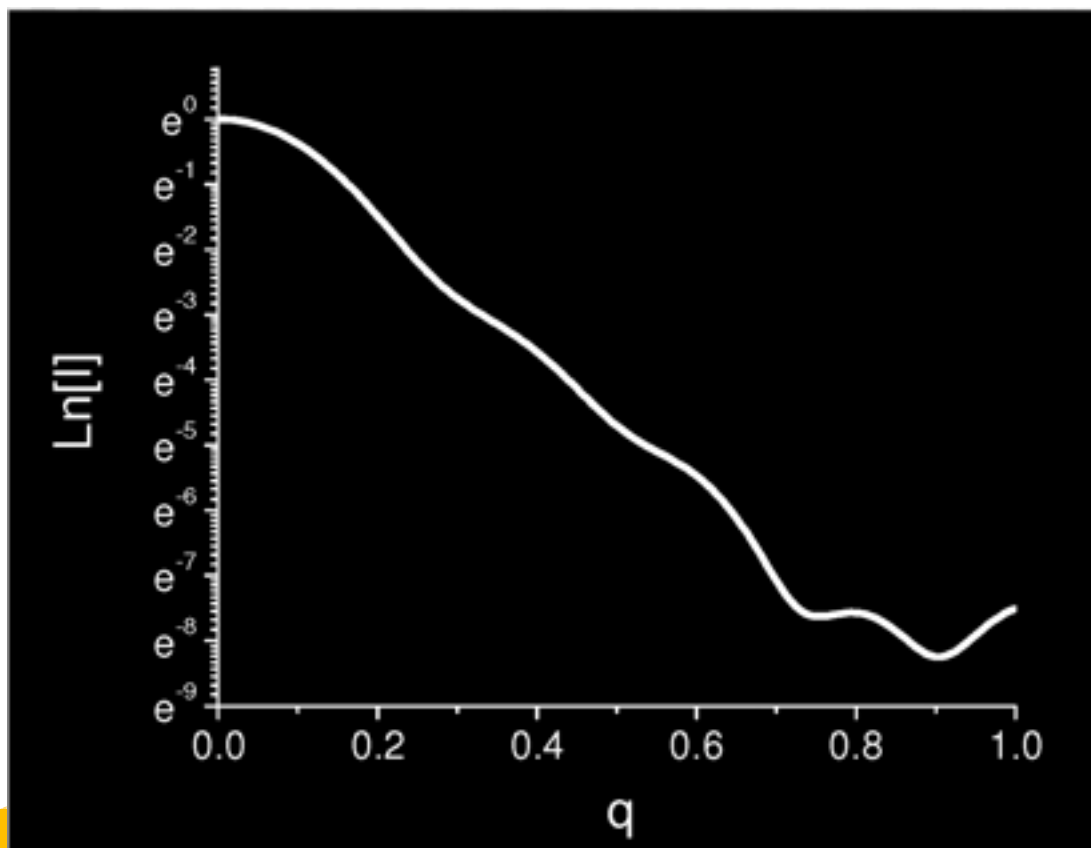
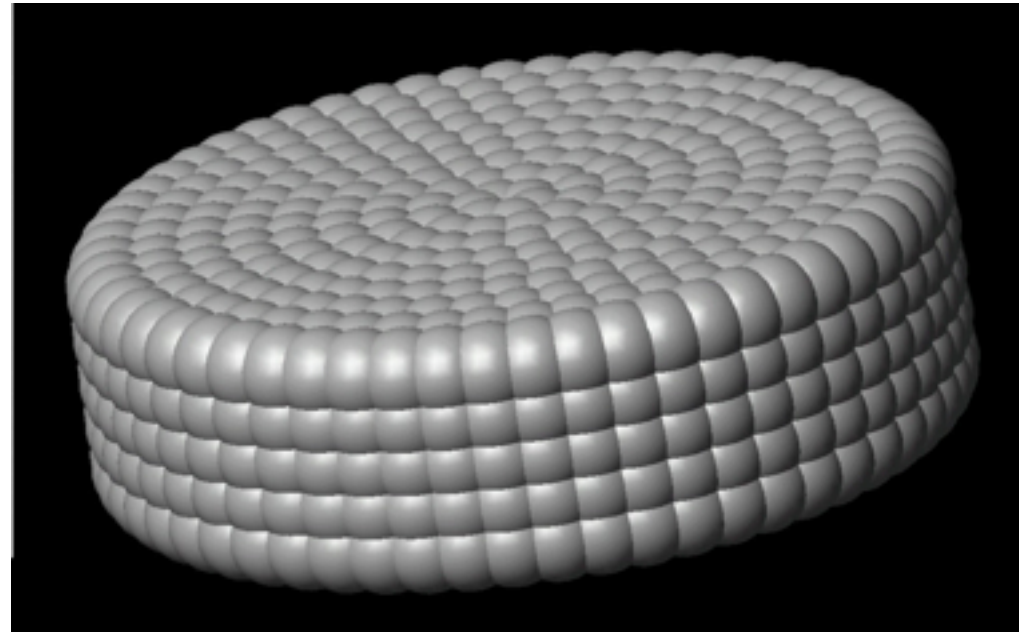
# Spherical particle



# Cylindrical particle

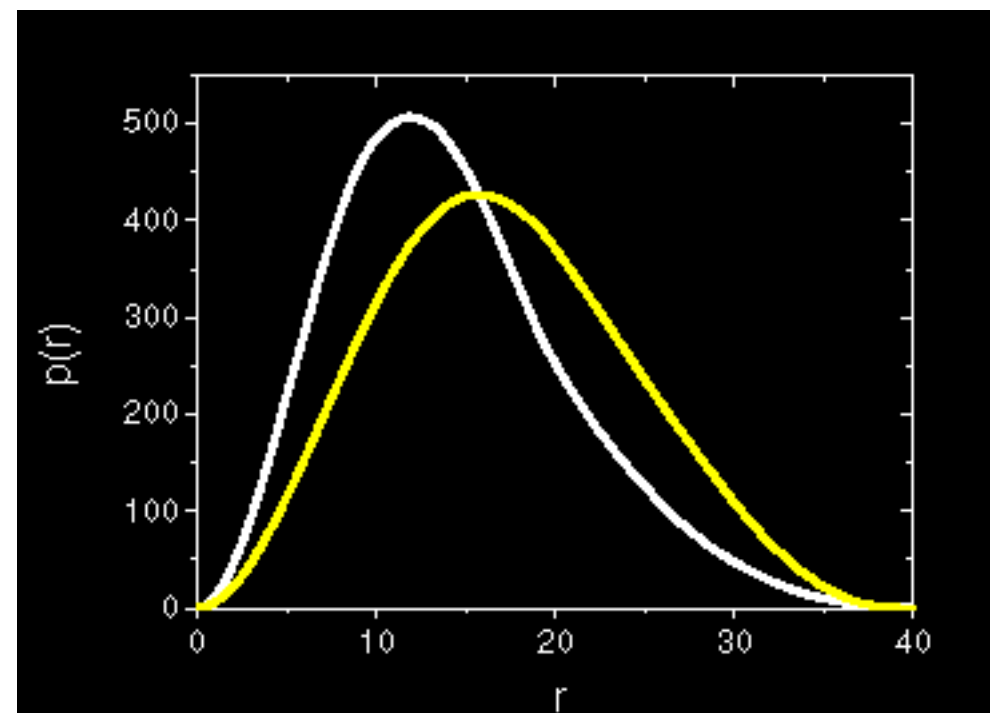
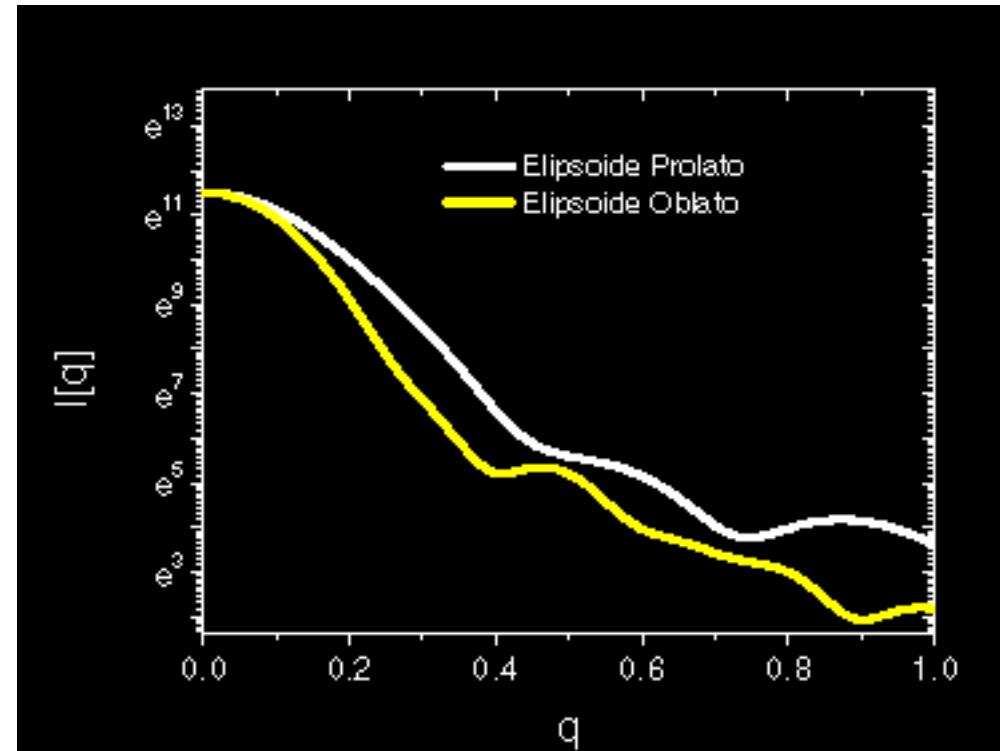
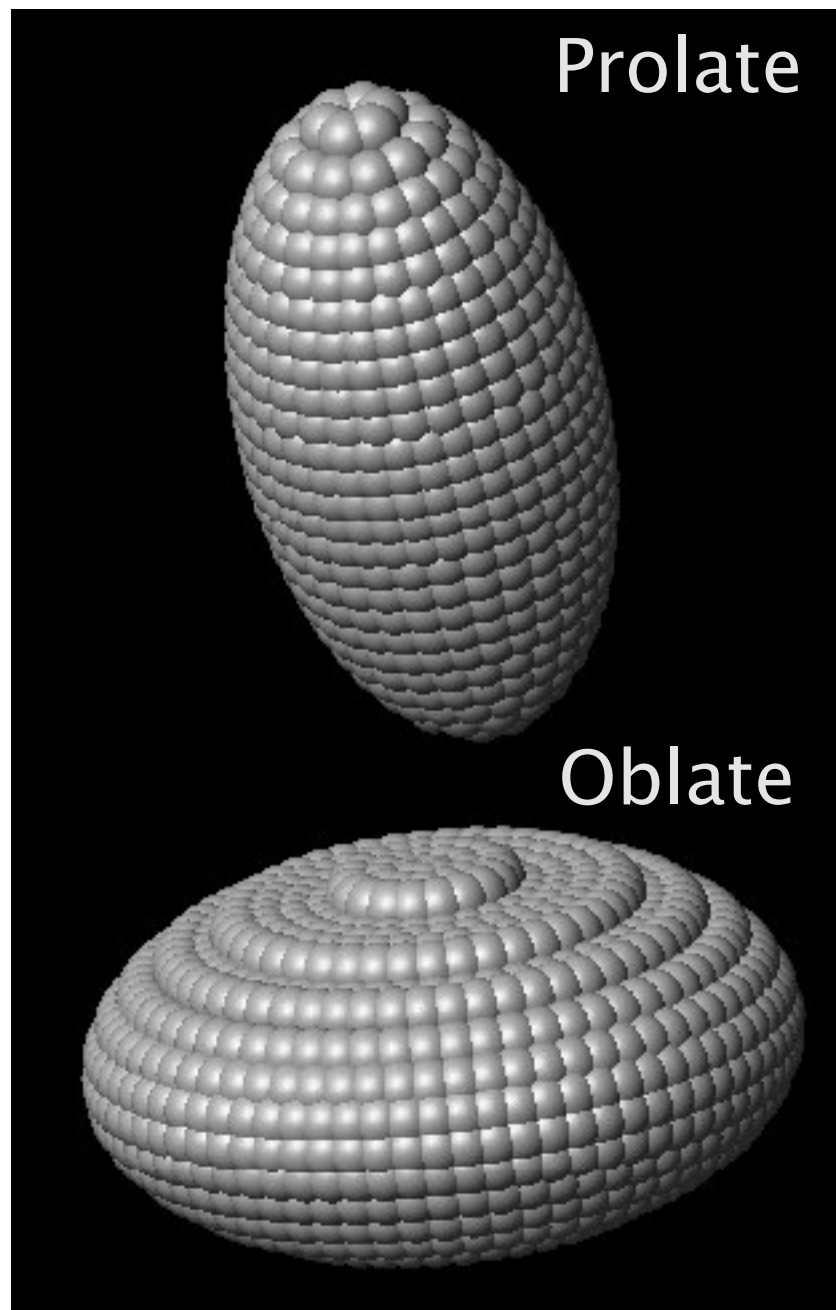


# Flat particle

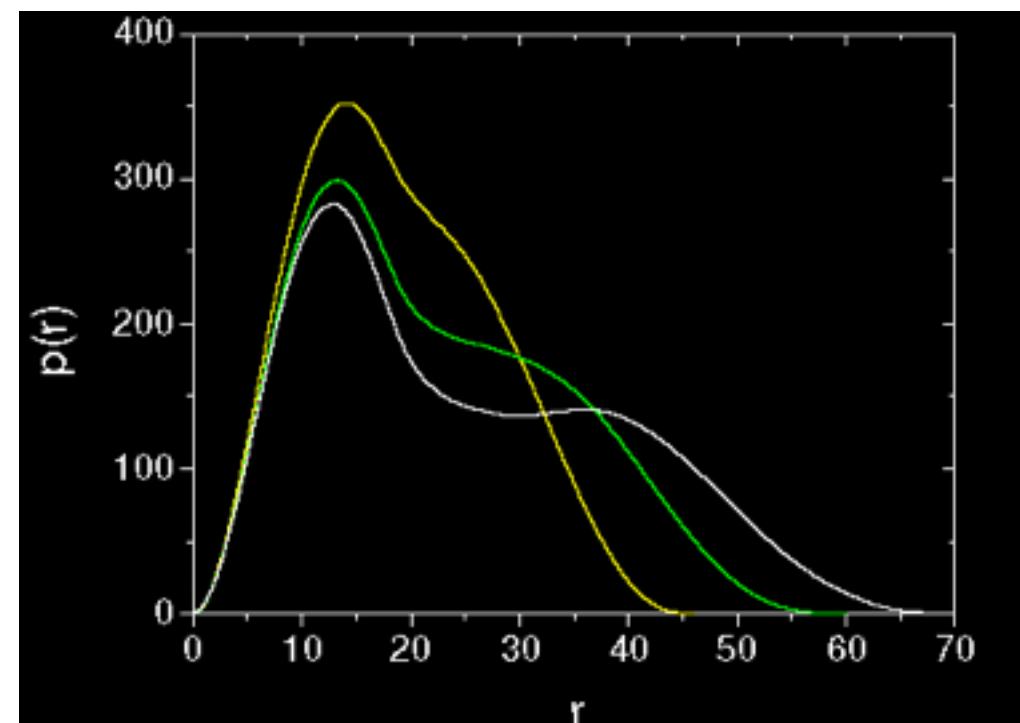
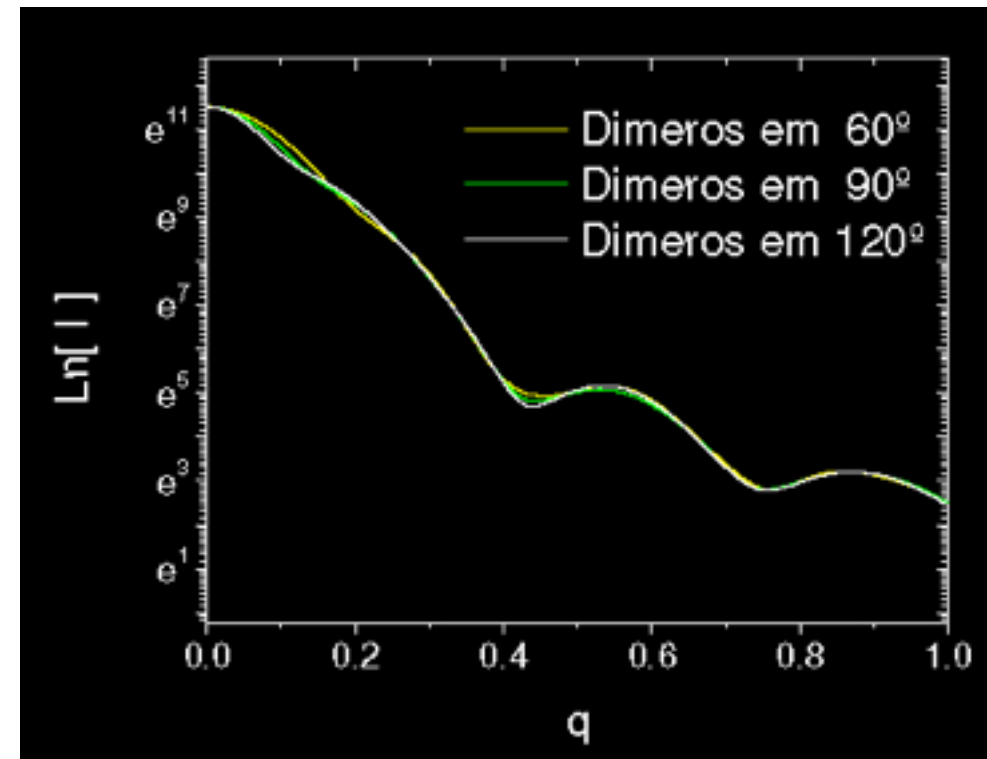
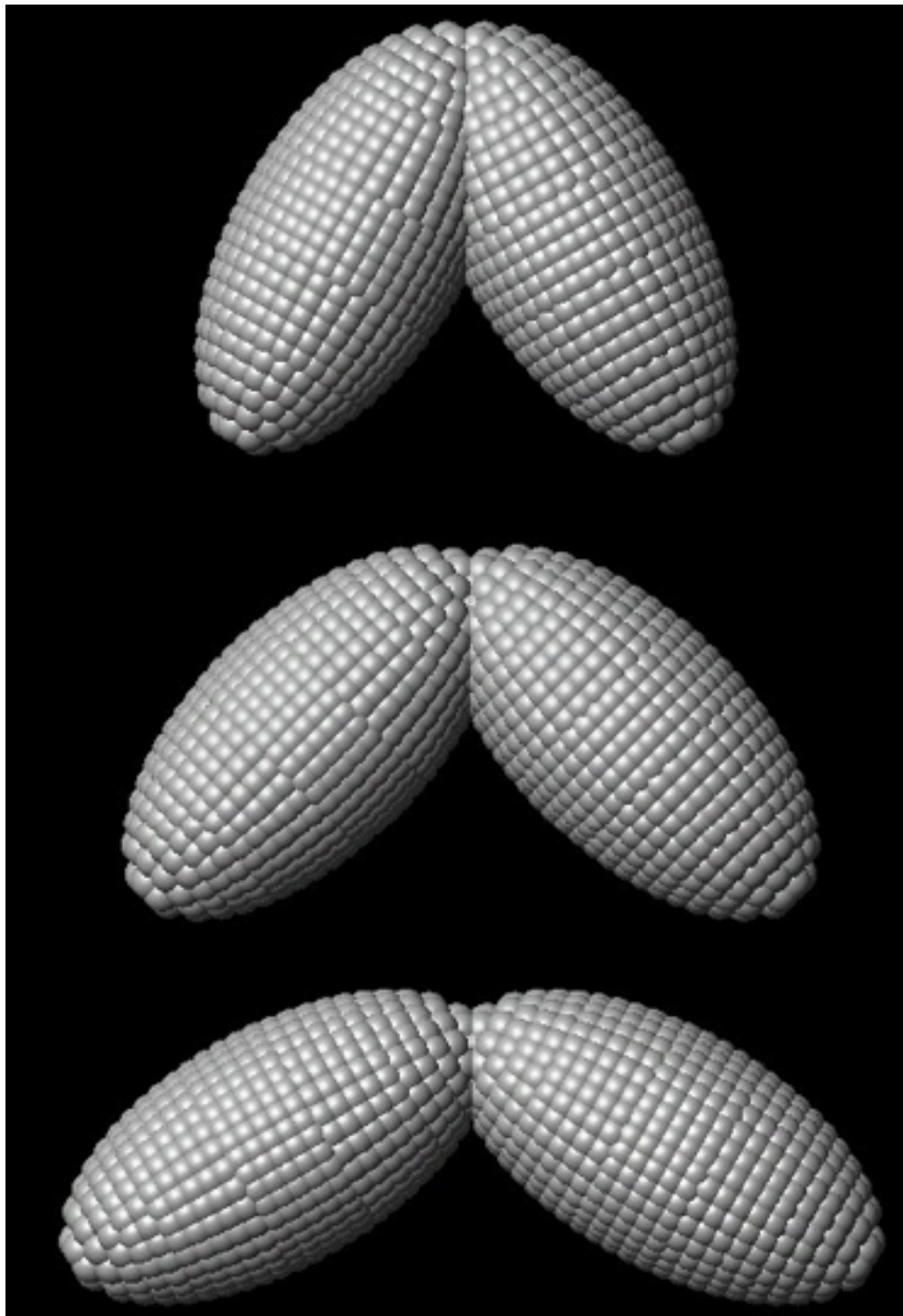




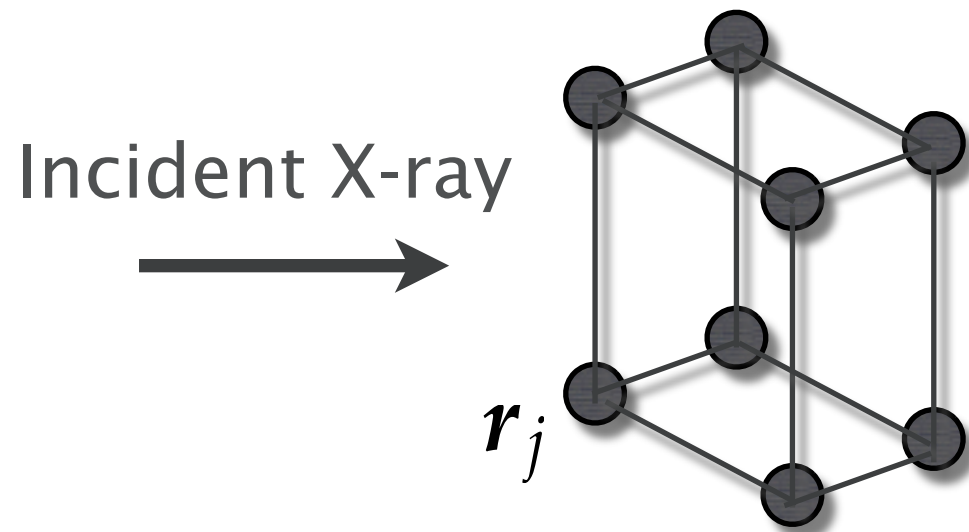
# Ellipsoids



# Two ellipsoid = dimer



# Diffraction from Periodic Structure



Diffraction from Unit cell (Crystalline structure factor)

$$F(\mathbf{q}) = \sum_j f(\mathbf{q}) \exp(-i\mathbf{q} \cdot \mathbf{r}_j)$$

$f(\mathbf{q})$  : Atomic Form Factor

Diffraction  
Intensity:

$$I(\mathbf{q}) \sim \left| \underline{G(\mathbf{q})} \right|^2 \left| F(\mathbf{q}) \right|^2$$

Laue function:  $\left| G(\mathbf{q}) \right|^2 = \frac{\sin^2(\pi N \mathbf{q} \cdot \mathbf{r})}{\sin^2(\pi \mathbf{q} \cdot \mathbf{r})}$

- Maximum  $\sim N^2$
- FWHM  $\sim 2\pi/N$ 
  - FWHM  $\rightarrow$  Size of crystal

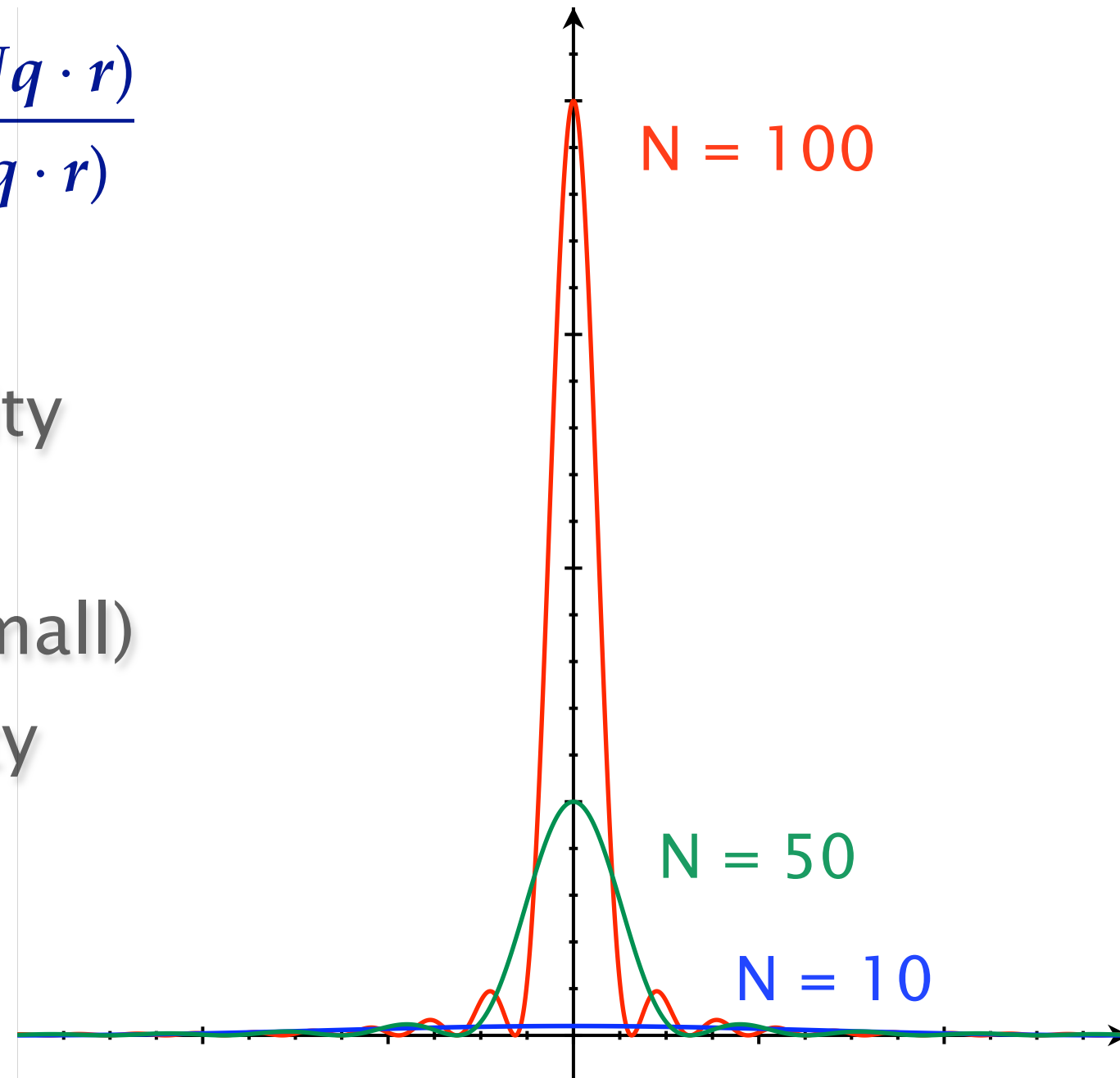


# Laue Function

Laue function:  $|G(\mathbf{q})|^2 = \frac{\sin^2(\pi N \mathbf{q} \cdot \mathbf{r})}{\sin^2(\pi \mathbf{q} \cdot \mathbf{r})}$

- Large crystal
  - High diffraction intensity
  - Narrow FWHM
- Soft matter (crystal size: small)
  - Low diffraction intensity
  - Wide FWHM

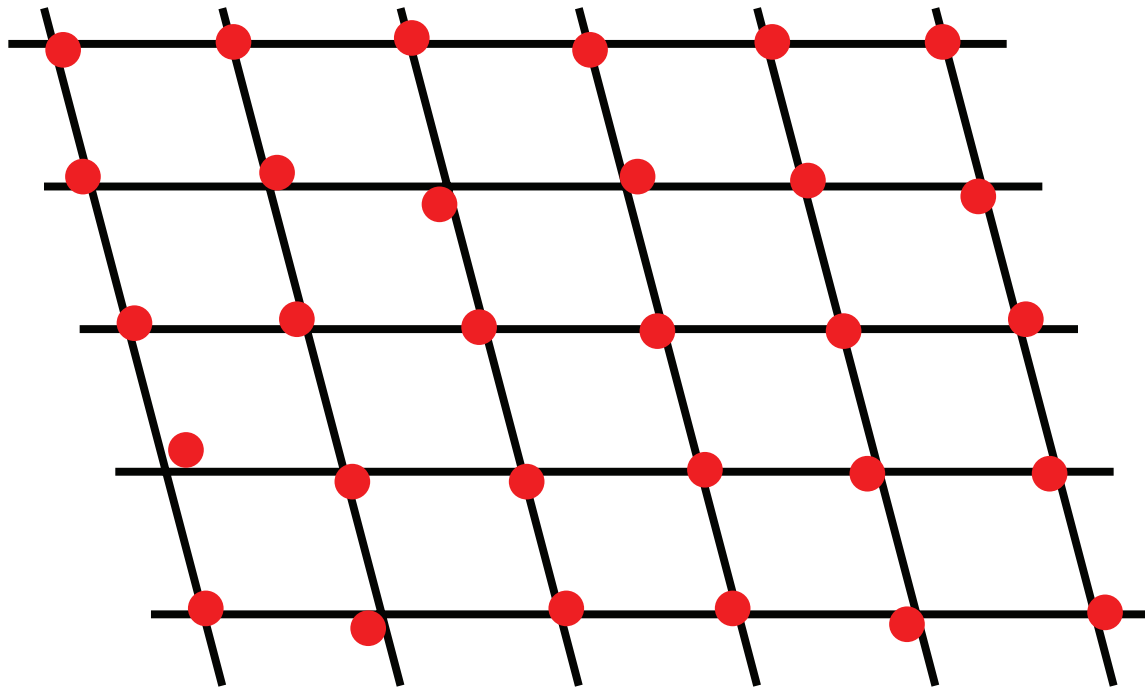
→ low S/N



Crystal size --> Intensity & FWHM of diffraction

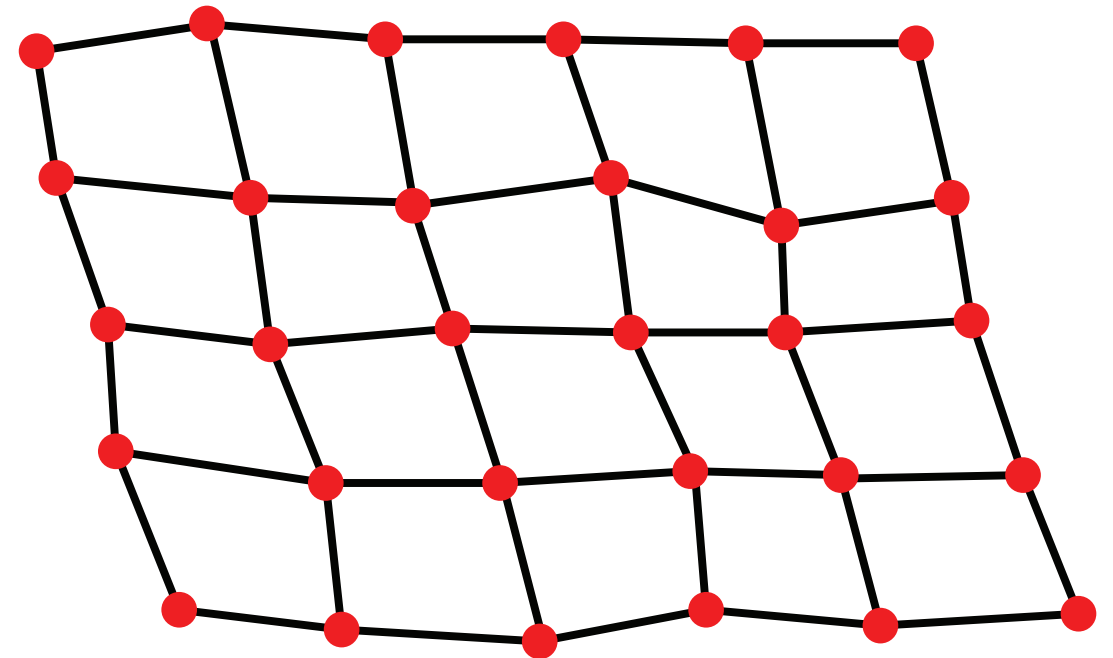
# Imperfection of crystal (2D)

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**Imperfection of 1st kind**

Thermal fluctuation etc.



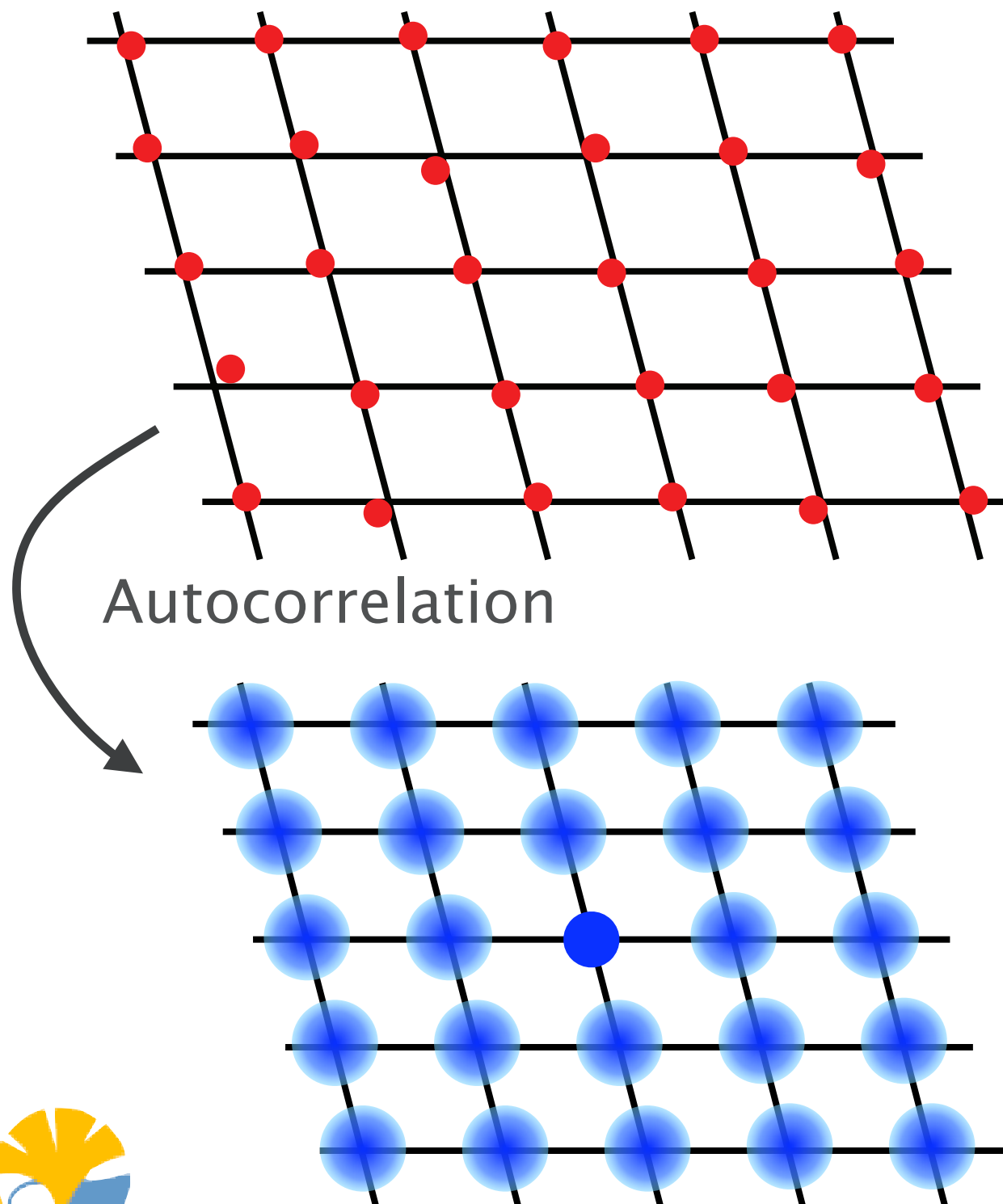
**Imperfection of 2nd kind**

in the case of soft matter

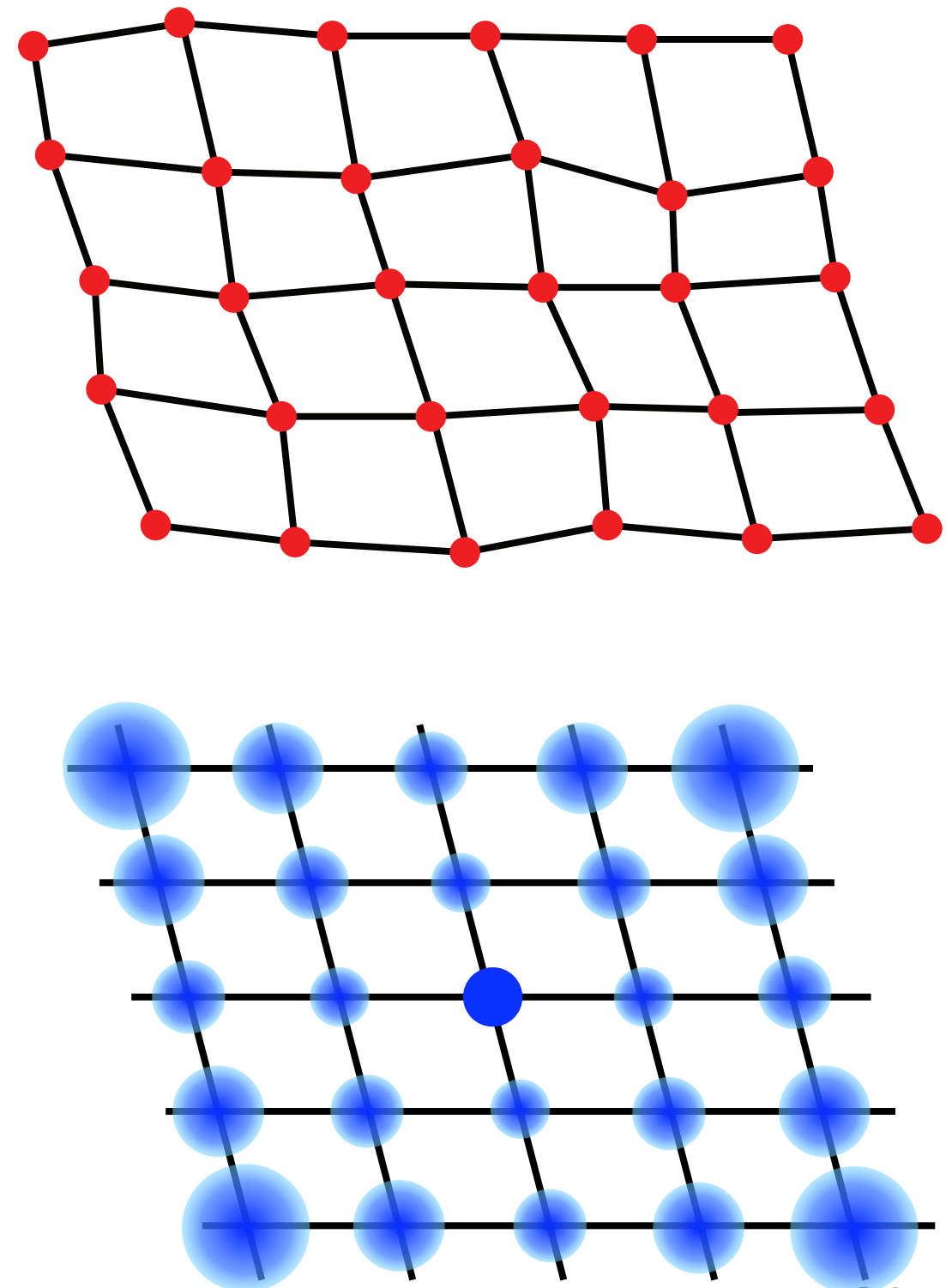


# Imperfection of crystal

## Imperfection of 1st kind



## Imperfection of 2nd kind







# Imperfection of lattice (1D)

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Perfect lattice 

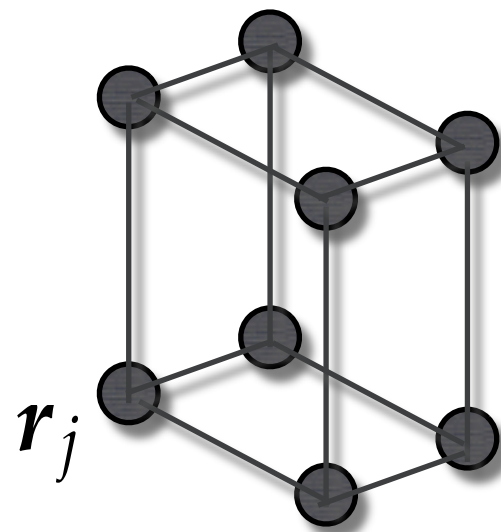
Imperfection of 1st kind 

Imperfection of 2nd kind 

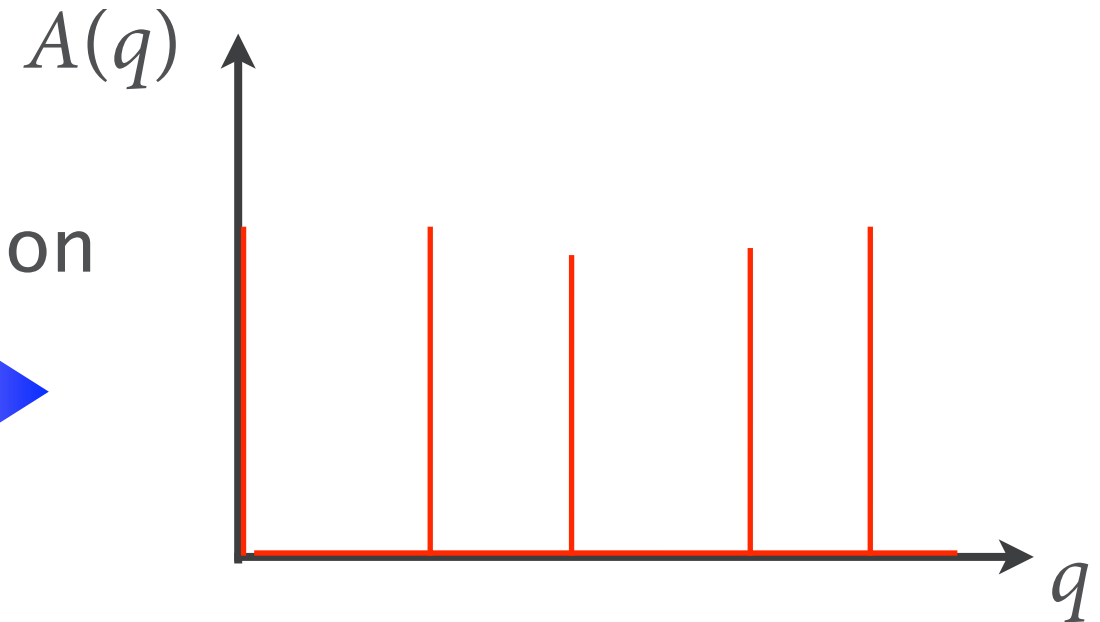
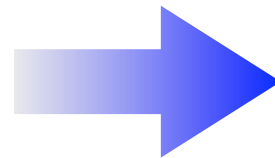
☞ Effect of imperfections on diffraction ?



# Diffraction from lattice-structure



Diffraction



$$\rho(\mathbf{r}) = \rho_u(\mathbf{r}) * \underline{z(\mathbf{r})}$$

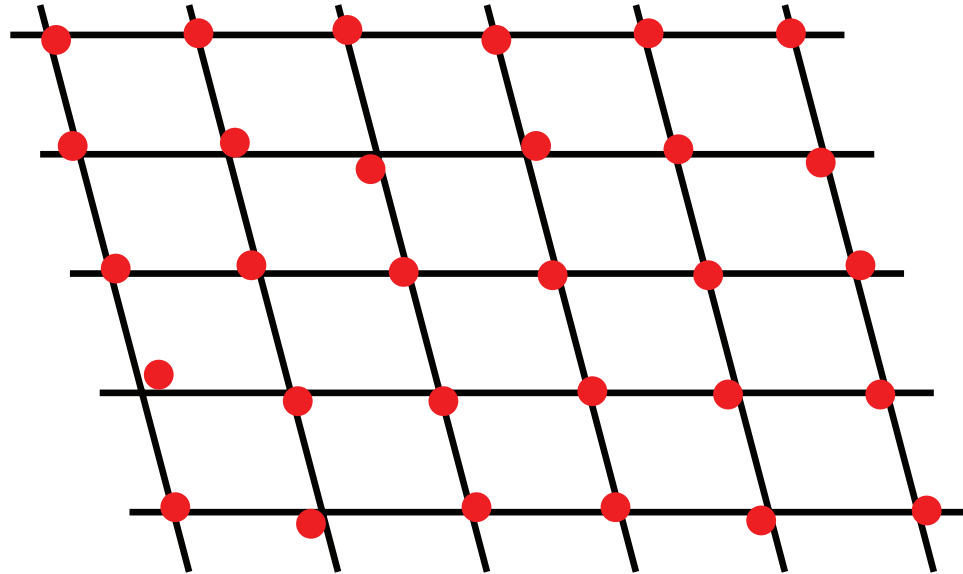
$$A(\mathbf{q}) = F(\mathbf{q}) \cdot \underline{Z(\mathbf{q})}$$

Form of lattice

$z(\mathbf{r})$  with imperfection ---> calculate  $Z(\mathbf{q})$



# Imperfection of 1st kind



$p(\mathbf{r})$  : distribution function

Fourier trans.  $\longrightarrow P(\mathbf{q})$

Diffraction with imperfection:  $|Z(\mathbf{q})|^2 = N \left[ 1 - \underbrace{|P(\mathbf{q})|^2} \right] + \underbrace{|P(\mathbf{q})|^2}_{\text{ideal lattice}} \underbrace{Z_0(\mathbf{q})}_{\text{ideal lattice}}$

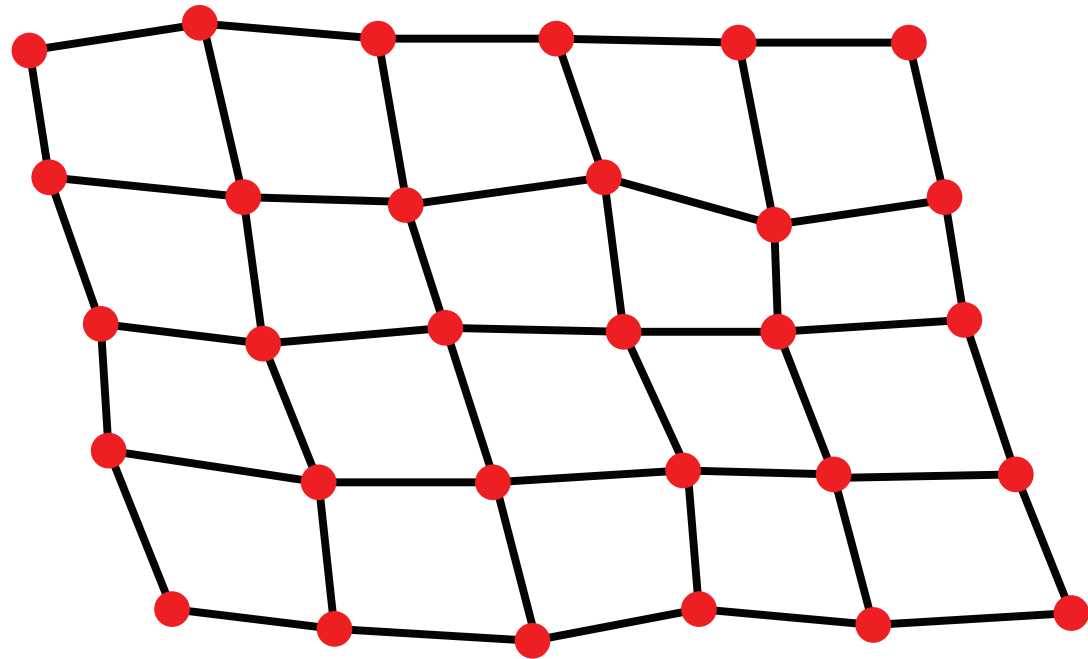
Thermal fluctuation ( $p(\mathbf{r})$ : Gaussian)

Debye-Waller factor:  $\exp\left(-\frac{1}{3}\sigma^2 q^2\right)$

- decrease diffraction intensity (no effect on FWHM)
- background at larger angle diffraction

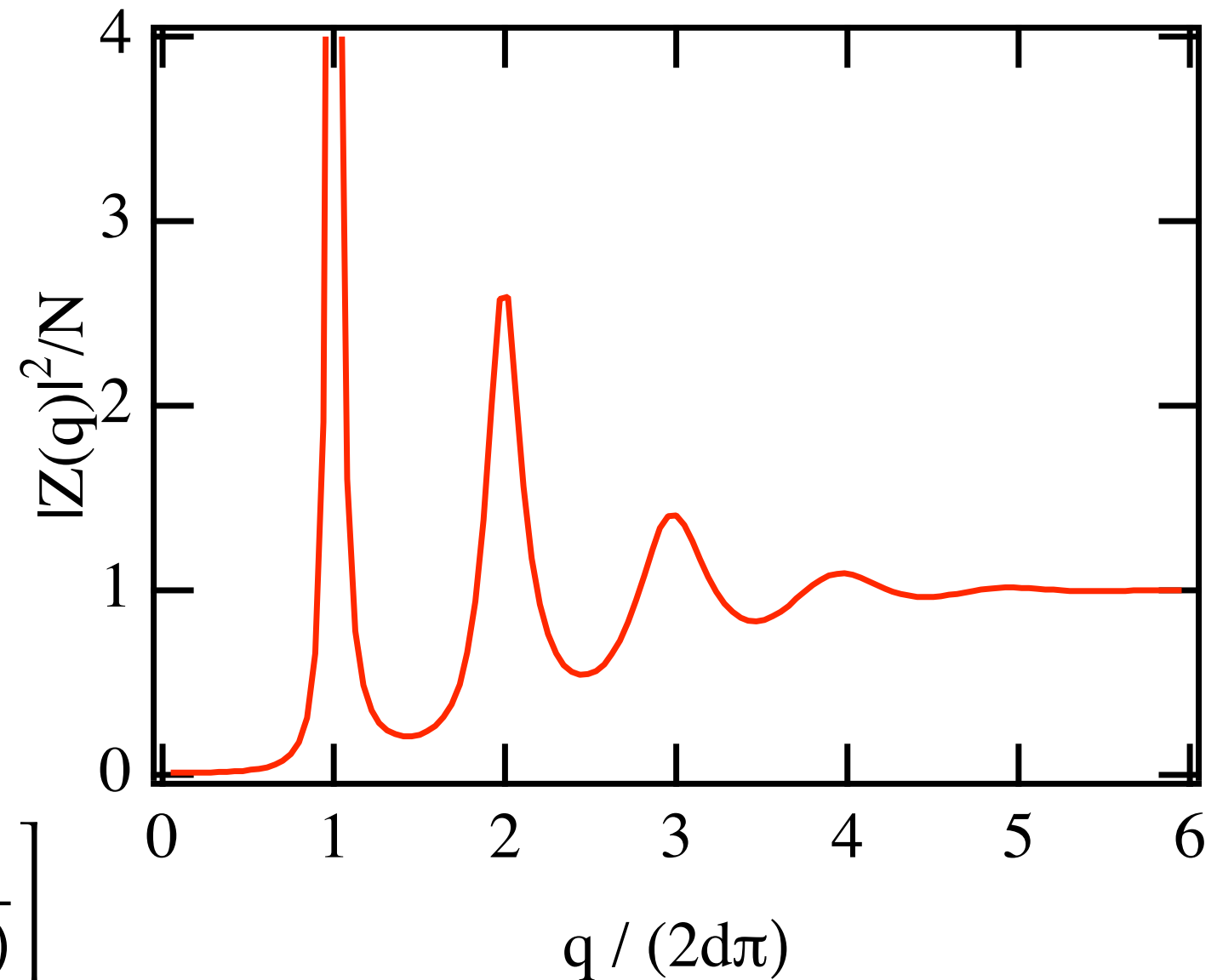


# Imperfection of 2nd kind



Paracrystal theory

$$|Z(q)|^2 = N \left[ 1 + \frac{P(q)}{1 - P(q)} + \frac{P^*(q)}{1 - P^*(q)} \right]$$



Decrease of diffraction intensity and  
Increase of FWHM



R. Hosemann, S. N. Bagchi, *Direct Analysis of Diffraction by Matter*, North-Holland, Amsterdam (1962).

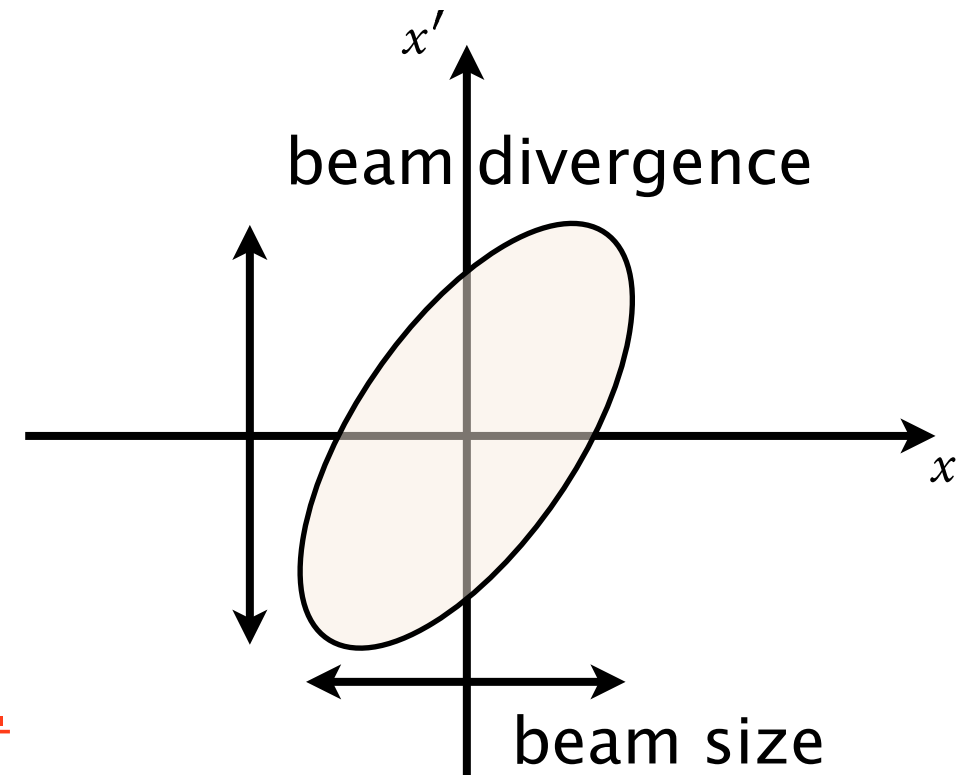
# X-ray Source for SAXS

Brilliance -- Product of size and divergence of beam

$$\text{Brilliance} = \frac{d^4 N}{dt \cdot d\Omega \cdot dS \cdot d\lambda/\lambda}$$

[photons/(s · mrad<sup>2</sup> · mm<sup>2</sup> · 0.1% rel.bandwidth)]

Brilliance is preserved (Liouville's theorem).



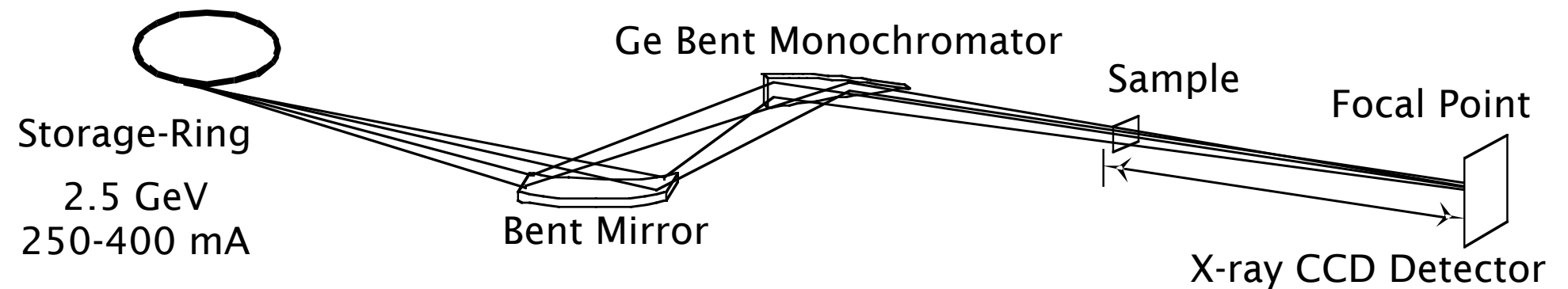
SAXS with a low divergence and small beam

→ High brilliance beam is required !

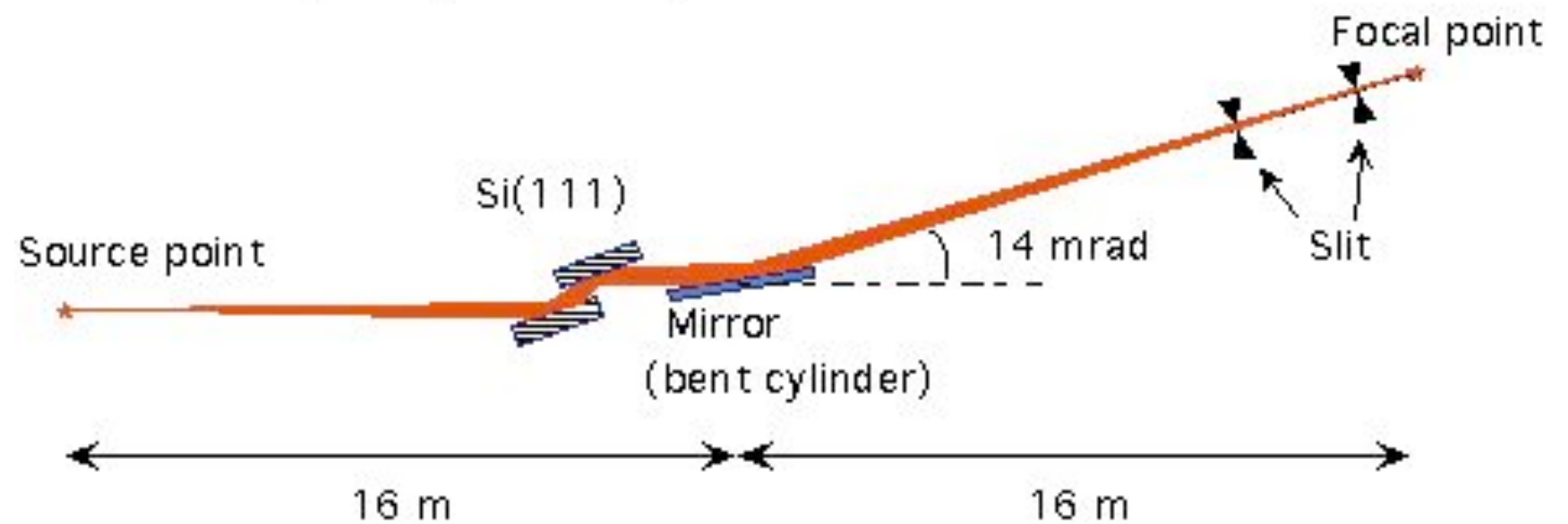


# SAXS Optics

PF BL-15A

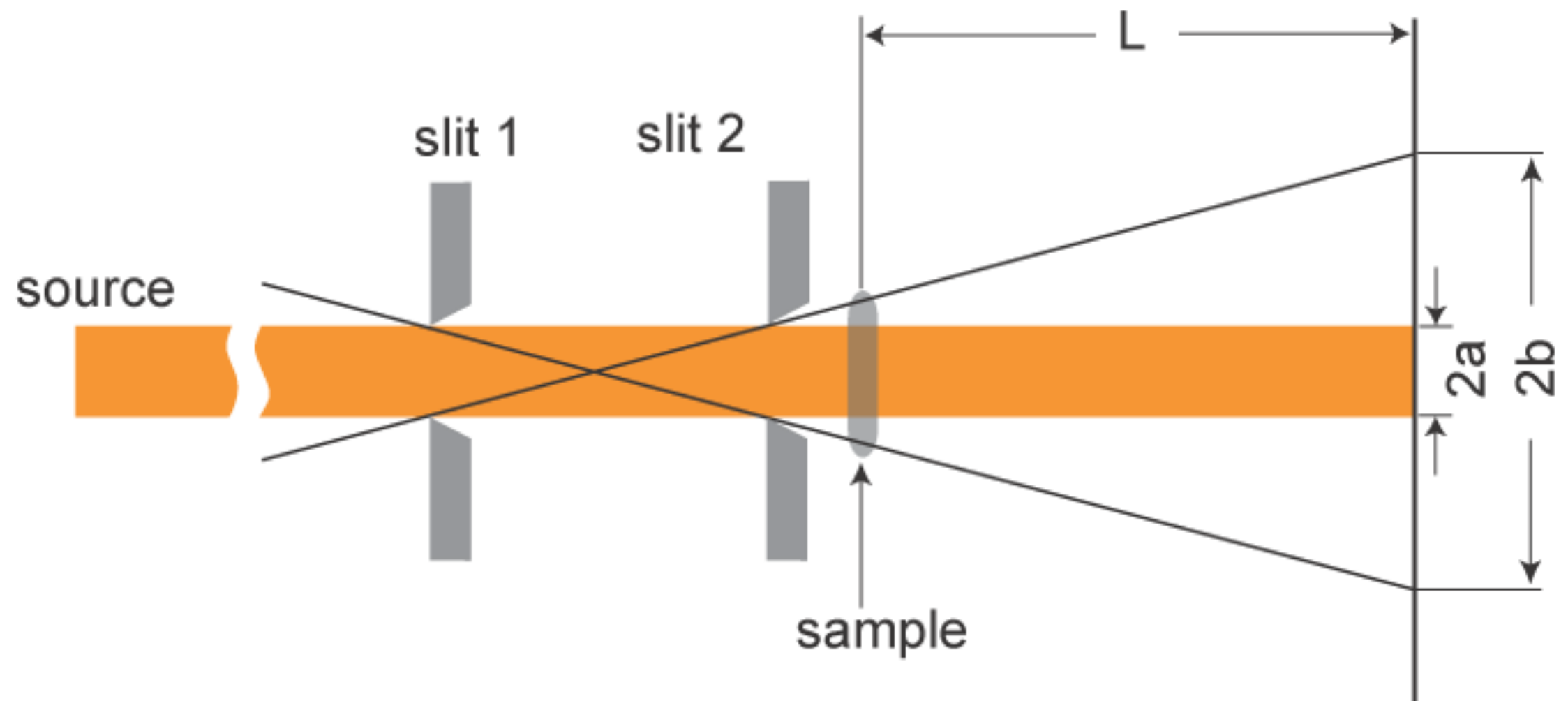


PF BL-10C





# SAXS slits

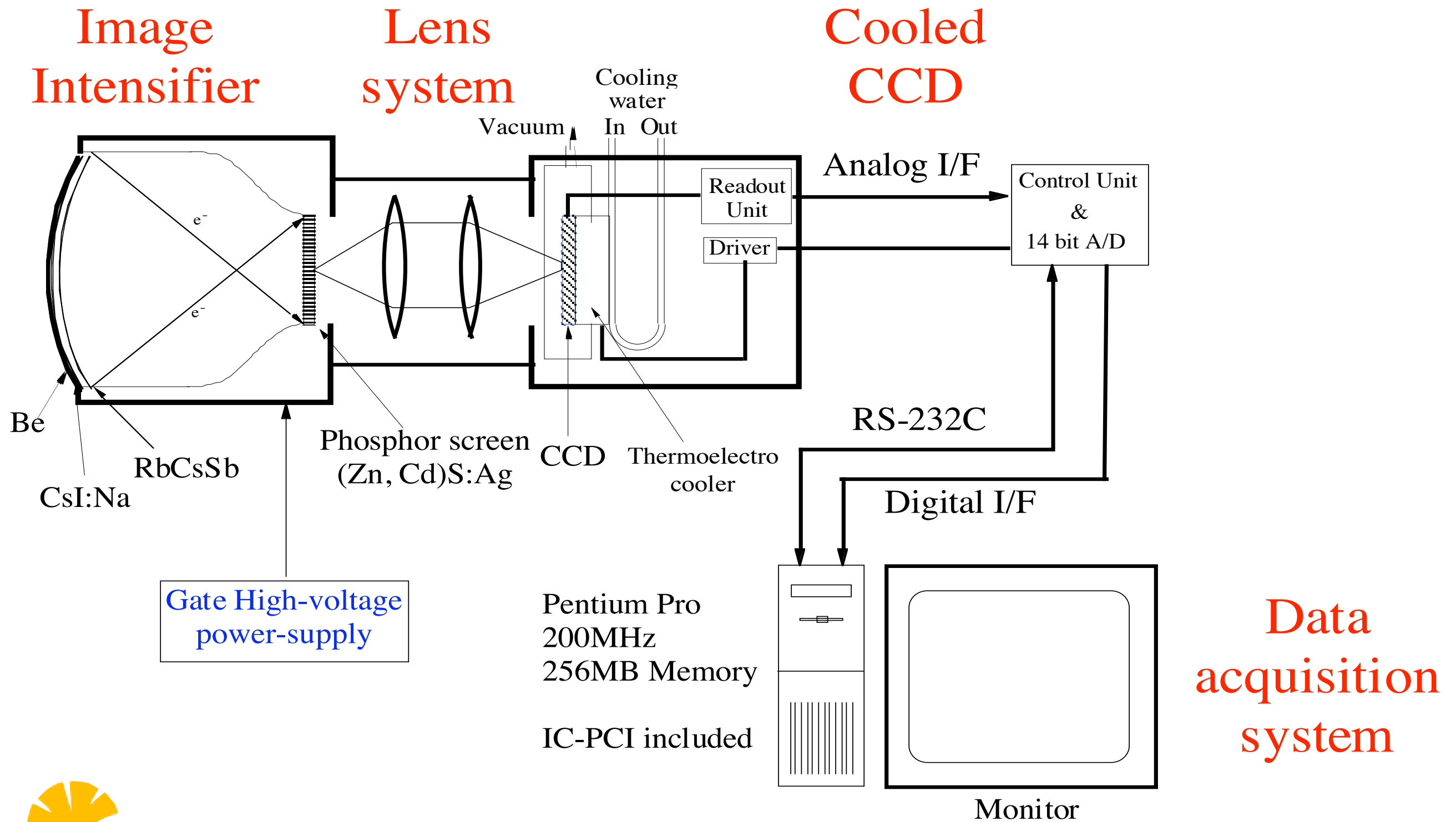


# Detectors for SAXS

	Good Point	Drawback
PSPC	<ul style="list-style-type: none"><li>• time-resolved</li><li>• photon-counting</li><li>• low noise</li></ul>	<ul style="list-style-type: none"><li>• counting-rate limitation</li></ul>
Imaging Plate	<ul style="list-style-type: none"><li>• wide dynamic range</li><li>• large active area</li></ul>	<ul style="list-style-type: none"><li>• slow read-out</li></ul>
CCD with Image Intensifier	<ul style="list-style-type: none"><li>• time-resolved</li><li>• high sensitivity</li></ul>	<ul style="list-style-type: none"><li>• image distortion</li><li>• low dynamic range</li></ul>
Fiber-tapered CCD	<ul style="list-style-type: none"><li>• fast read-out</li><li>• automated measurement</li></ul>	<ul style="list-style-type: none"><li>• not good for time-resolved</li></ul>



# X-ray CCD detector with Image Intensifier



# Advanced SAXS

## Microbeam X-ray

- Inhomogeneity of nano-structure
- local time evolution of structure

## Time-resolved

- time evolution of structure

## GI-SAXS

- surface, interface, thin films

## SAXS

## XPCS

- structural fluctuation
- dynamics

## Combined measurement with DSC, viscoelasticity wide-q (USAXS-SAXS-WAXS) 2D measurement

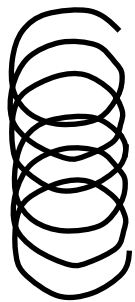
- hierarchical structure

- anisotropic structure

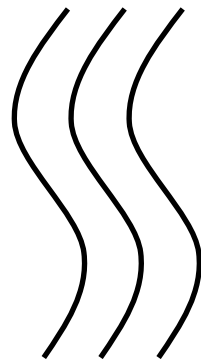


# Application of paracrystal theory

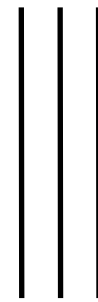
Collab. with Kao Ltd.



African



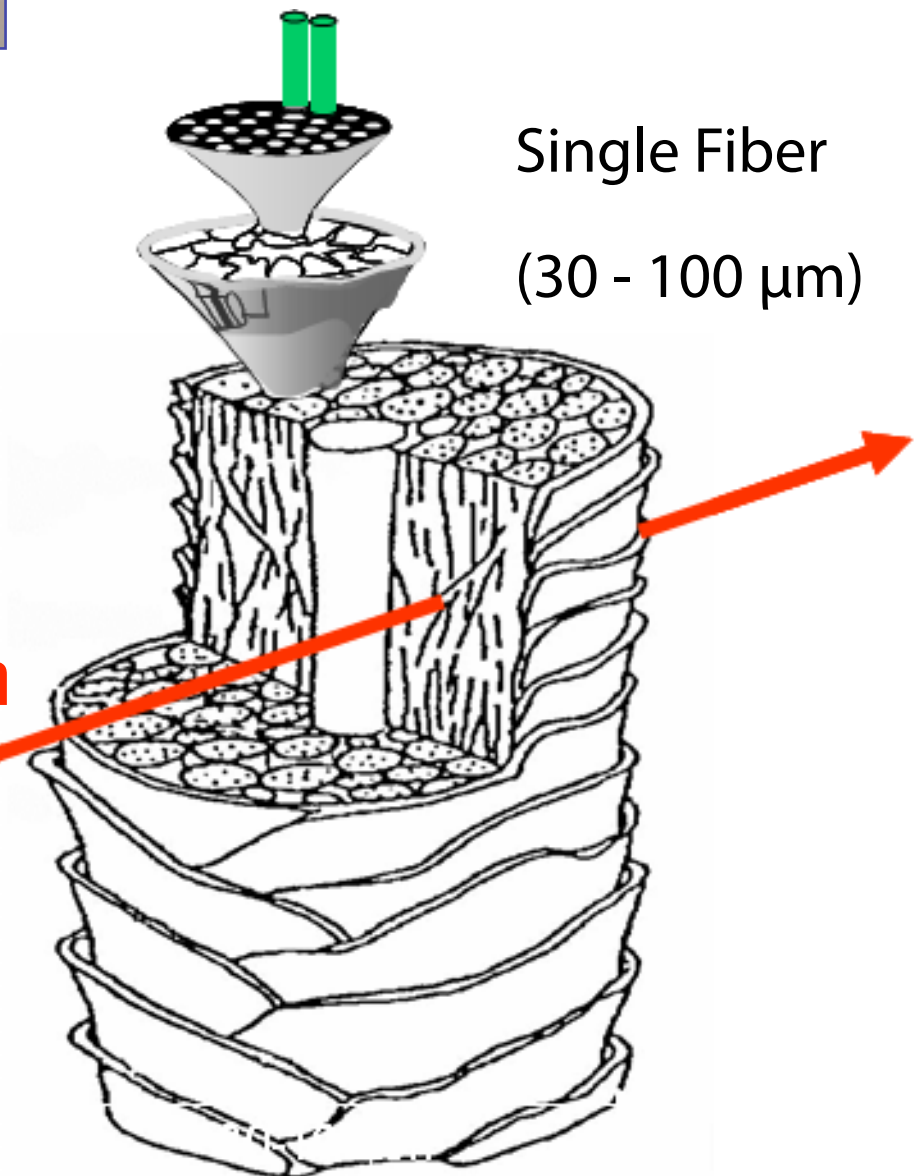
Caucasian



Asian

Relationship between macroscopic form and microscopic structure?

**X-ray Microbeam**  
(5  $\mu\text{m}$  x 5  $\mu\text{m}$ )



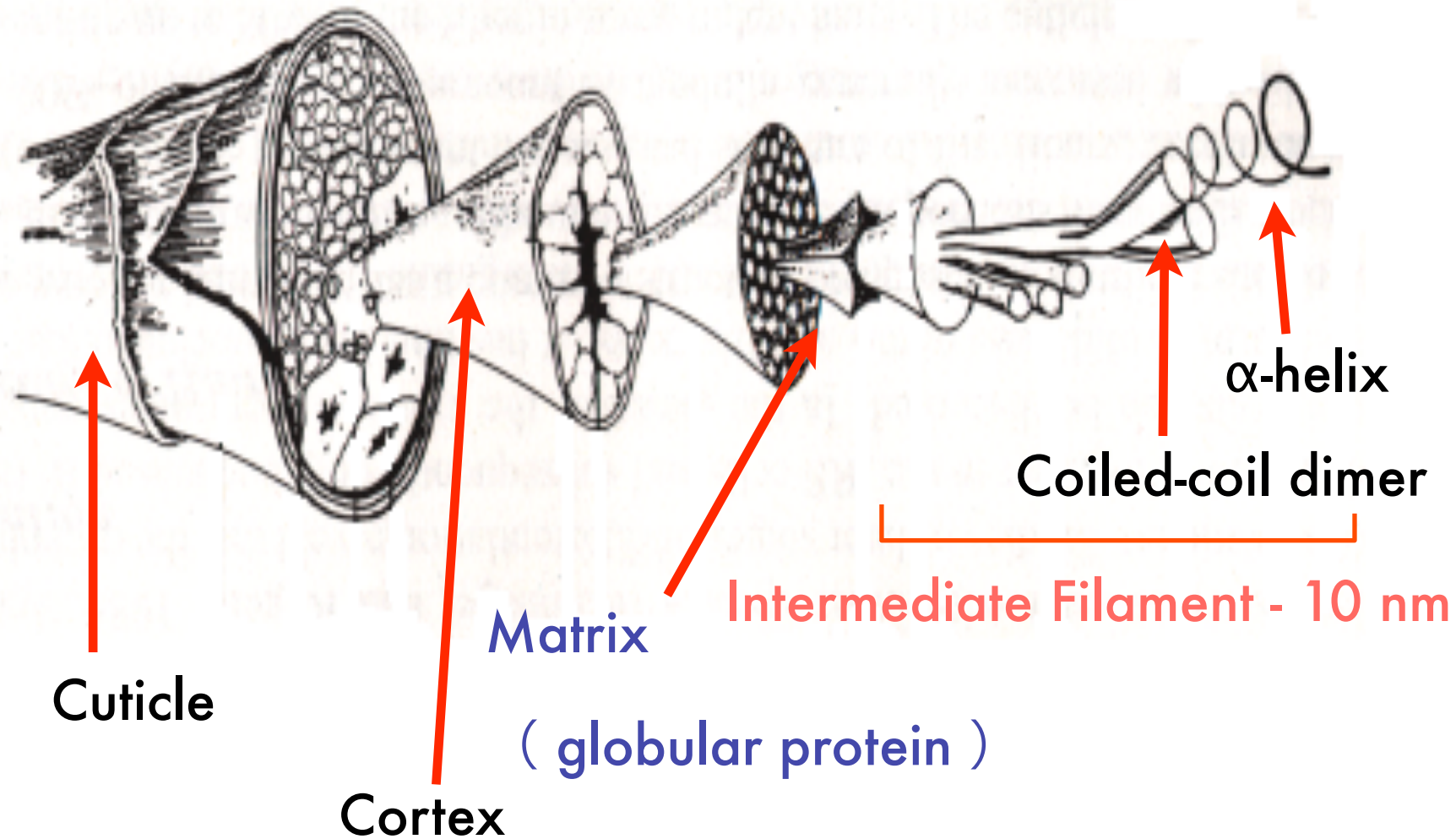
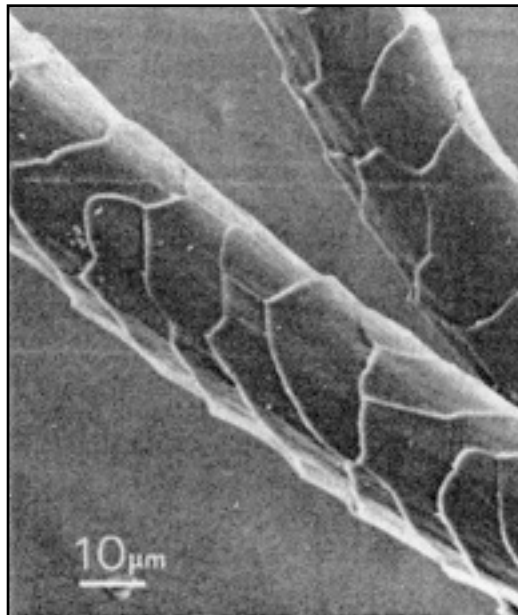
Local observation with an X-ray microbeam



# Internal structure of wool



SEM 像



R. D. B. Fraser et al., Proc. Int. Wool Text. Res. Conf., Tokyo, II, 37, (1985) partially changed.

H. Ito et al., Textile Res. J. 54, 397-402 (1986).

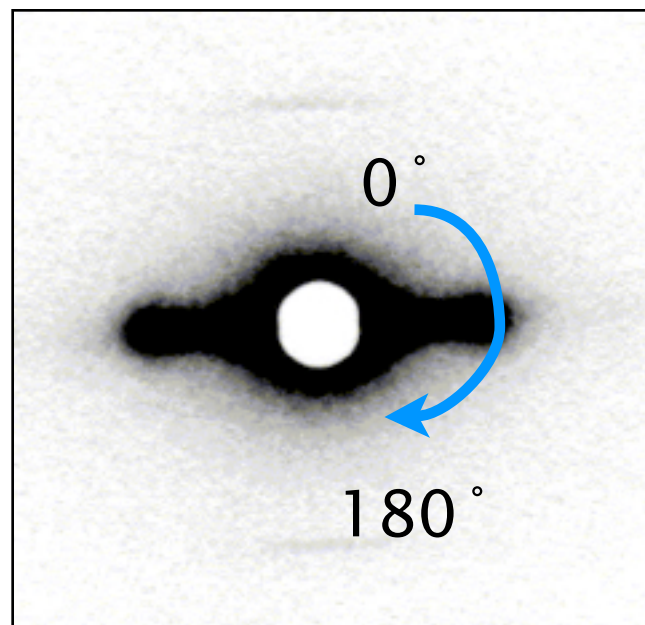
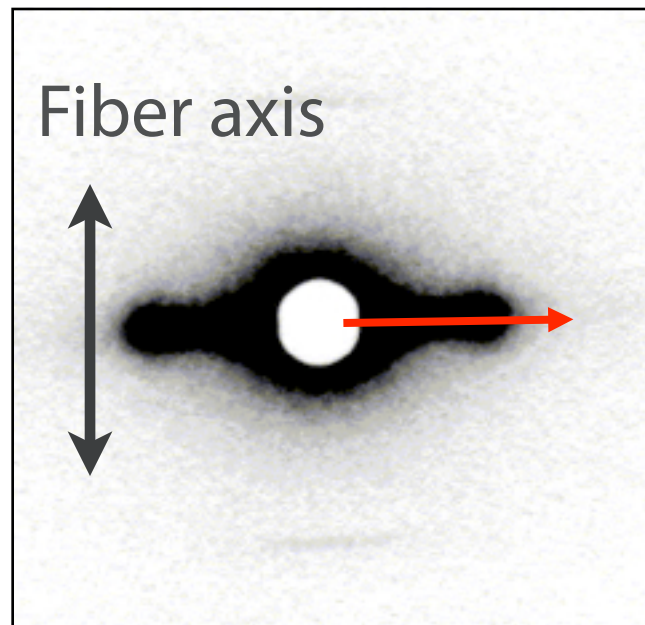
Relationship between IF distribution and hair curliness?



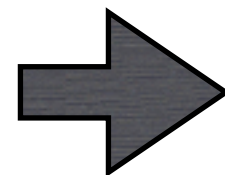
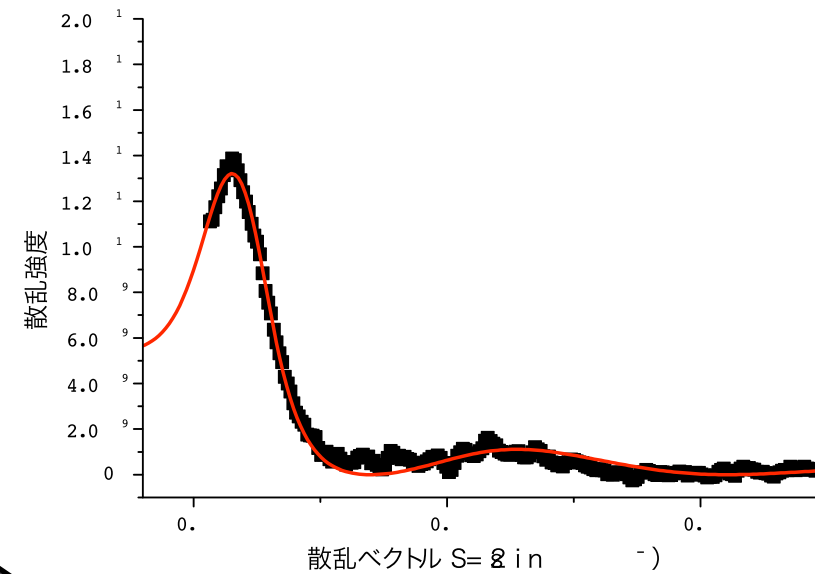


# Structure of Intermediate Filament

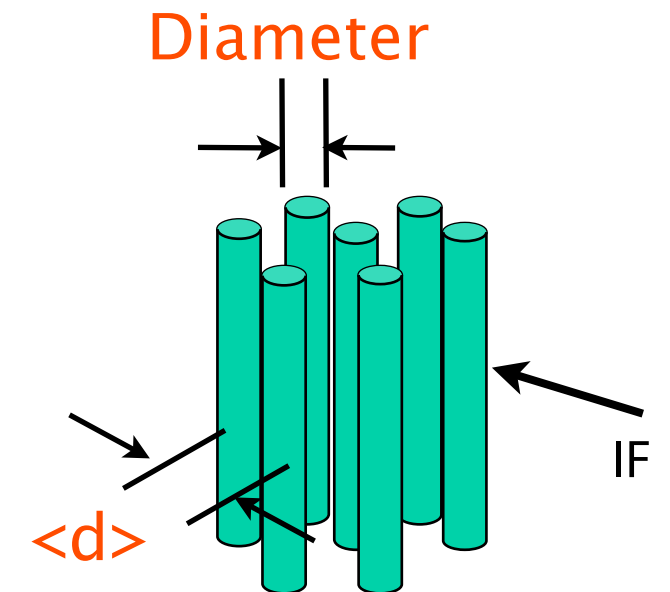
Scattering pattern



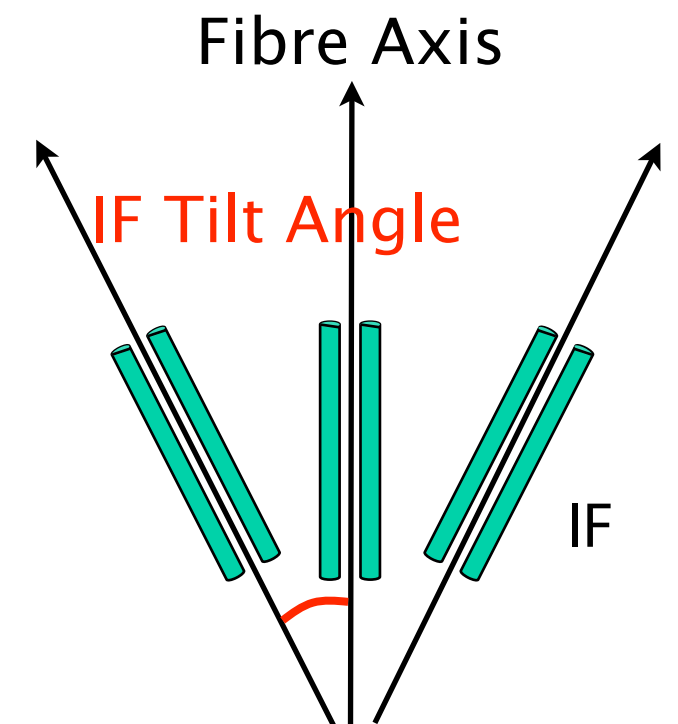
1 D intensity profile



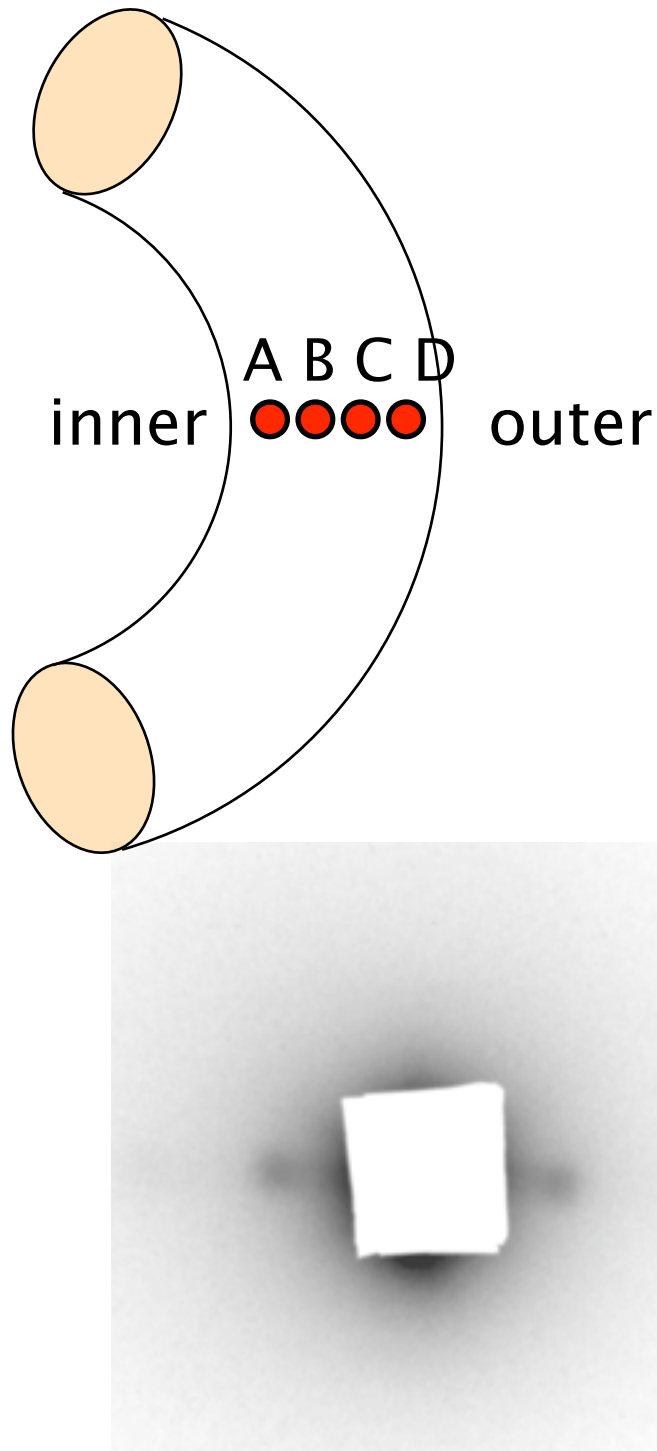
Real space structure



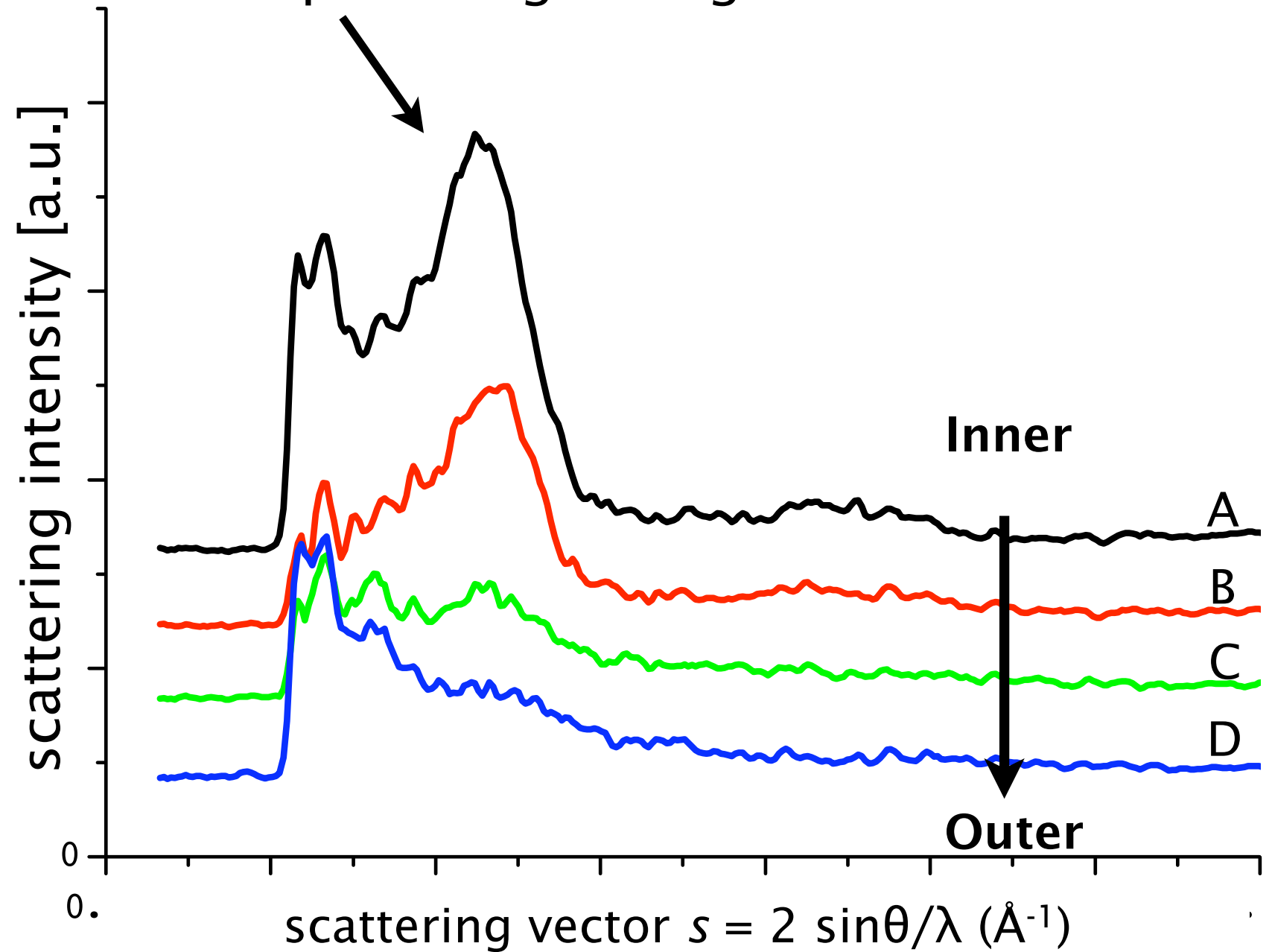
IF-IF Distance



# Diffraction intensity profiles

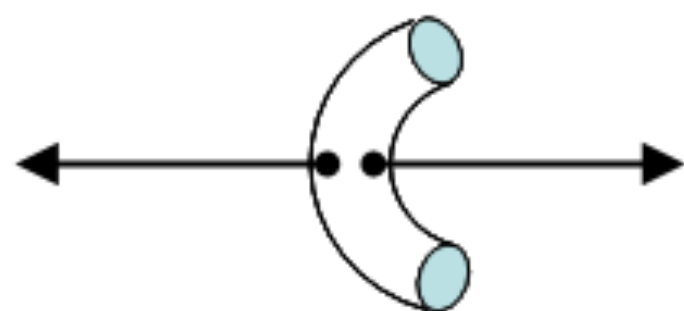
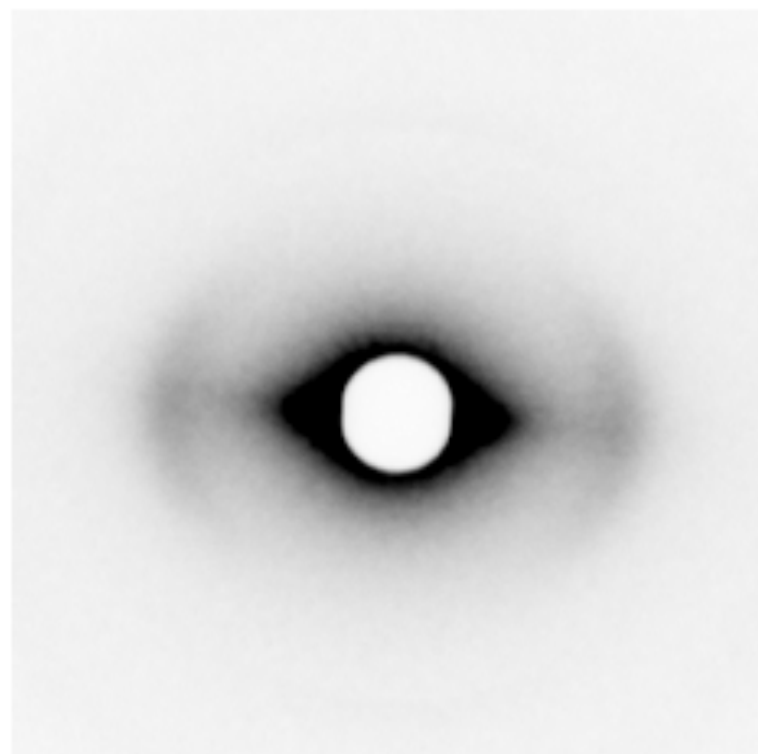


Diffraction peak originating from IF

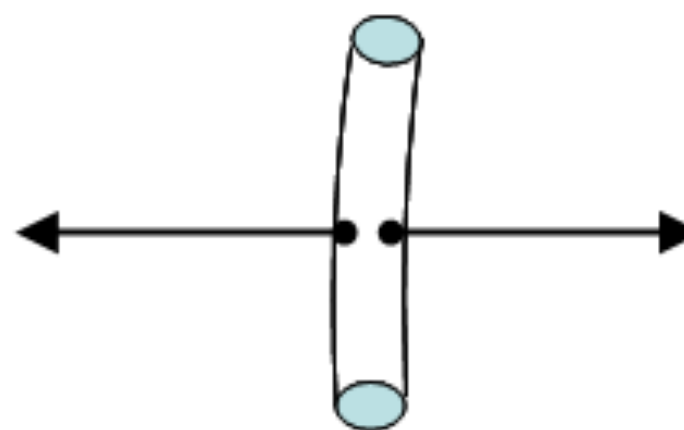
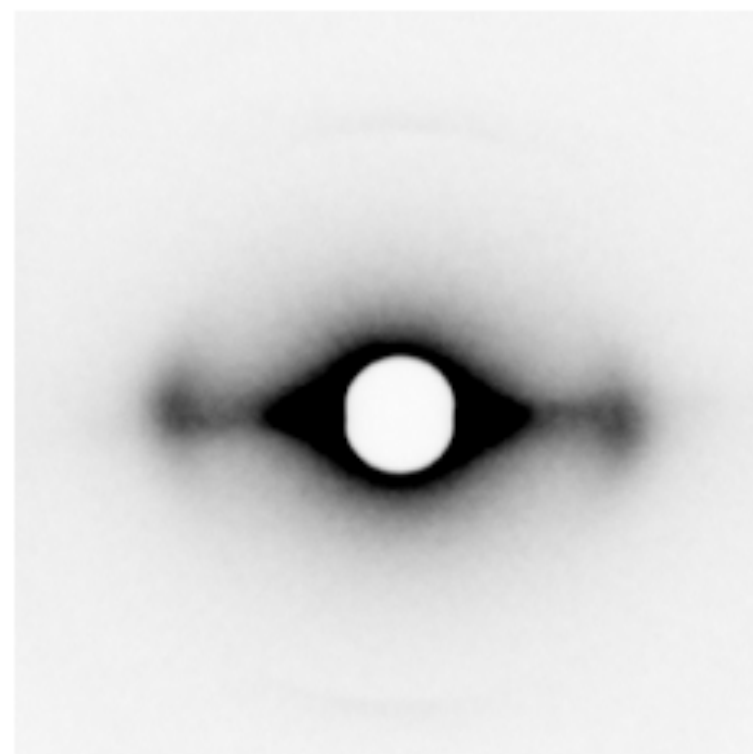


Difference in diffraction intensity  
--> Structural difference in cortex.

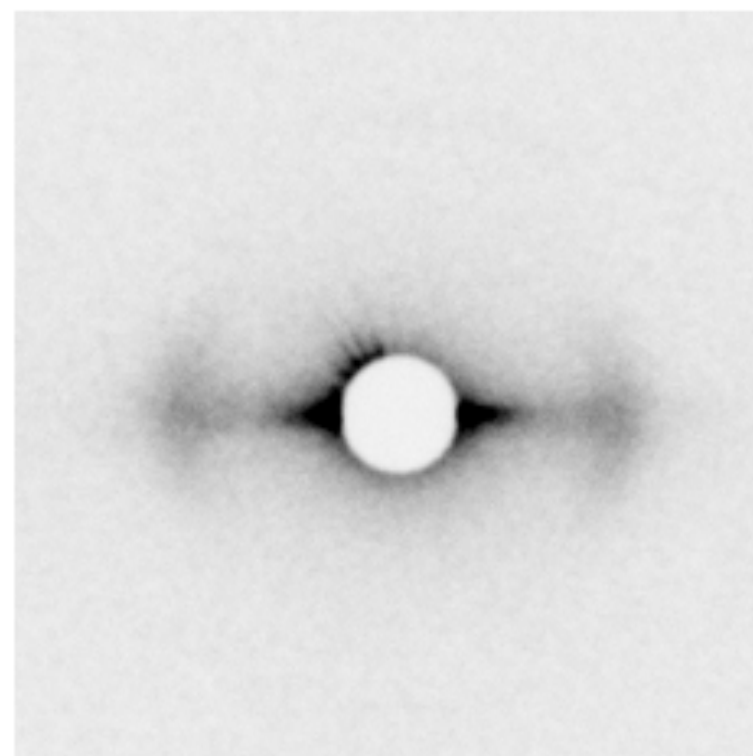




**Curly**  
(ROC = 1.5cm)

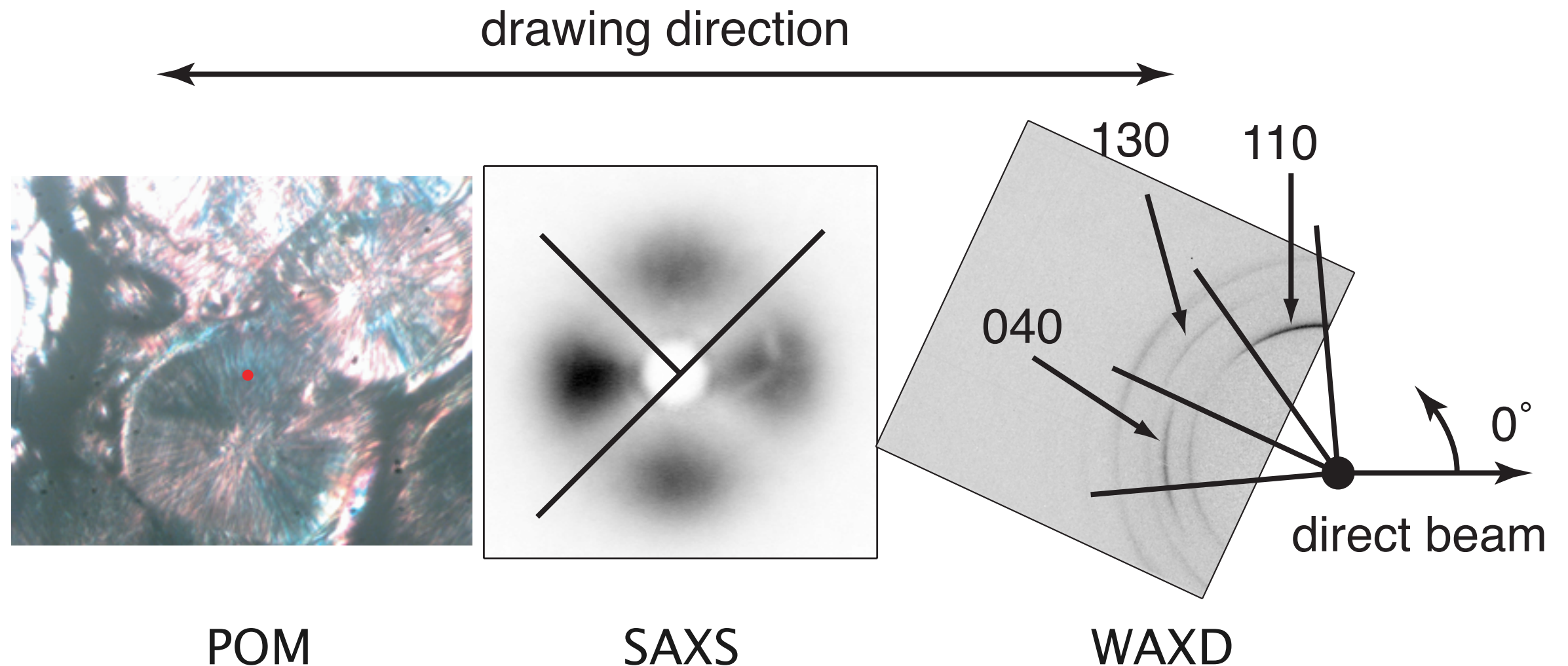


**Nearly Straight**  
(ROC ~ 10cm)



ROC: Radius of Curvature

# Deformation process of spherulite



BL40XU @ SPring-8

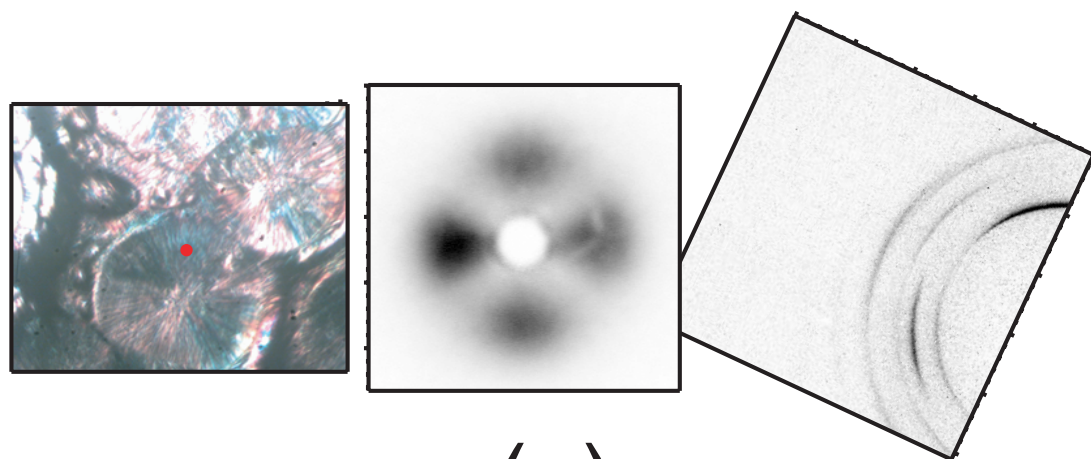
Local deformation manner of polypropylene during uniaxial elongation process



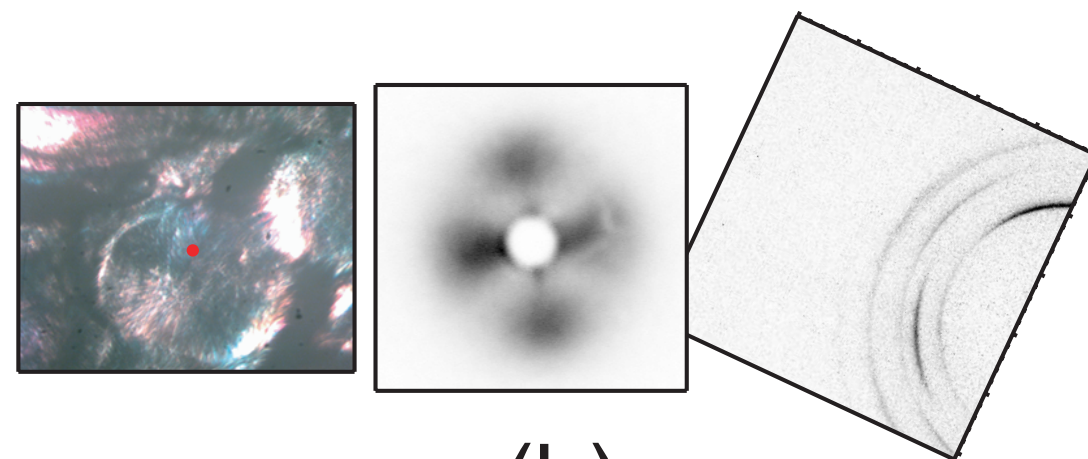
Combined measurement of polarized microscope and microbeam SAXS/WAXD.



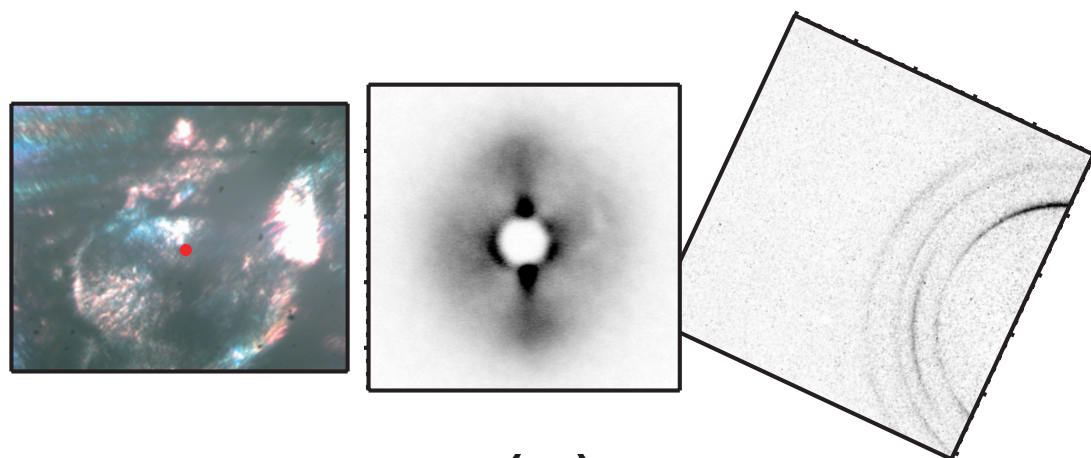




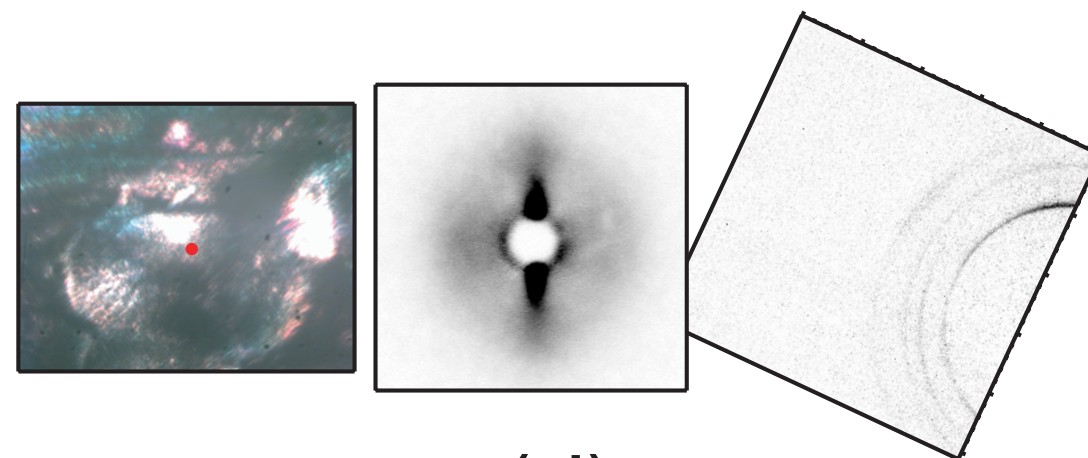
(a)



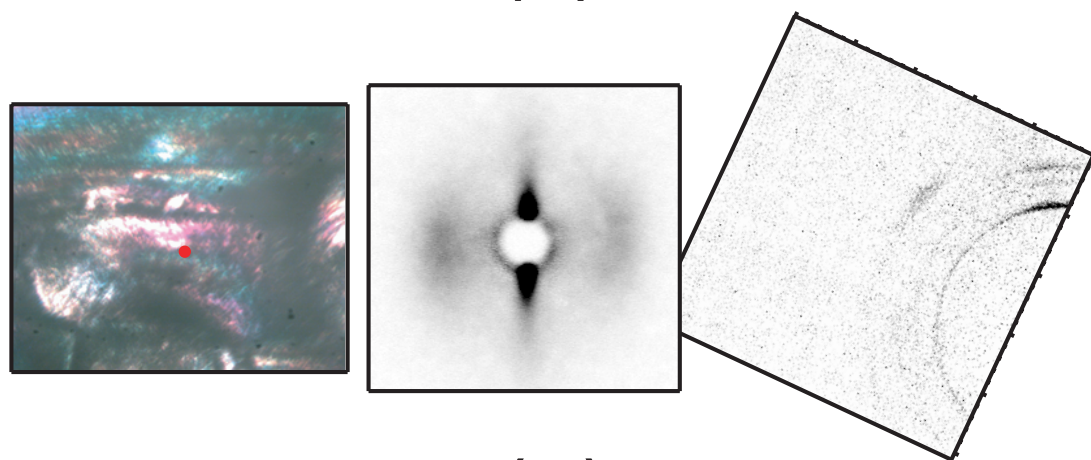
(b)



(c)

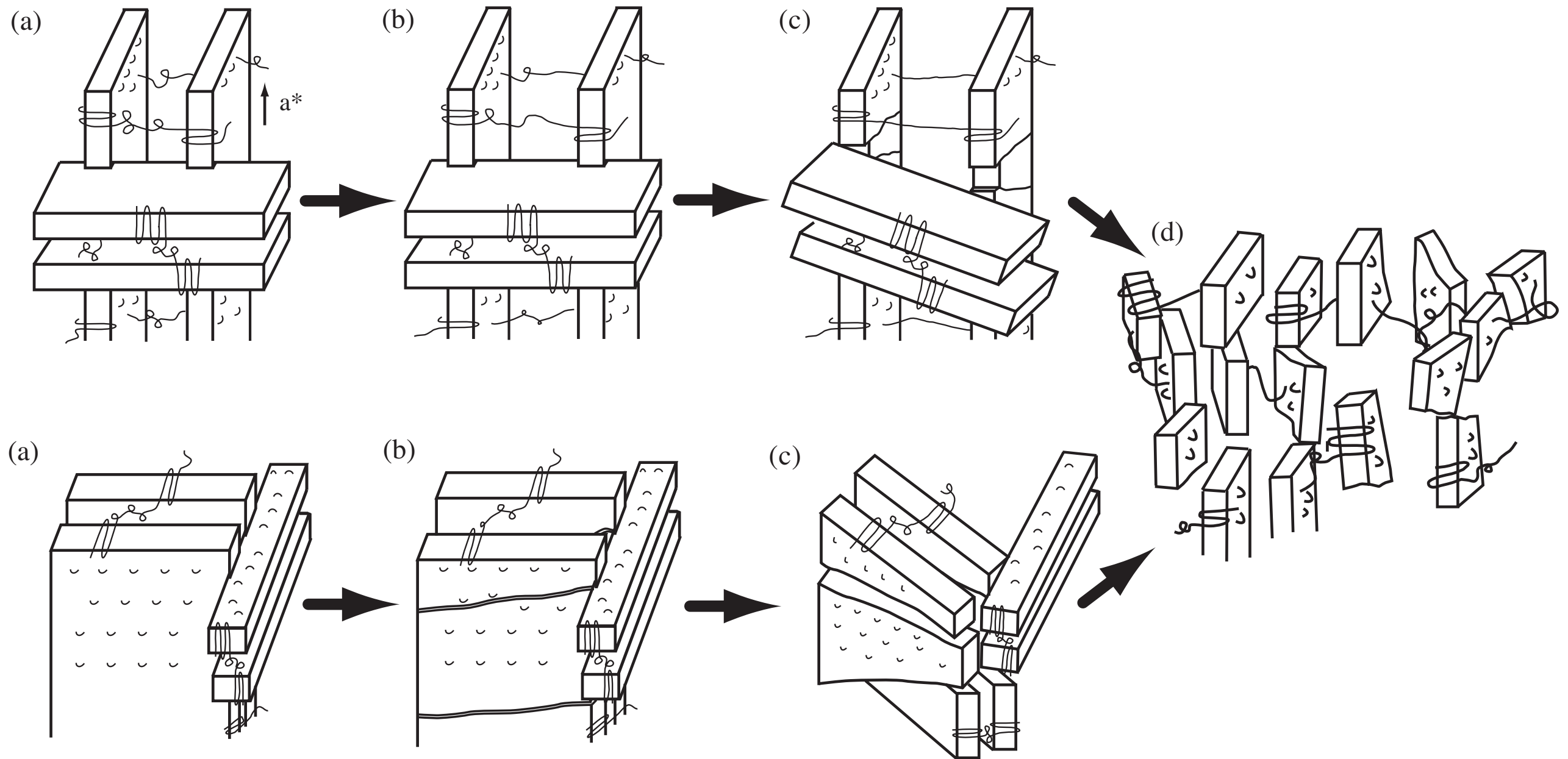


(d)



(e)

# Deformation model of PP



Y. Nozue, Y. Shinohara, Y. Ogawa et al., *Macromolecules*, **40**, 2036 (2007).

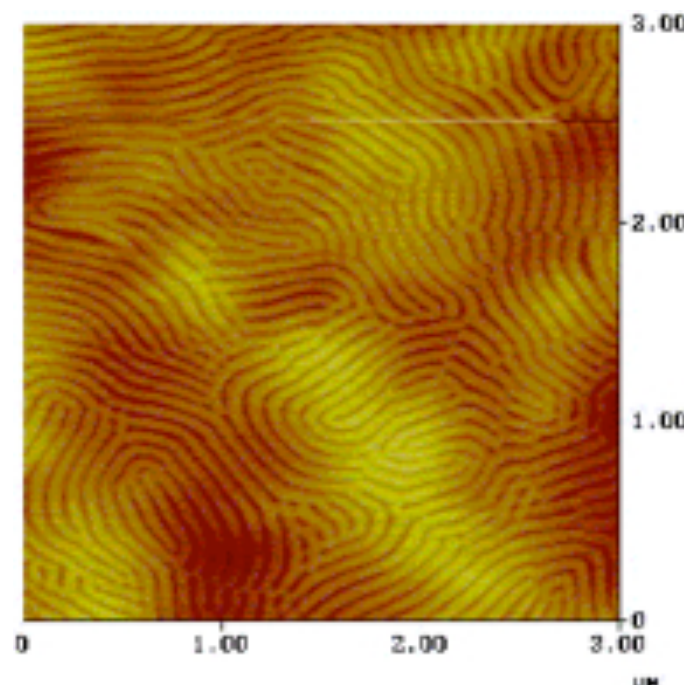
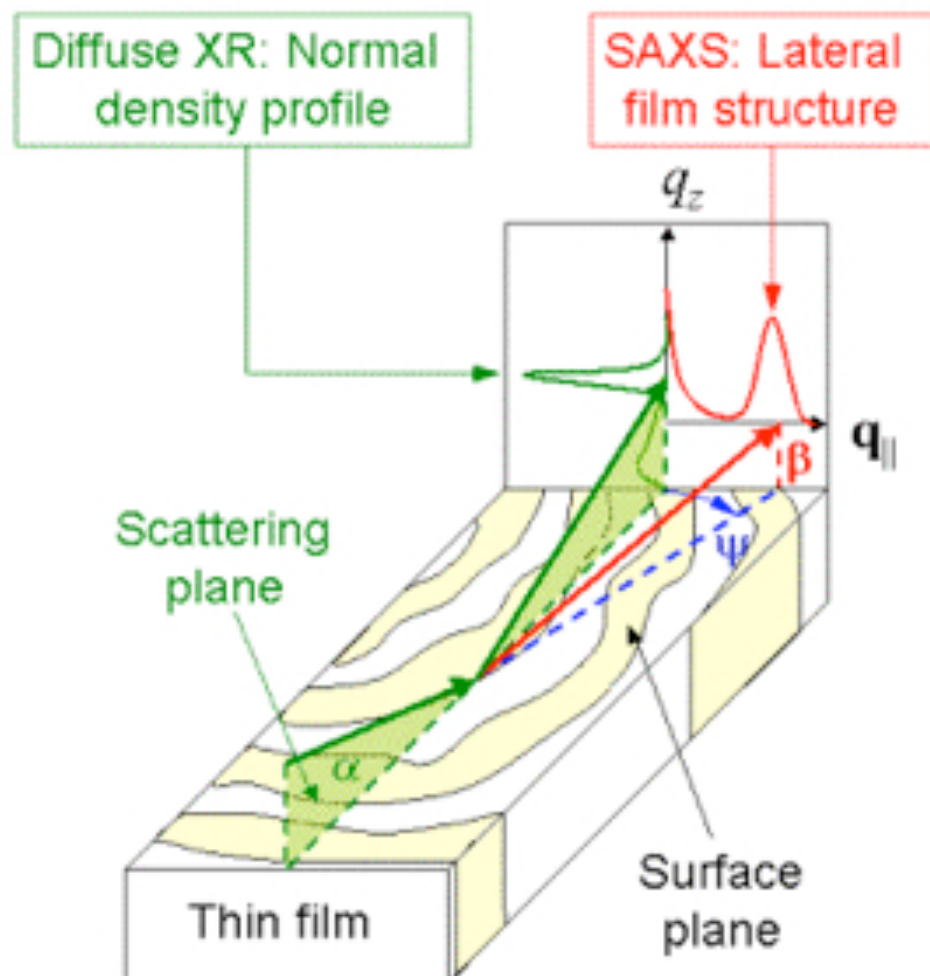


# Grazing Incidence SAXS

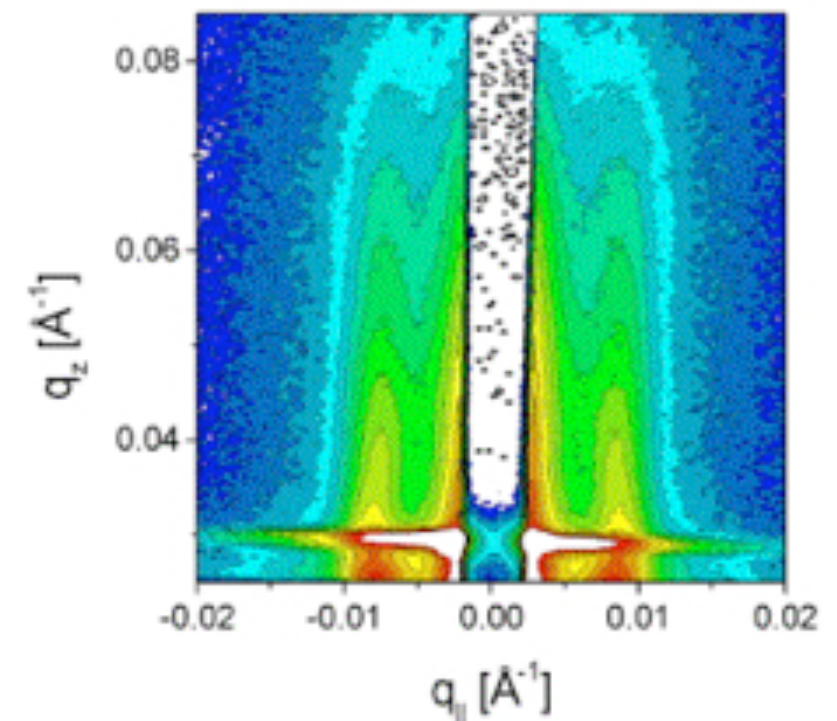
## Advantage

- Surface/interface sensitive (beam footprint).
- In-plane structure and out-of-plane structure can be separated.
- Thin film sample on substrate can be measured.

Ex: from Web page of Dr. Smilgies @ CHESS

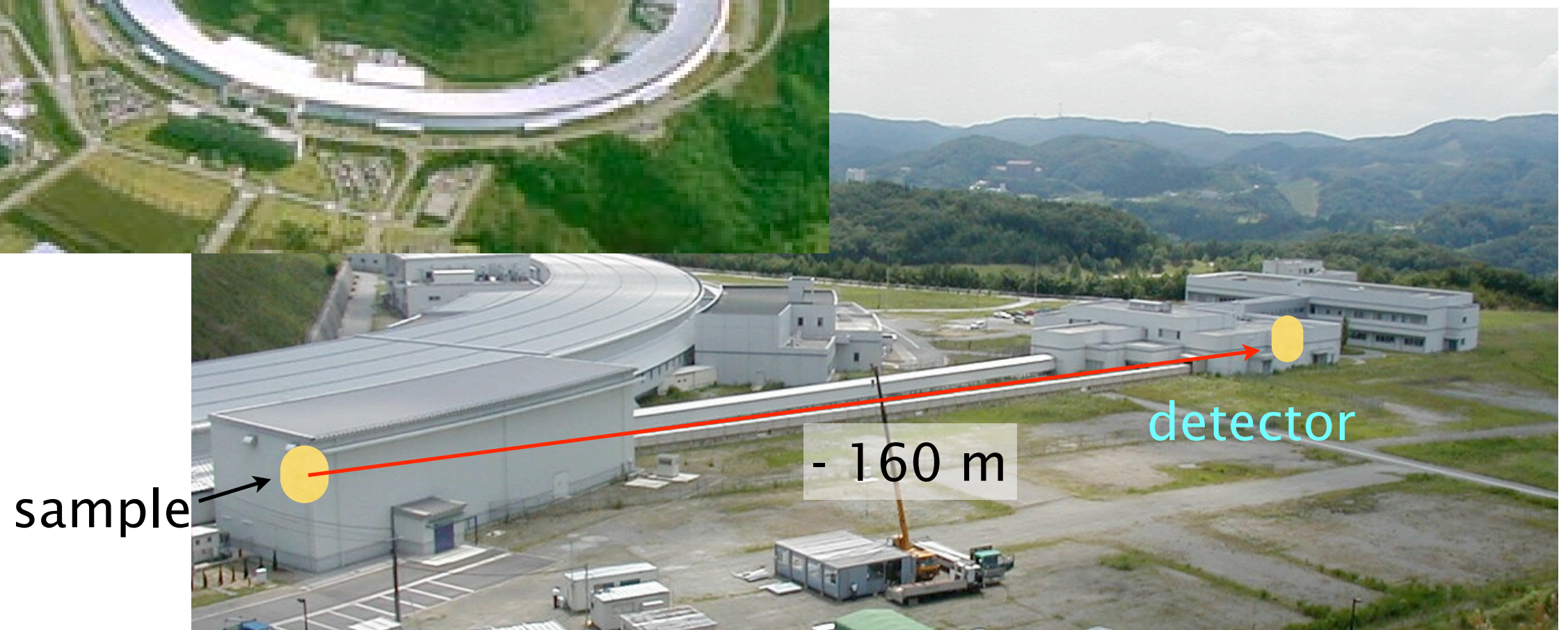


AFM image



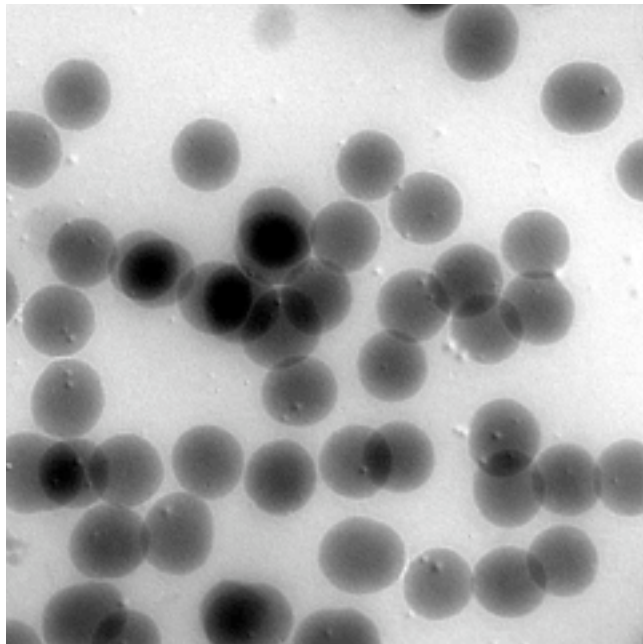
GI-SAXS image

# USAXS using medium-length beamline



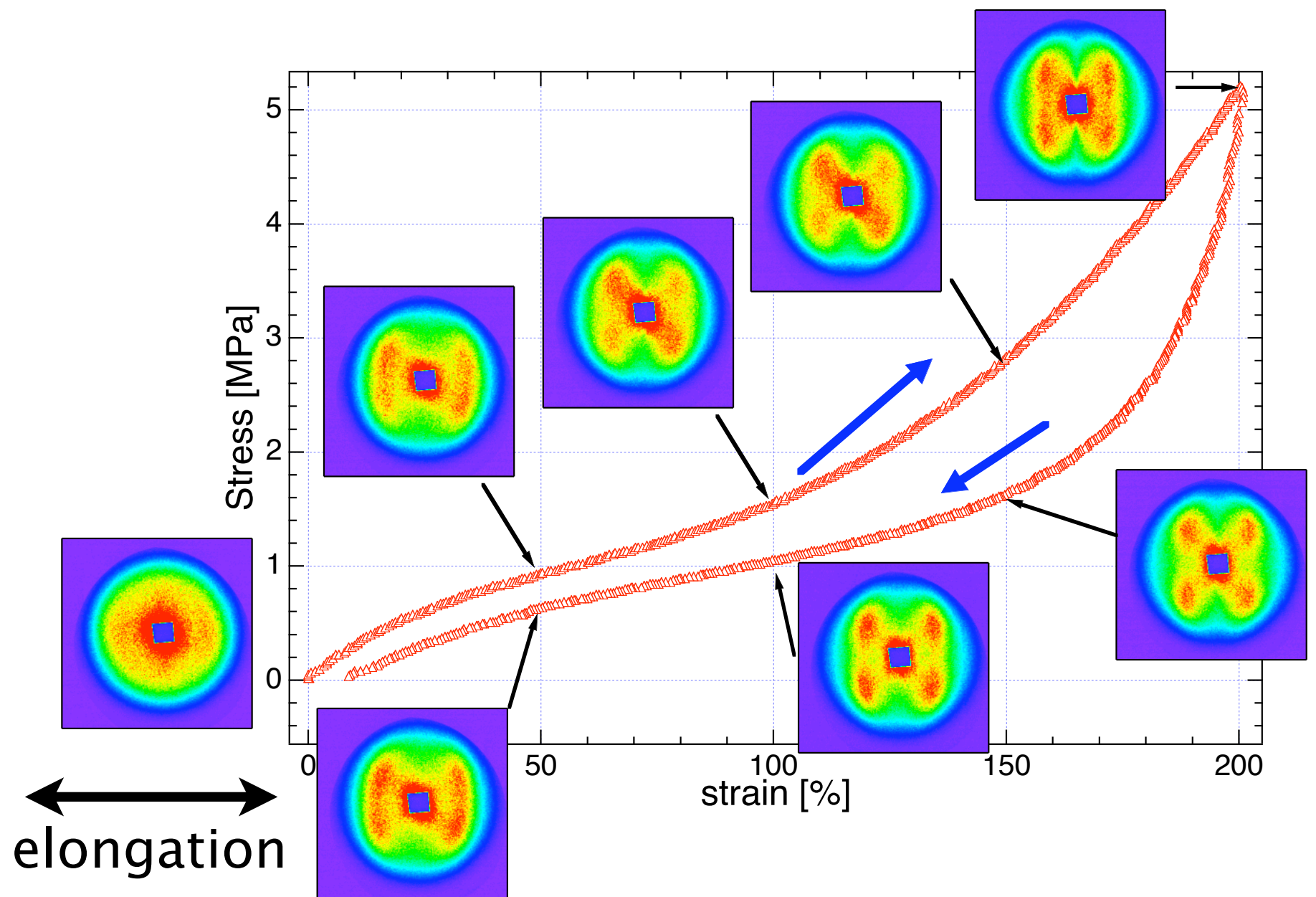


# USAXS patterns from elongated rubber



TEM image

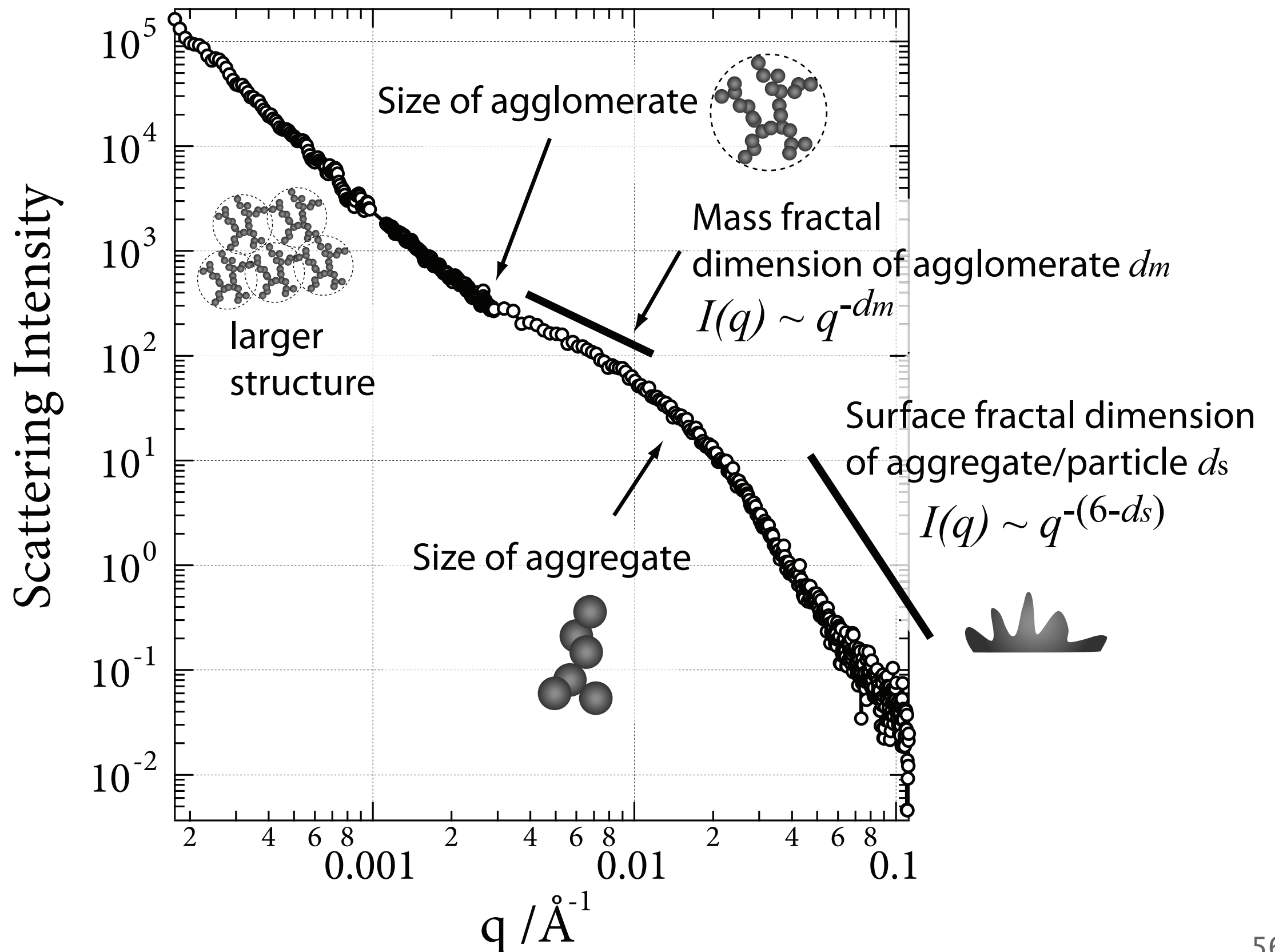
Rubber filled with spherical silica



Scattering pattern also shows hysteresis.

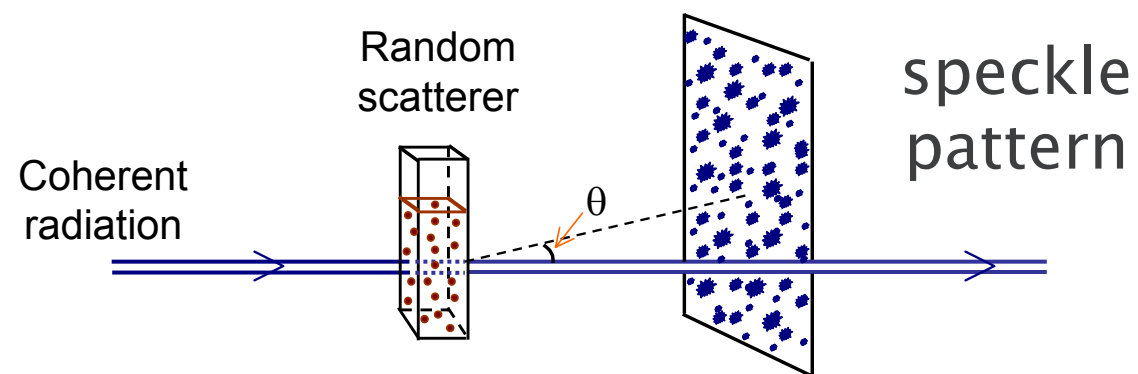


# Structural information from USAXS



# X-ray Photon Correlation Spectroscopy: XPCS

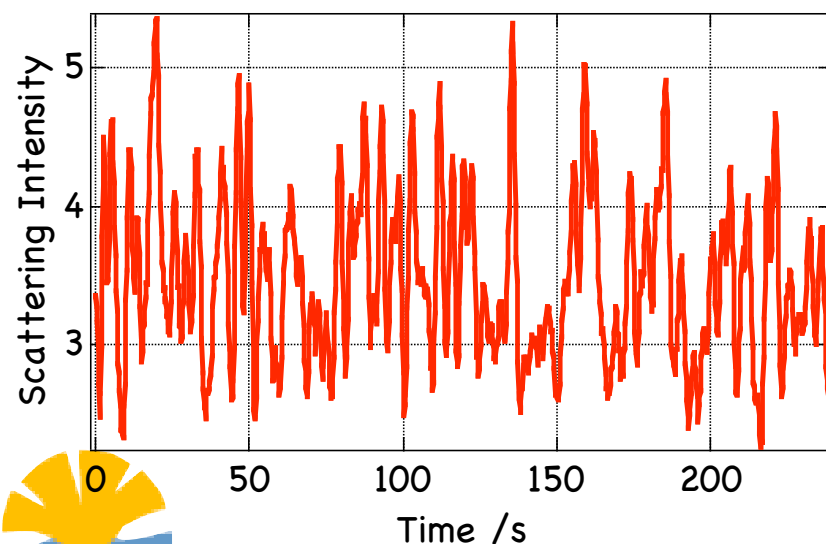
- Measurement of fluctuation of X-ray scattering intensity  
--> Structural fluctuation in sample



$$g^{(2)}(q, \tau) = \frac{\langle I(q, 0) I^*(q, \tau) \rangle}{\langle I(q) \rangle^2}$$

Time-resolved SAXS with coherent X-ray

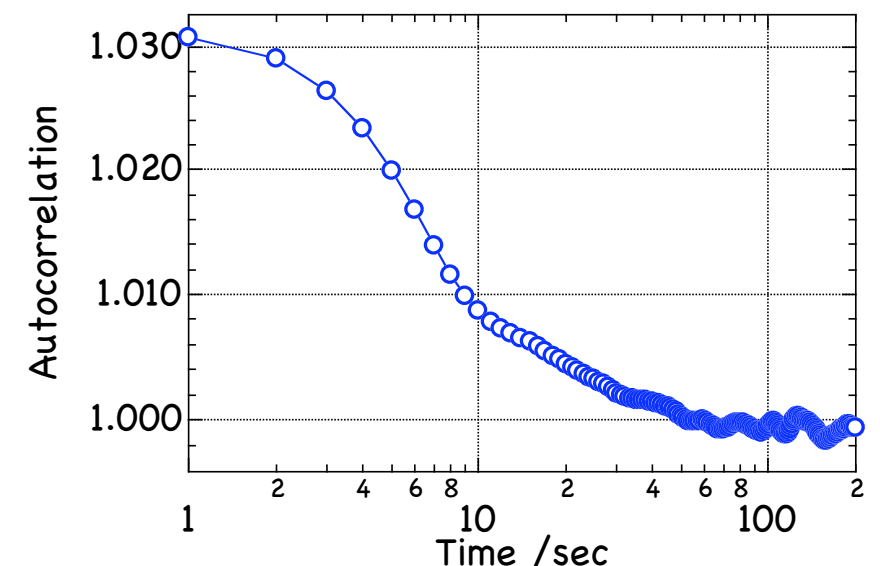
Fluctuation of intensity



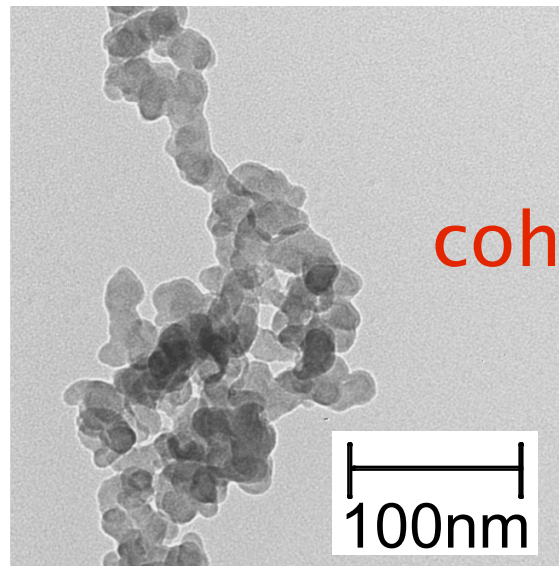
Autocorrelation



relaxation time in system

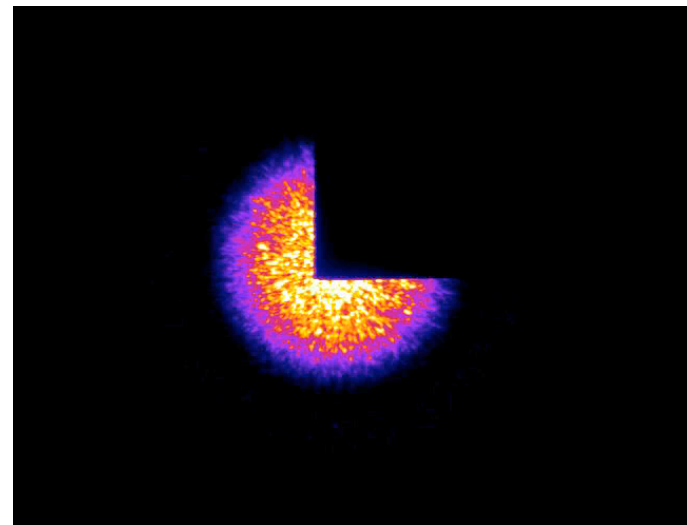


# Dynamics of nanoparticles observed with XPCS

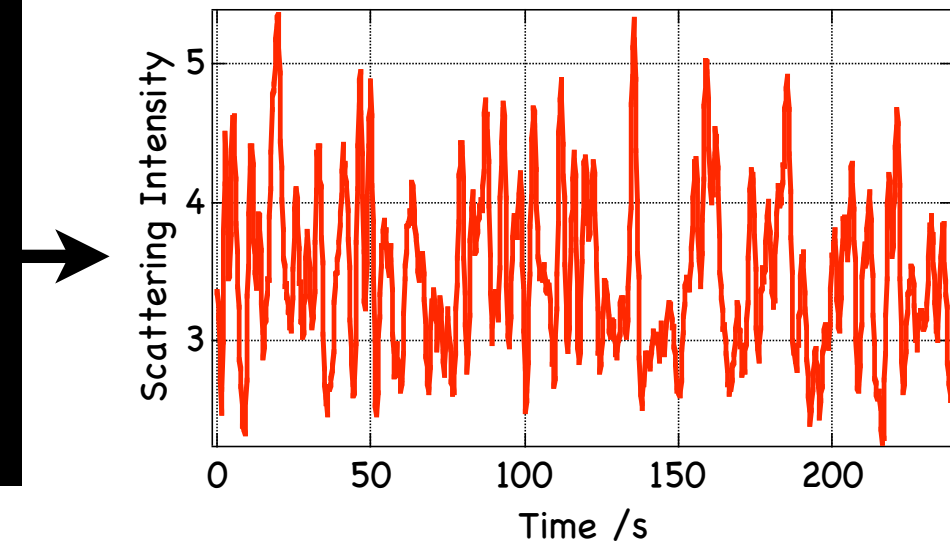


nano-particles in rubber

coherent x-ray



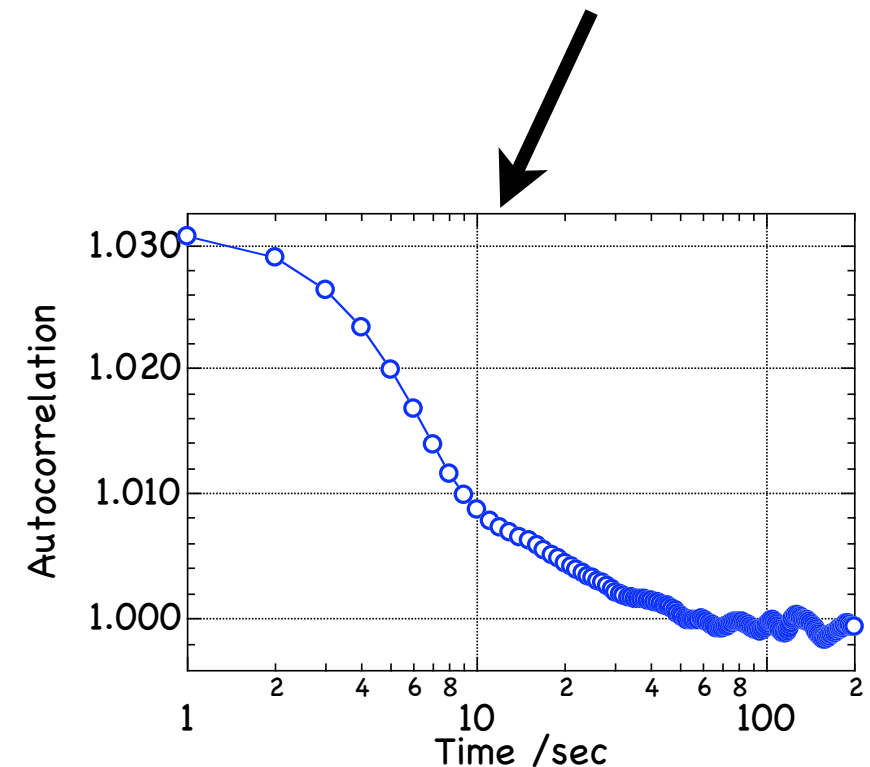
speckle pattern



fluctuation of scattering intensity

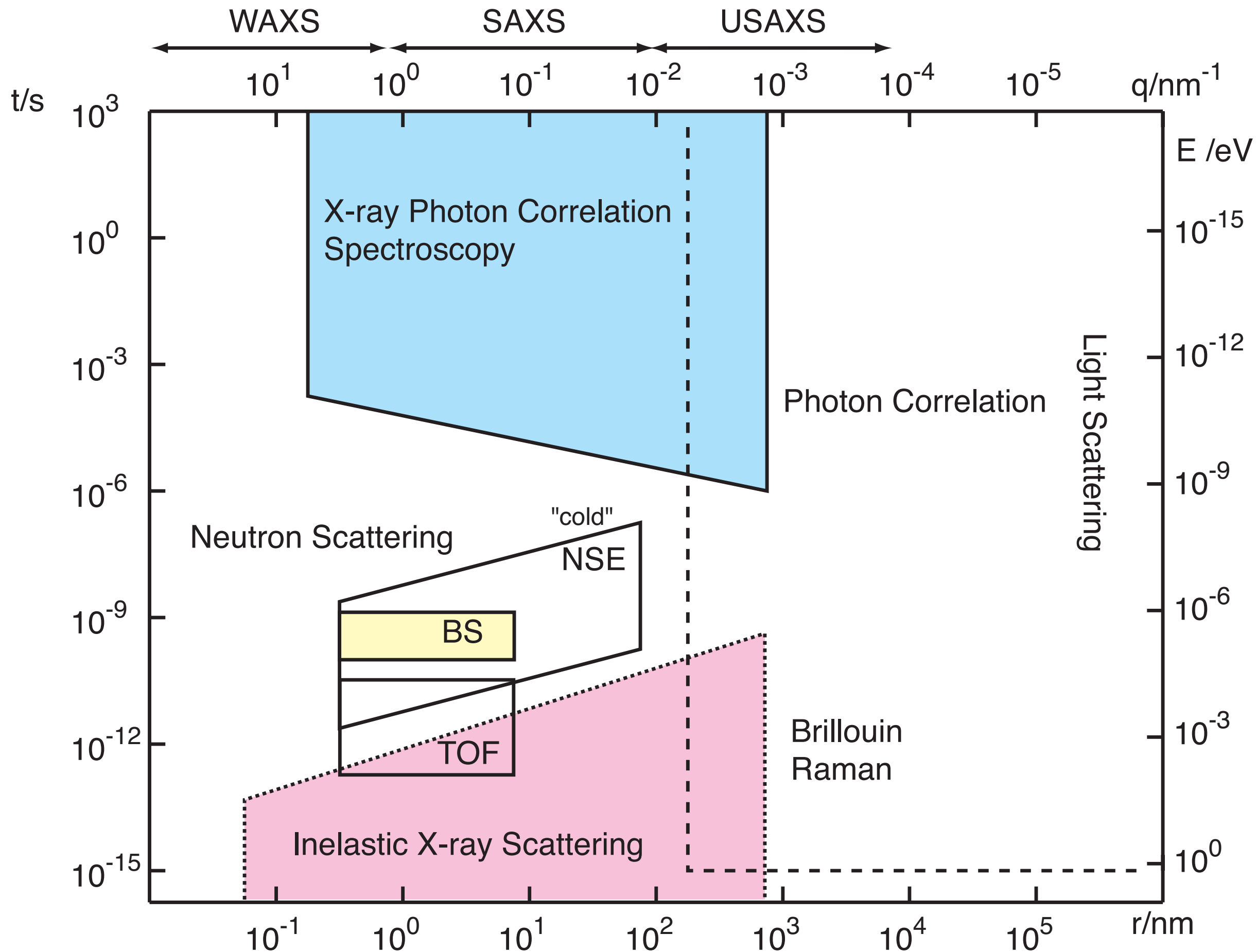
## Dependence of dynamics on...

- Volume fraction of nano-particles
  - Vulcanization (cross-linking)
  - Type of nano-particles
  - Temperature
- etc.



Dynamics of Filler in Rubber







# Bibliography

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- ❧ A. Guinier and A. Fournet (1955) “Small angle scattering of X-rays” Wiley & Sons, New York. **out-of-print**
- ❧ O. Glatter and O. Kratky ed. (1982) “Small Angle X-ray Scattering” Academic Press, London. **out-of-print**
- ❧ L. A. Feigin and D. A. Svergun (1987) “Structure Analysis by Small Angle X-ray and Neutron Scattering” Plenum Press. **out-of-print ?**
- ❧ P. Lindner and Th. Zemb ed. (2002) “Neutron, X-ray and Light Scattering: Soft Condensed Matter”, Elsevier.
- ❧ Proceedings of SAS meeting (2003 & 2006). Published in J. Appl. Cryst.
- ❧ R-J. Roe (2000) “Methods of X-ray and Neutron Scattering in Polymer Science”, Oxford University Press.

